

Coherent X-rays & applications



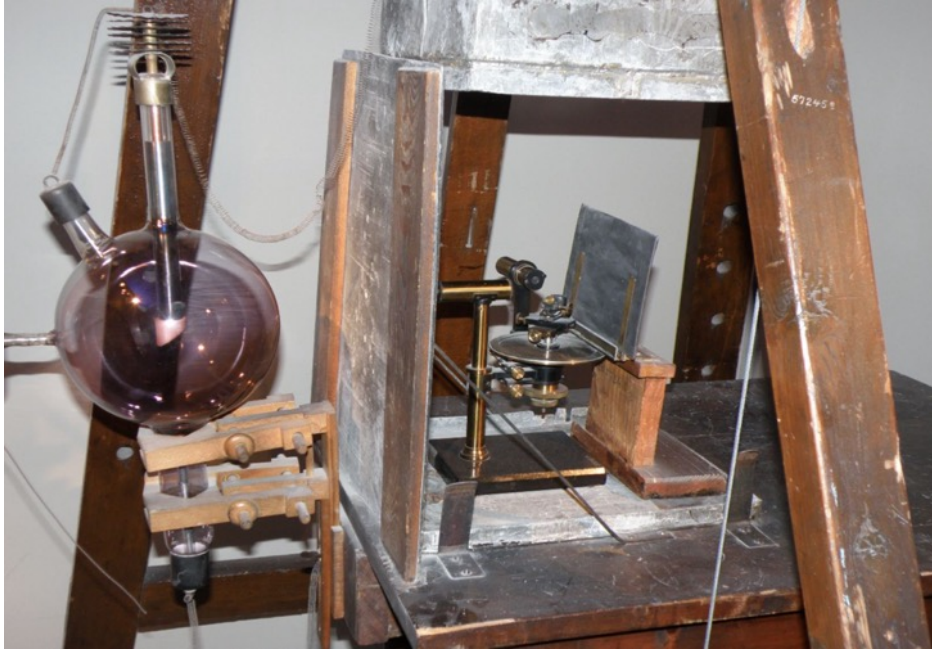
The European Synchrotron

Vincent Favre-Nicolin
Algorithms & scientific Data Analysis

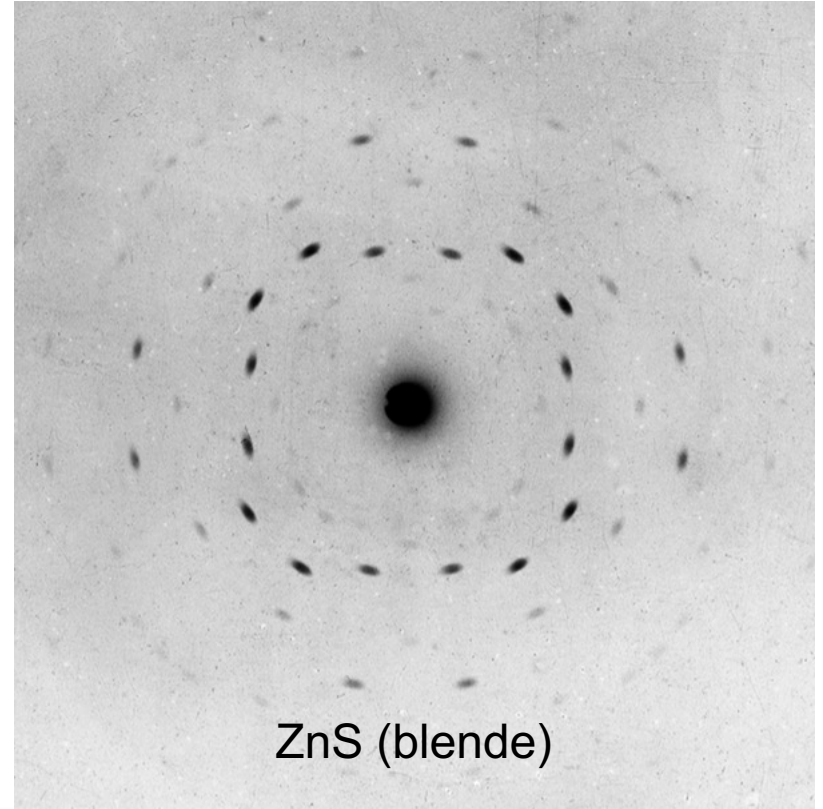
on leave from Univ. Grenoble Alpes



COHERENT X-RAYS ?



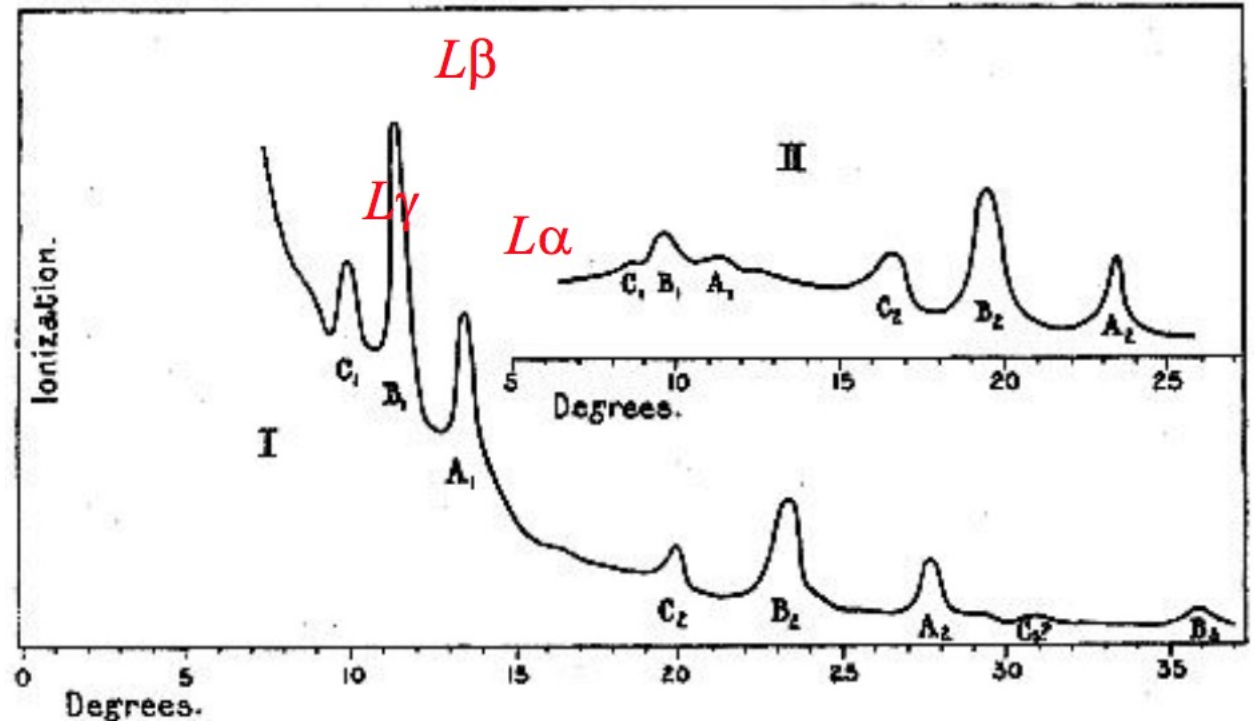
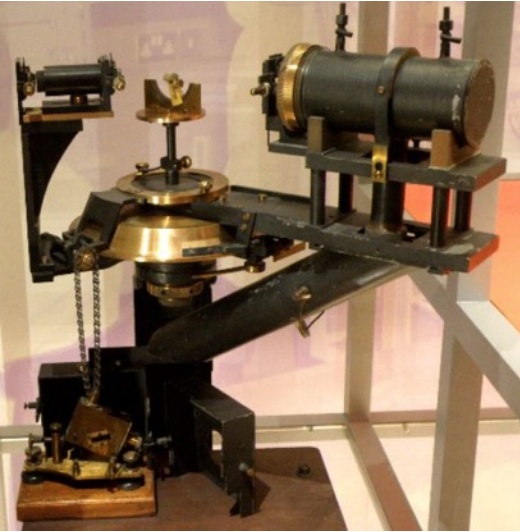
Friedrich, Knipping, Laue



ZnS (blende)

Illustrations from Authier « Early days of crystallography, Oxford University Press

COHERENT X-RAYS ?



W & H Bragg : first X-ray spectrum : NaCl diffraction from Pt radiation

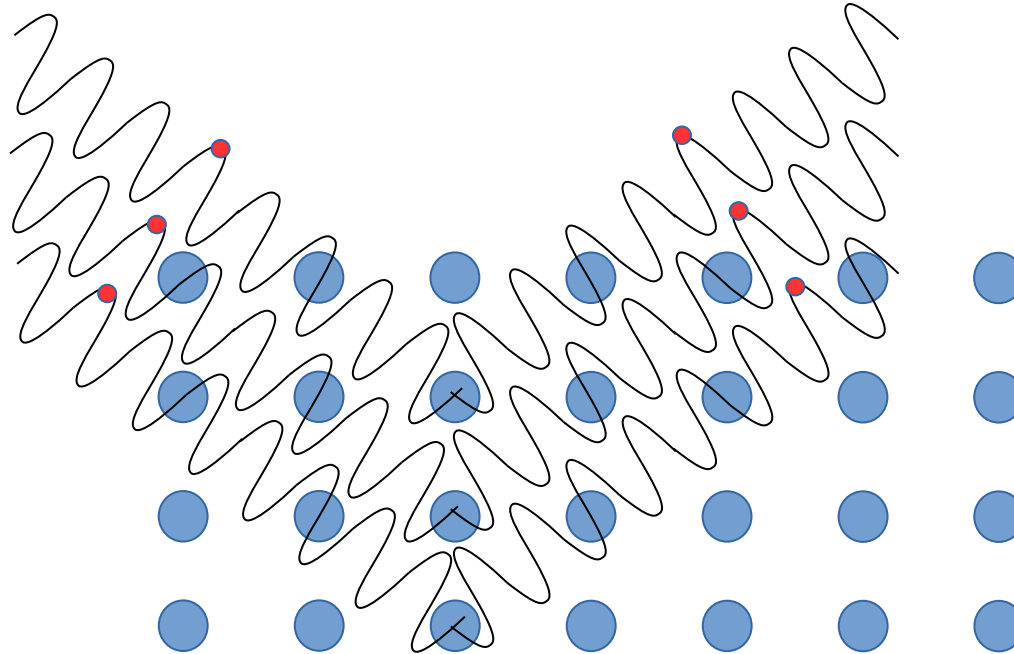
Illustrations from Authier « Early days of crystallography, Oxford University Press

COHERENT X-RAYS ?

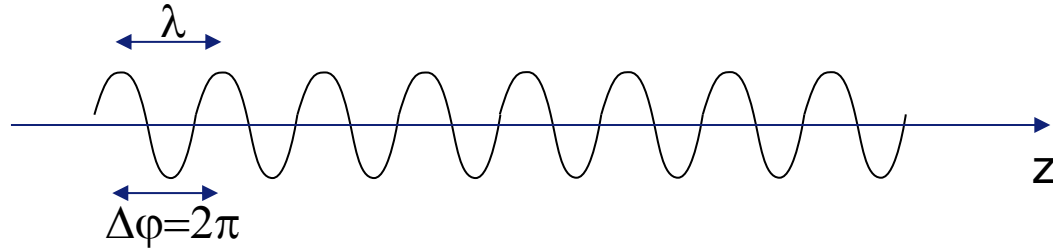
X-rays can produce (Bragg) diffraction even from “low-quality” X-ray sources :

- Large source size
- Non-monochromatic (but discrete spectrum)

Diffraction implies interferences => coherence



WAVE PROPAGATION BASICS



$$A_0 \cos(\omega t - kz) = A_0 \cos\left(\omega t - \frac{2\pi z}{\lambda}\right) = A_0 \cos(\varphi)$$

- λ : wavelength (0.01 to 1 nm for X-rays)
- ω : pulsation = $2\pi/\nu$
- ν : frequency
- φ : phase

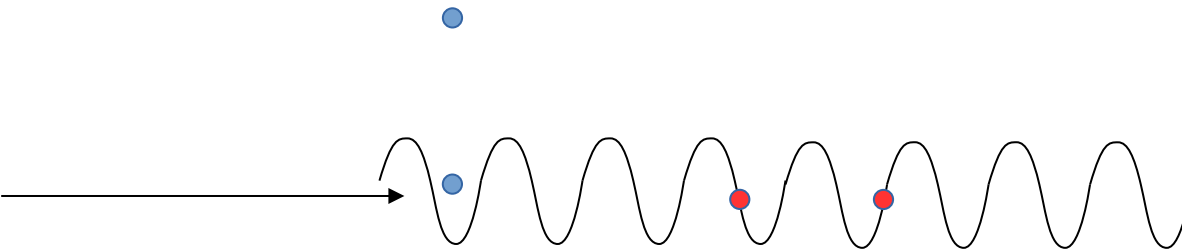
COHERENT XRAYS ?

Definition (from Malcolm Howells):

- “Optical coherence exists in a given radiating region if the **phase differences between all pairs of points** in that region have definite values which are constant with time”

Two types of coherence must be considered:

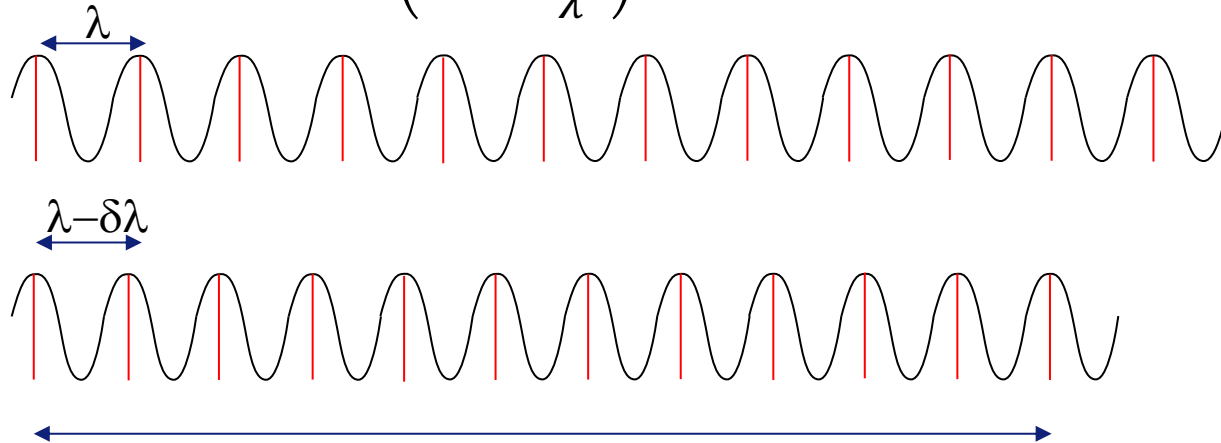
- Transverse coherence, **between points in plane perpendicular to the wavevector**
- Longitudinal (temporal) coherence **between points along the wavevector**



LONGITUDINAL COHERENCE

- Longitudinal (temporal) coherence
- Related to the monochromaticity
- For a beam with $\delta\lambda$ spectrum width (typical $\delta\lambda/\lambda=10^{-4}$: for Si 111 monochromator)

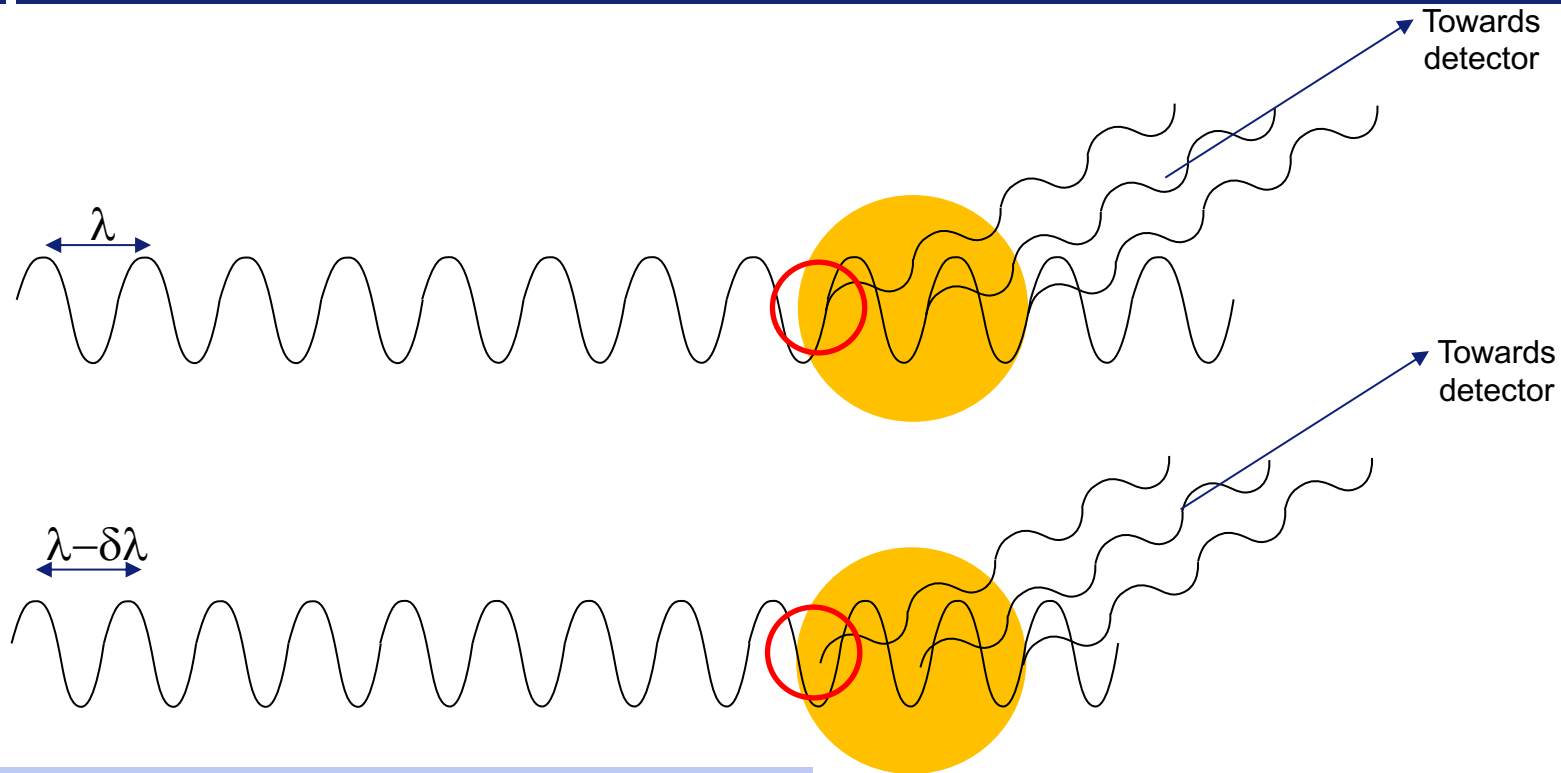
$$A_0 \cos\left(\omega t - \frac{2\pi z}{\lambda}\right) = A_0 \cos(\varphi)$$



Longitudinal coherence length = length for a 2π shift due to $\delta\lambda$.

$$\Lambda_{long} = \lambda \frac{\lambda}{\delta\lambda} = \frac{\lambda^2}{\delta\lambda} \quad \Rightarrow \text{typically } 0.5 \text{ to } 1 \mu\text{m}$$

LONGITUDINAL COHERENCE & SCATTERING



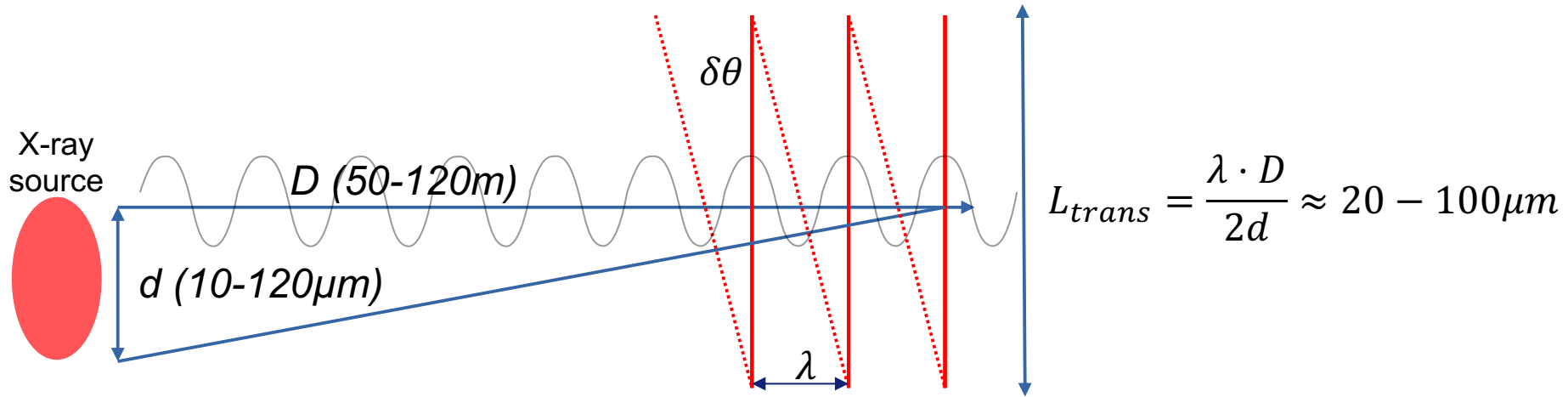
There is a **phase difference** between the beginning and the end of the particle.

As long as this difference is small, it does not affect the scattering signal

For crystallography, the coherence length is much larger than a unit cell ($\sim 1\text{nm}$), which is why we can always see Bragg diffraction with 'low' coherence

TRANSVERSE COHERENCE

- Transverse coherence, **between points in plane perpendicular to the wavevector**
- Related to the source size
- X-ray sources (tubes, bending magnet, undulators, ~~XFEL~~) are incoherent (different points within the source emit with random phase shifts)
- For a source width d (a few 10's of μm), seen from a distance D (50-120m):



There is a **phase difference** between incident waves emitted by different parts of the source.

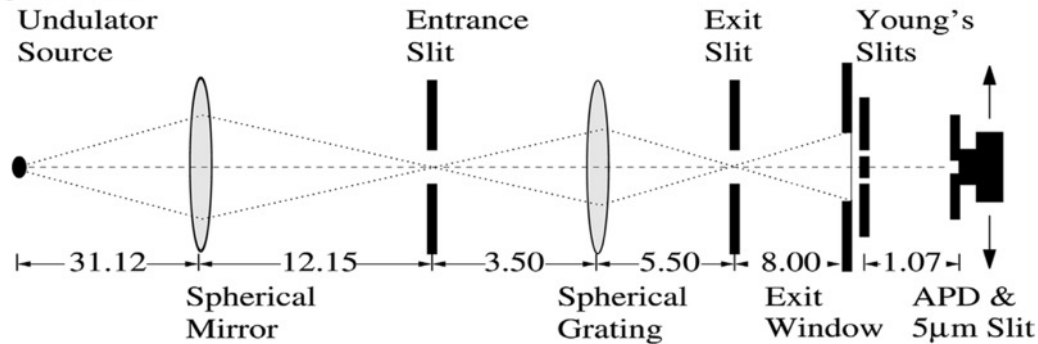
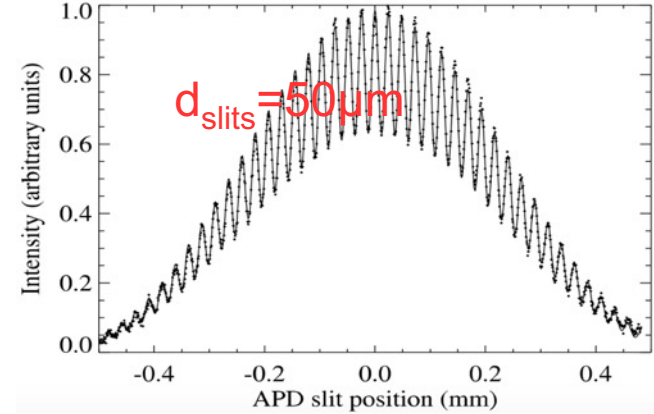
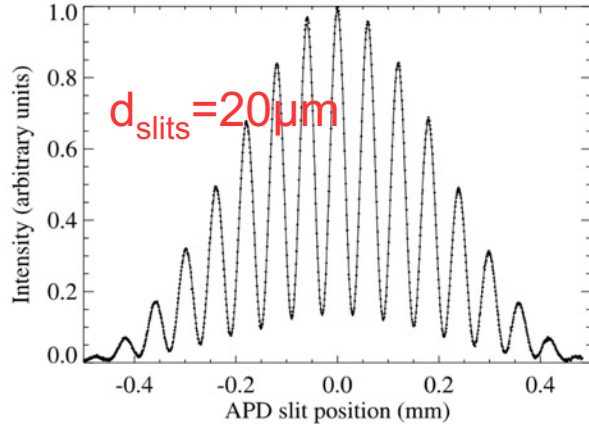
As long as this difference is small, it does not affect the scattering signal

High coherence implies:

- Small sources
- Long beamlines

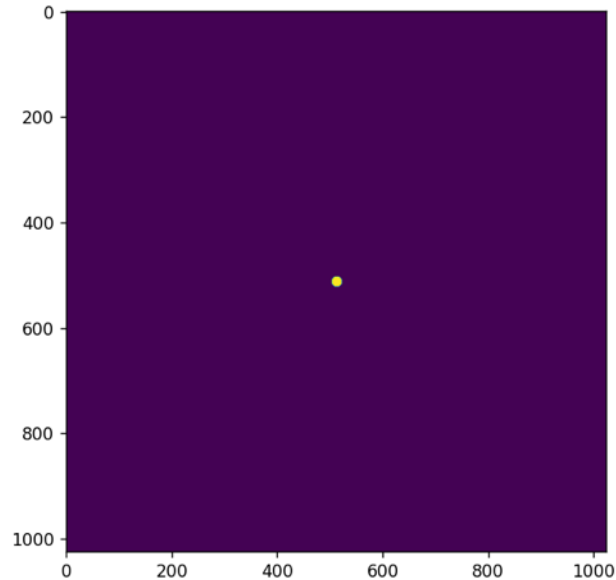
TRANSVERSE COHERENCE: YOUNG SLITS

- Transverse coherence can be evaluated by fringe visibility:



Fringes for a Young double-slit experiment @1.1 keV, 8m from the source

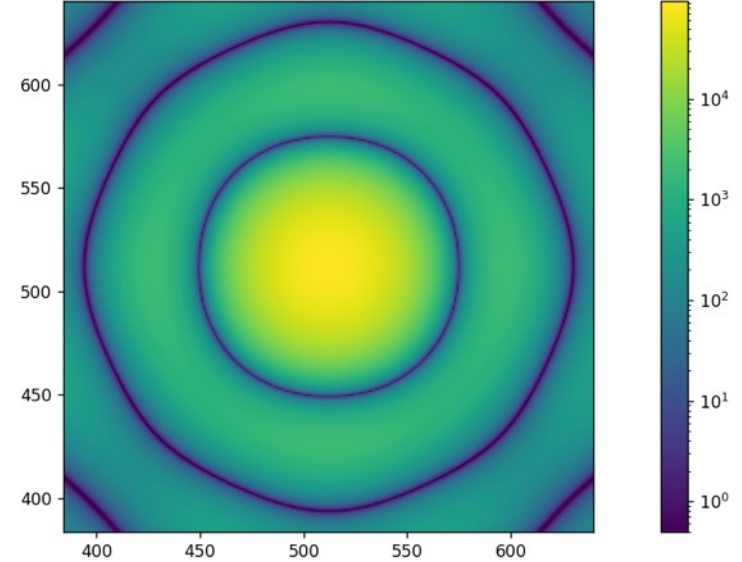
COHERENT SCATTERING: 1 PARTICLE



Circular particle
($r=10$ pixels)



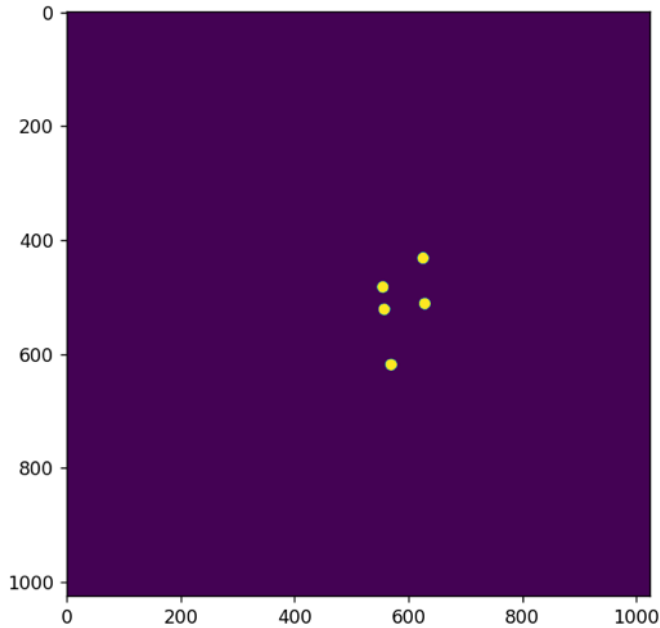
Fourier
transform



Detector (20 cm to a few m away)
Oscillations with a period in $1/r$

The scattering from a **coherently illuminated** particle can be simply computed using a Fourier transform

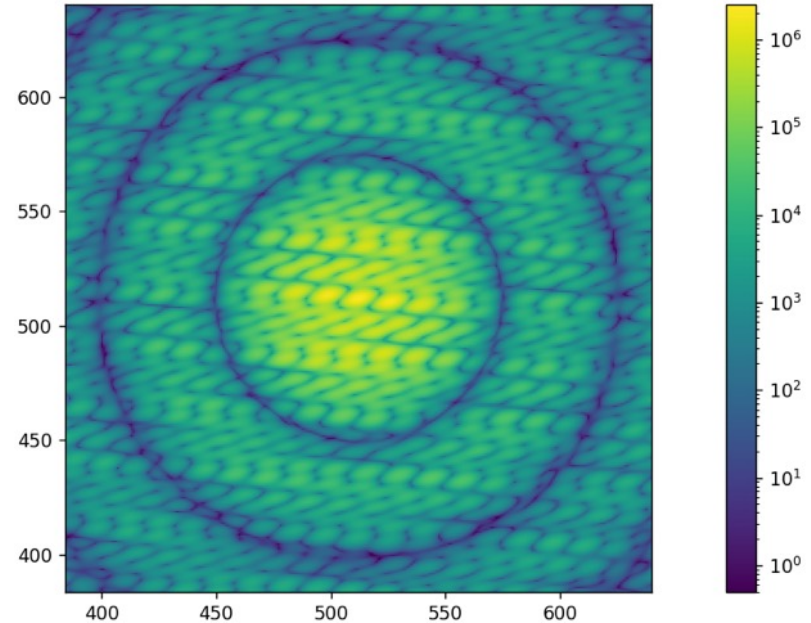
COHERENT SCATTERING: 5 PARTICLES



N=100 circular particles
($r=10$ pixels)



Fourier
transform

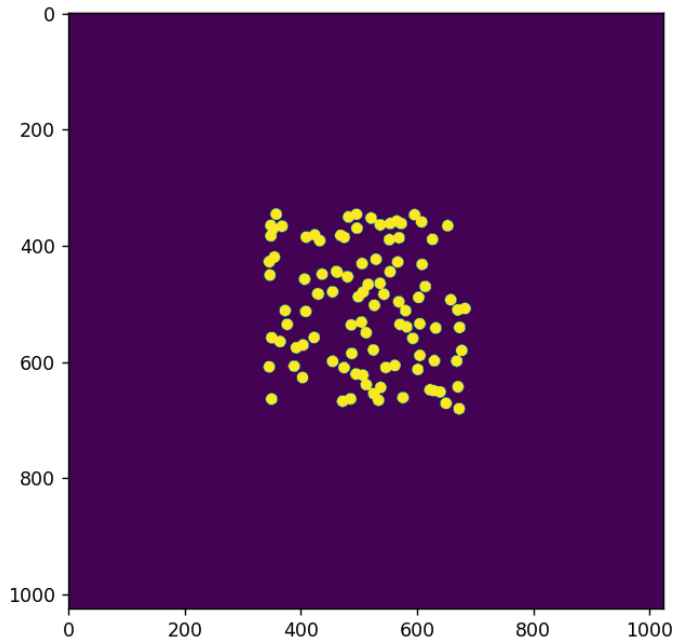


Detector (20 cm to a few m away)
Oscillations with a period in $1/r$

If the particles were *not* coherently illuminated, the detector image would be exactly the same as for a single particle.

+ 'speckles' due to interferences
between the scattering of all particles

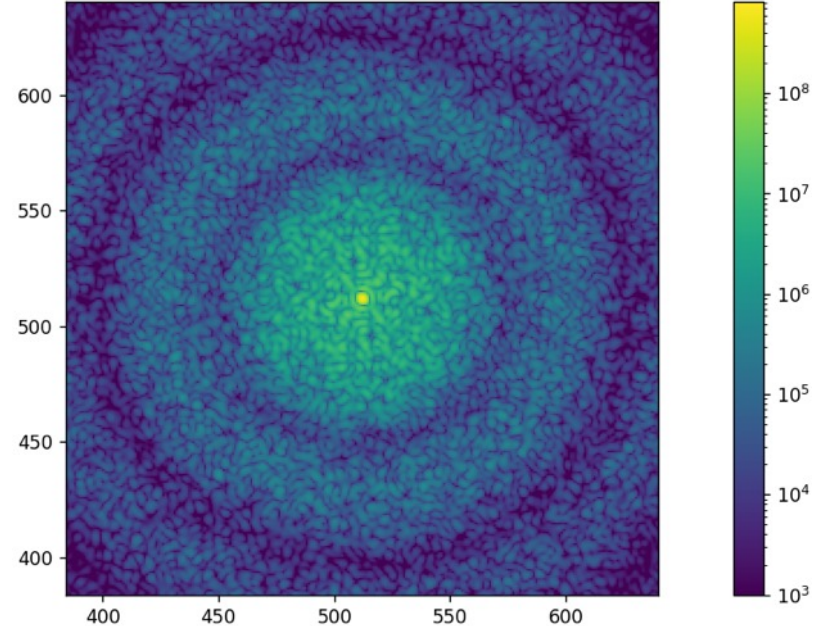
COHERENT SCATTERING: 100 PARTICLES



N=100 circular particles
($r=10$ pixels)



Fourier
transform

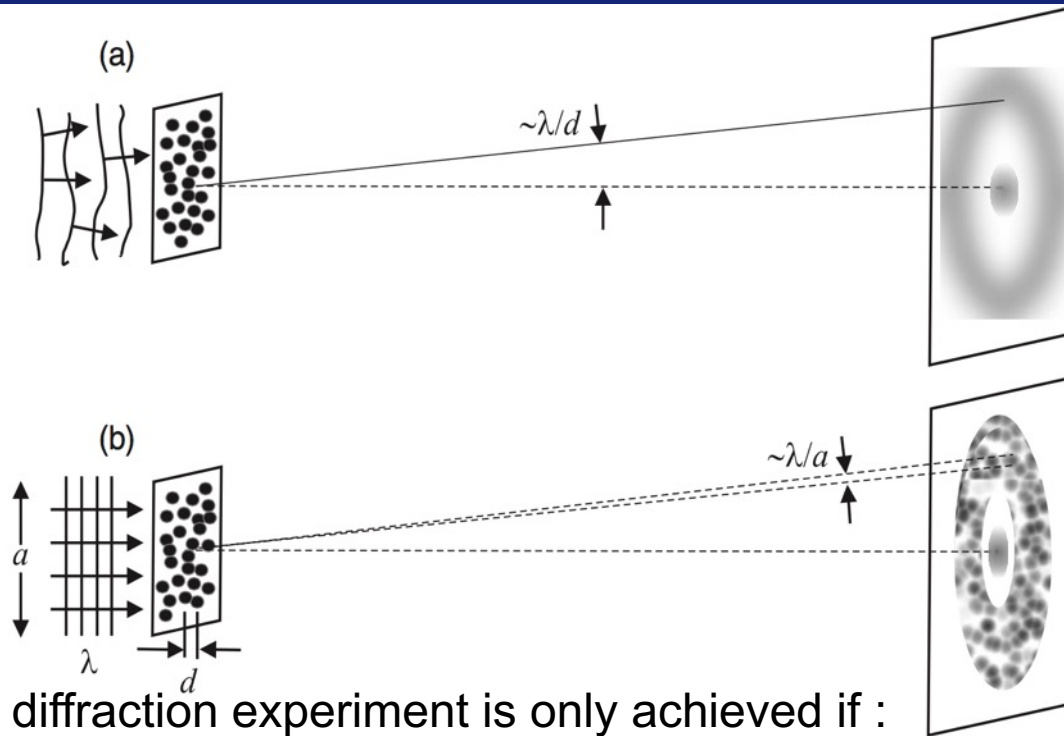


Detector (20 cm to a few m away)
Oscillations with a period in $1/r$
+ 'speckles' due to interferences
between the scattering of all particles

Features which can be extracted:

- Average particle size from oscillations
- Average distance between particles from speckles
- Time evolution

COHERENT ILLUMINATION

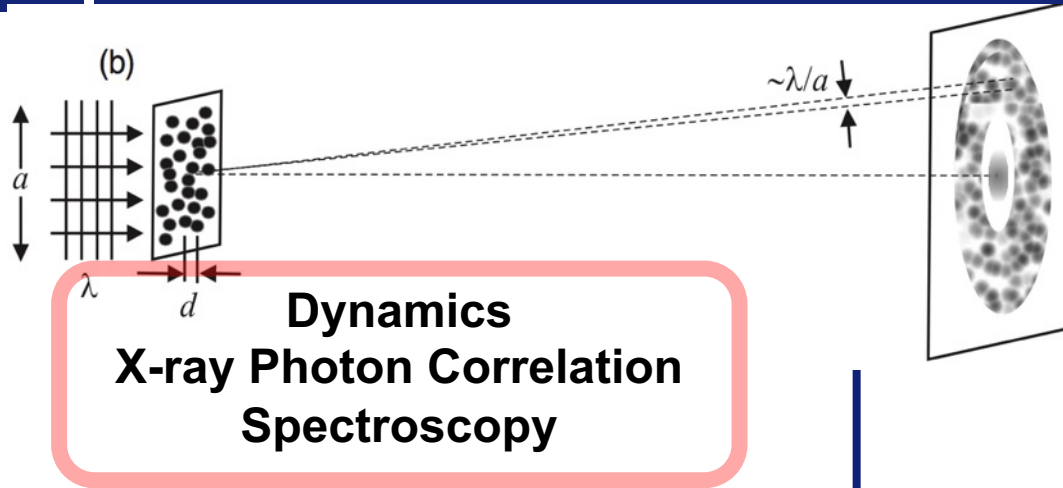


A coherent diffraction experiment is only achieved if :

- The sample is smaller than the coherent length
- Or the beam is collimated to a smaller size than the coherent length

In 3D, the term 'coherent volume' can be used

COHERENT X-RAYS: DYNAMICS & IMAGING



The diffraction pattern is the sum of the interference of the scattering from all particles

$$A(\vec{s}) = \sum \rho_i e^{2i\pi\vec{s}\cdot\vec{r}_i}$$

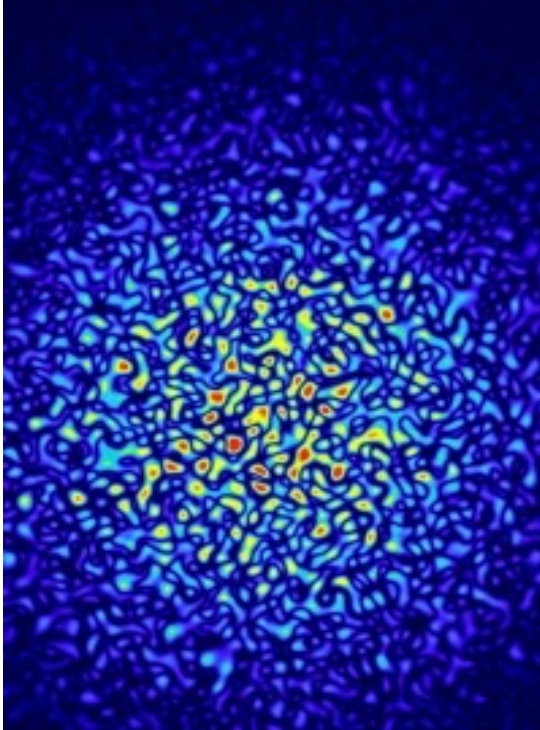
High-resolution imaging

- Study the frame-by-frame evolution of the diffraction pattern (image cross-correlation)
- Information about the evolution of relative position between particles
- Study **diffusion coefficients**, from atomistic to mesoscopic to length scales (colloids, glasses, magnetic systems, etc...)

- The diffraction image obtained by **Fourier Transform(s)** of the illuminated object
- Once the phase has been recovered (algorithms), the **object can be reconstructed** (in **2D** or **3D**)
- The **resolution** is inversely proportional to the angular extent of the scattering (far field)

COHERENT ILLUMINATION

If the object is too complex (too many parameters vs available information), then the experiment will only at best yield a 'speckle pattern'



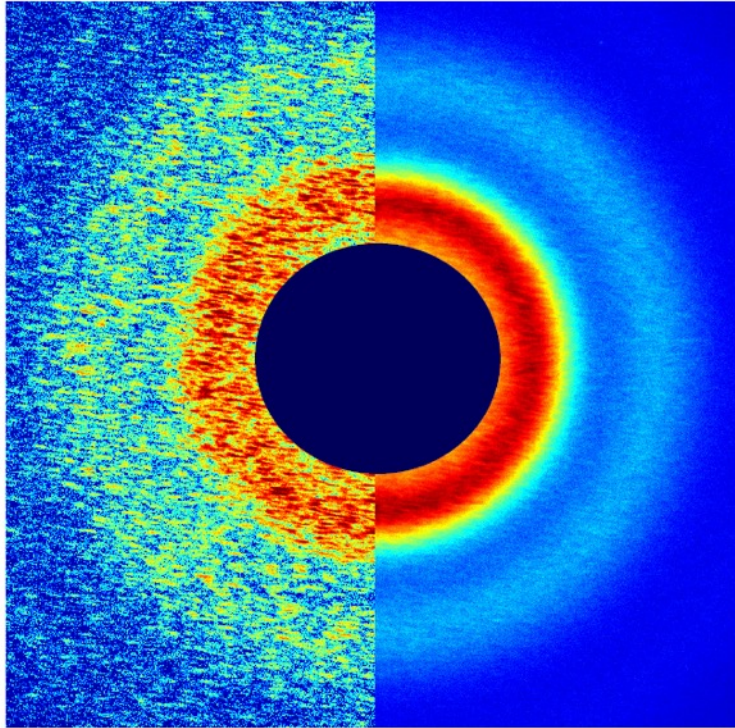
From Oleg Shpyrko web page

Statistical information :

- Width of speckle pattern ↔ average particle size
- Number of speckles peaks along one
 - dimension ↔ number of particles

Time-resolved experiments : XPCS

X-RAY PHOTON CORRELATION SPECTROSCOPY (XPCS)



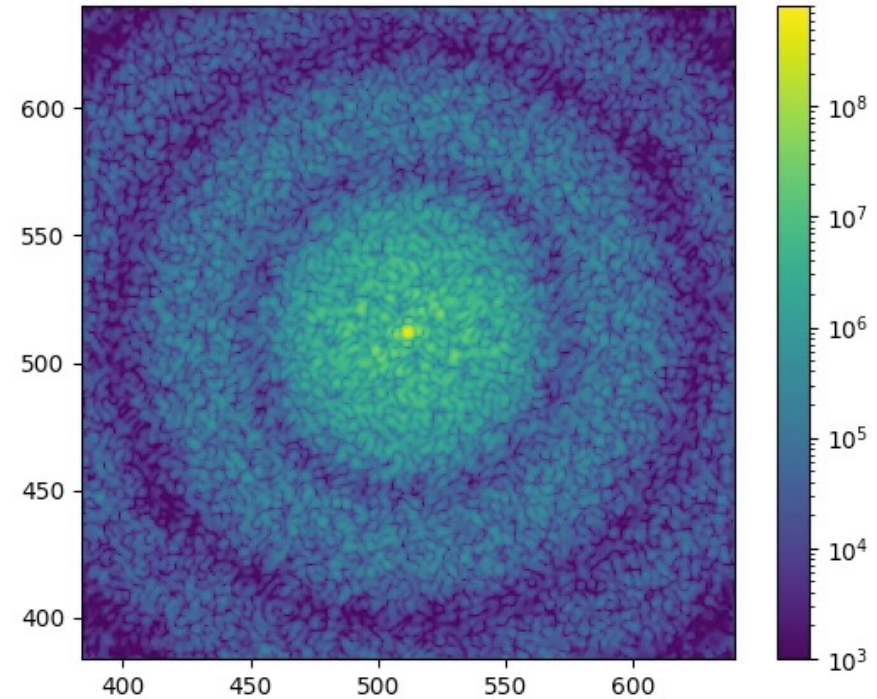
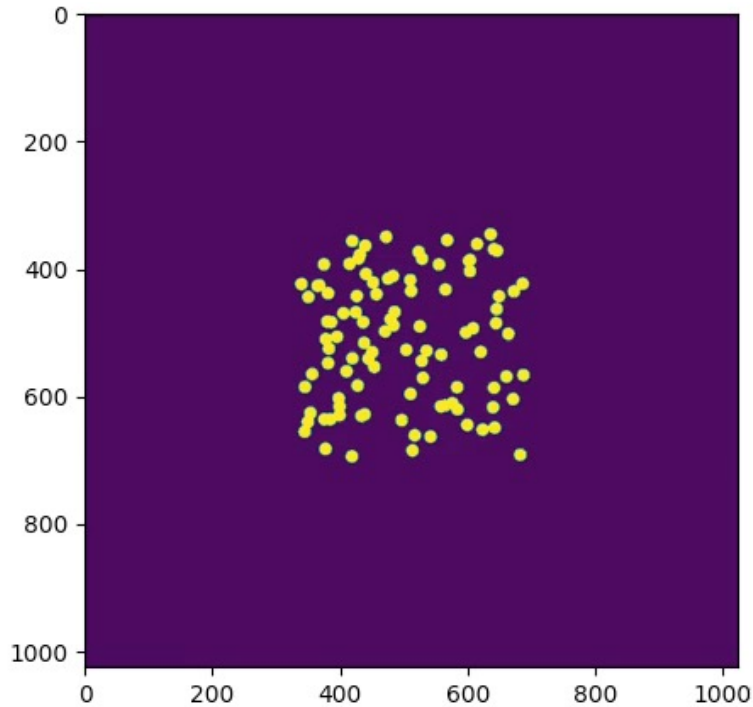
High coherence Low coherence

C. Gutt

XPCS:

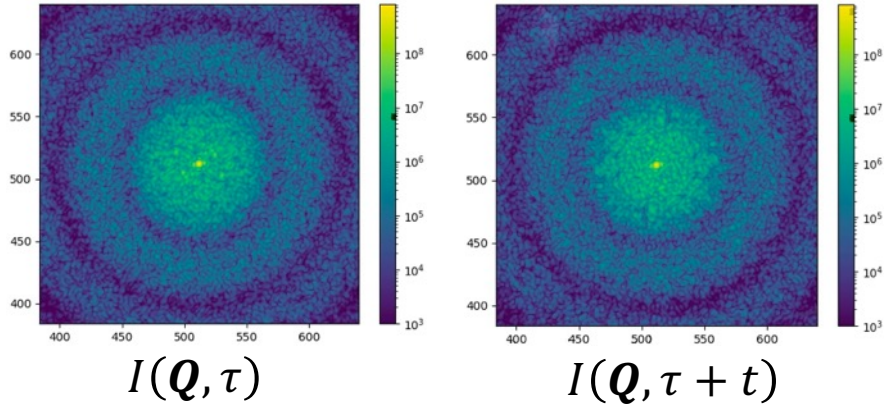
- Measure the correlation of the scattered intensity as a function of time
- *Not* a spectroscopy technique

X-RAY PHOTON CORRELATION SPECTROSCOPY (XPCS)



- Movement of particles (molecules, atoms..) induce changes in the speckles
- Higher angular momentum (far from the centre) are more sensitive to smaller displacements
- Movement (diffusion) can be quantified by correlation

X-RAY PHOTON CORRELATION SPECTROSCOPY (XPCS)

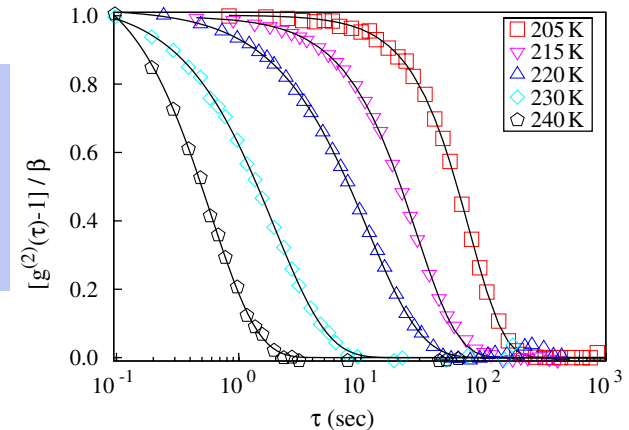


Example 1: Dynamics of silica colloidal nano-particles in super-cooled propanediol

- $T_f = 245\text{K}$ for propanediol
- $T < T_f \Rightarrow$ supercooled liquid, towards a glass transition

Temporal intensity auto-correlation function:

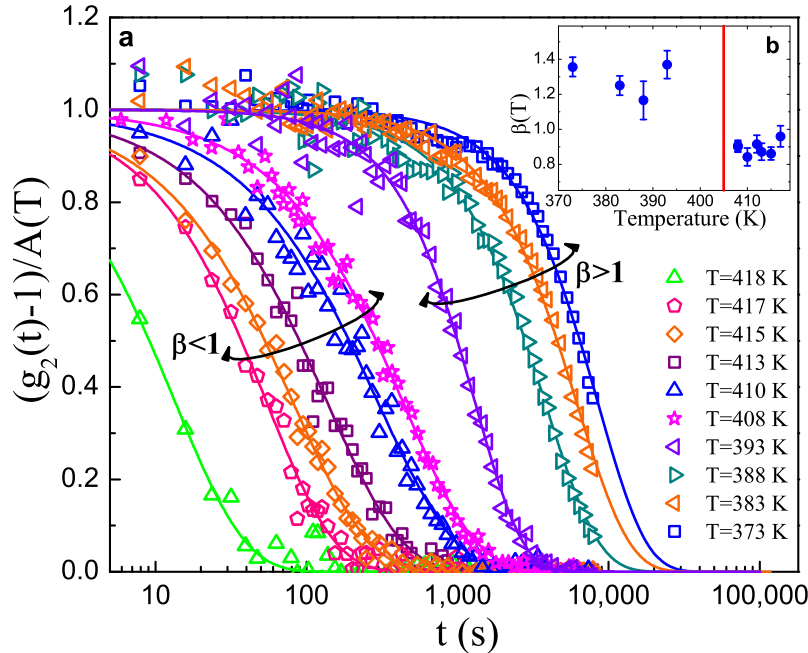
$$g^{(2)}(\mathbf{Q}, t) = \frac{\langle I(\mathbf{Q}, \tau) I(\mathbf{Q}, \tau + t) \rangle}{\langle I(\mathbf{Q}, \tau) \rangle^2}$$



PRL 100, 055702 (2008)

X-RAY PHOTON CORRELATION SPECTROSCOPY (XPCS)

Example 2: atomic-scale dynamics & in a metallic glass ($\text{Mg}_{65}\text{Cu}_{25}\text{Y}_{10}$)

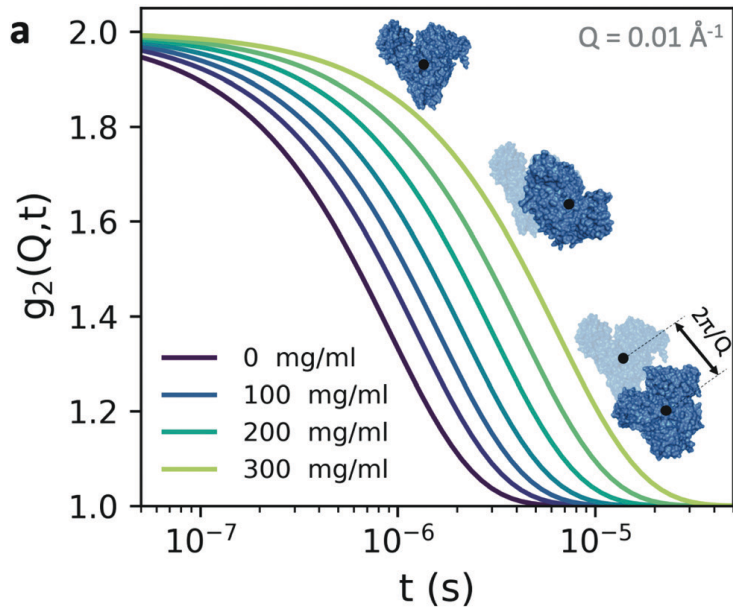


Phys. Rev. Lett. 109, 165701 (2012)

- Glasses are seen as ‘frozen liquids’
- ... they are *still* liquids, but with a very large viscosity
- $T_g = 405\text{K}$
- There is still atomic motion below T_g
- The β value changes between above and below T_g – different motion regimes

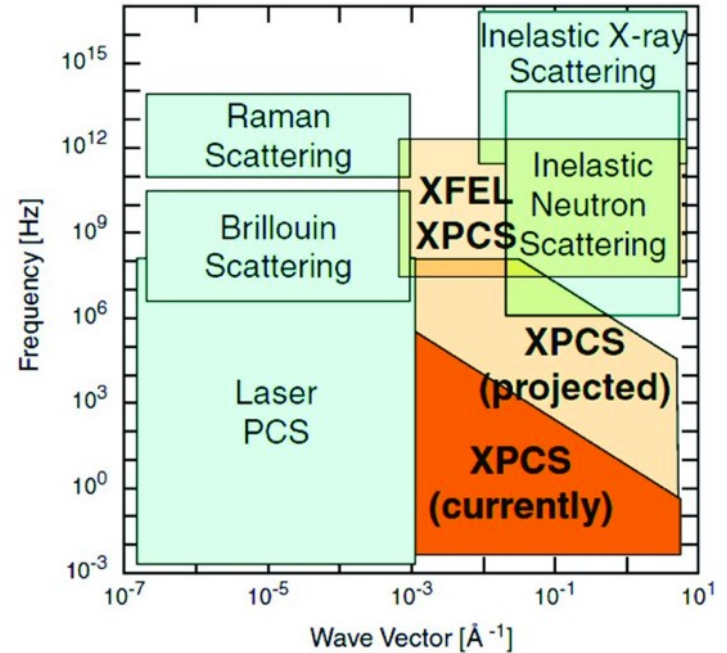
$$g^{(2)}(\mathbf{Q}, t) = 1 + Ae^{-2\left(\frac{t}{\tau}\right)^\beta}$$

X-RAY PHOTON CORRELATION SPECTROSCOPY (XPCS)



XPCS also applies to protein solutions

Phys. Chem. Chem. Phys. **22**, 19443 (2020)



Dynamics can be probed on a wide range of time-scales, which increases with modern synchrotron sources and XFEL

Pawel Kwasniewski PhD (Grenoble)

COHERENT X-RAYS: DYNAMICS & IMAGING



Dynamics X-ray Photon Correlation Spectroscopy

The diffraction pattern is the sum of the interference of the scattering from all particles

$$A(\vec{s}) = \sum \rho_i e^{2i\pi\vec{s}\cdot\vec{r}_i}$$

High-resolution imaging

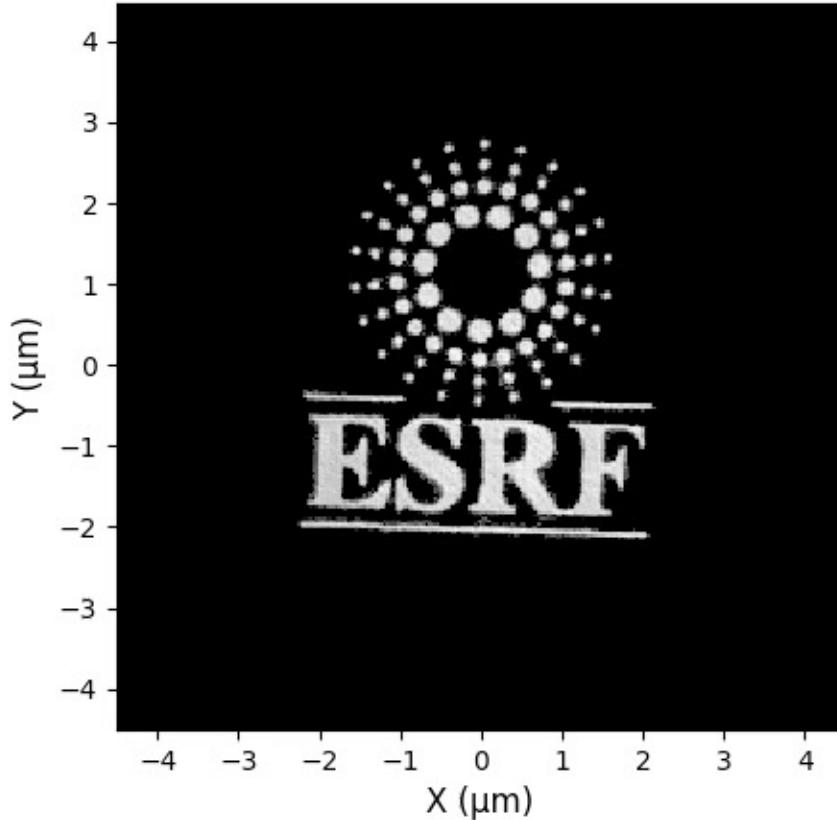
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- The **resolution** is inversely proportional to the angular extent of the scattering (far field)

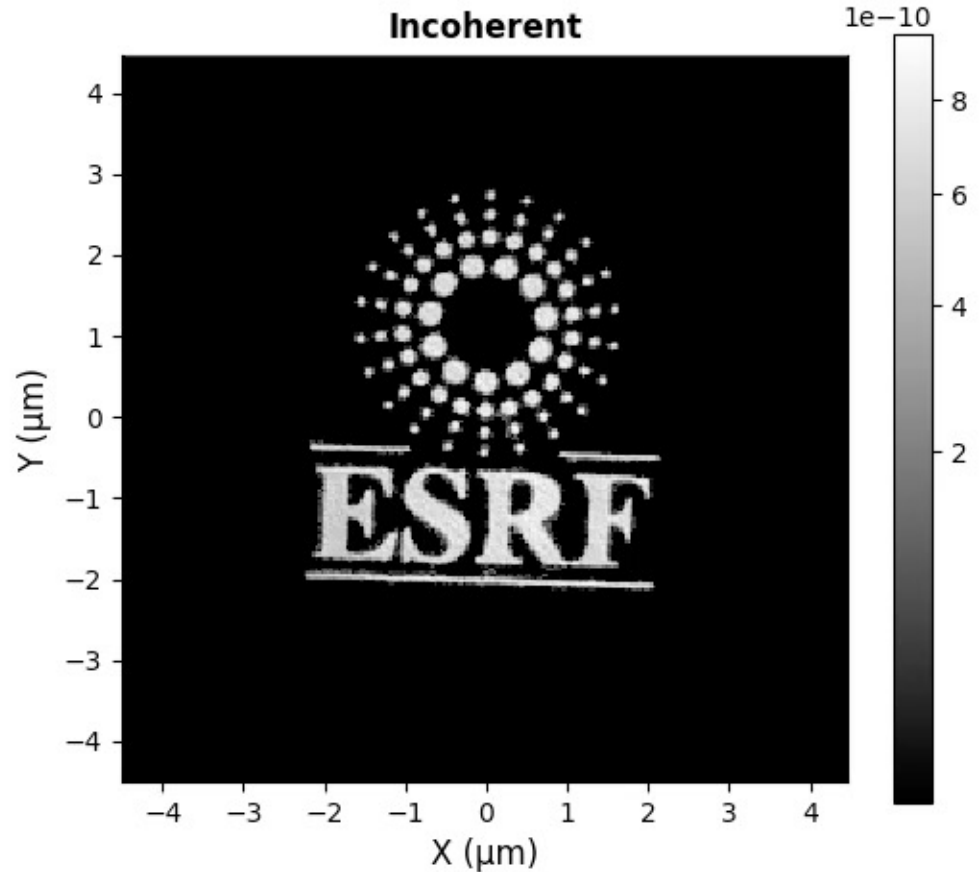
COHERENT VS INCOHERENT IMAGING

ESRF logo propagation (amplitude), $z = 100.0$ nm

Coherent

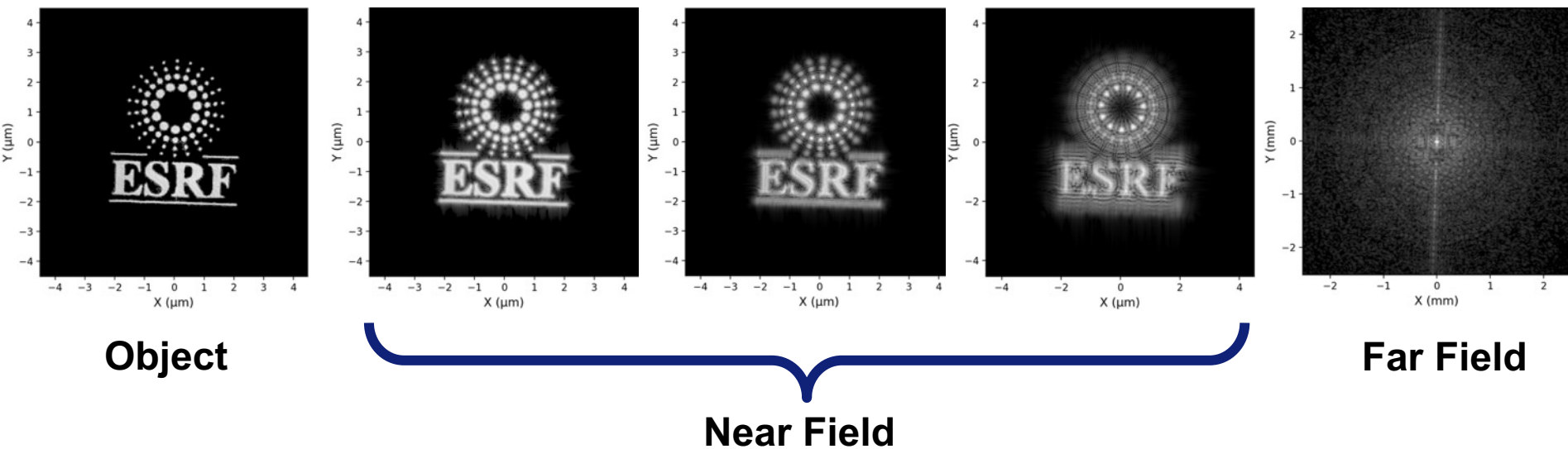


Incoherent



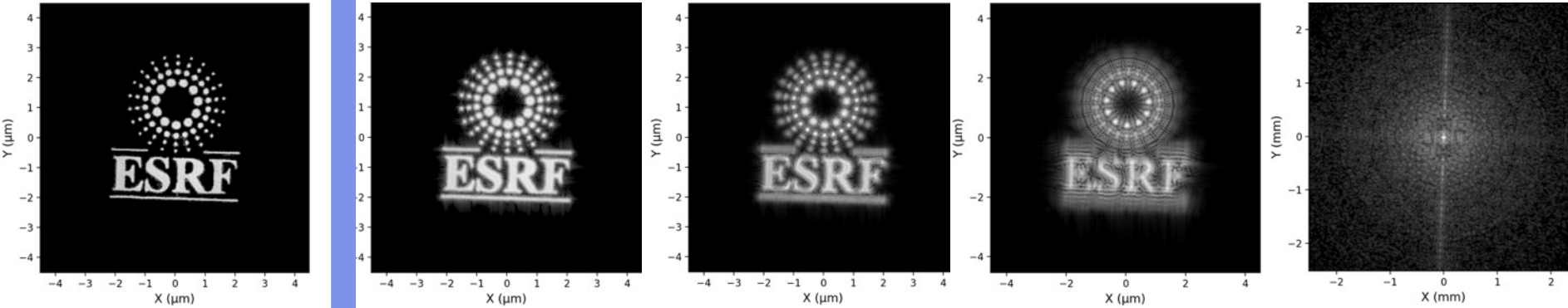
COHERENT X-RAY IMAGING TECHNIQUES

Propagation distance



COHERENT X-RAY IMAGING: ALGORITHMS ?

Propagation distance



Object

The Phase problem

- Only the intensity is measured
- Complex algorithms required to reconstruct the object
- Iterative processes are used to yield the highest resolution

AMPLITUDE & PHASE



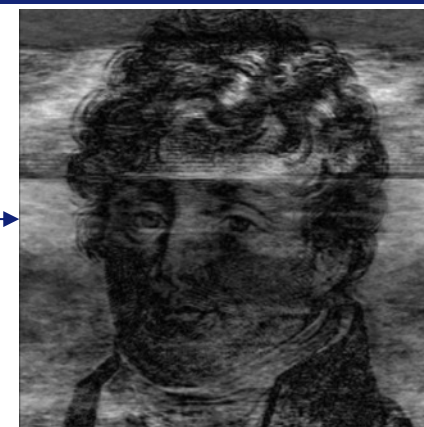
Joseph Fourier

↓ FT



Ada Lovelace

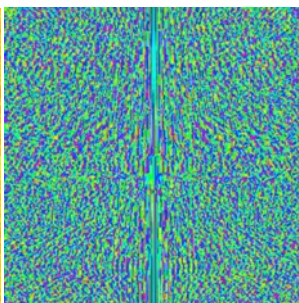
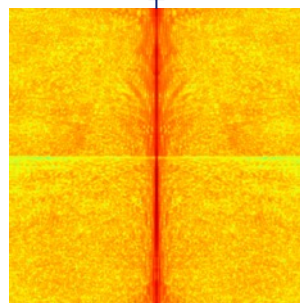
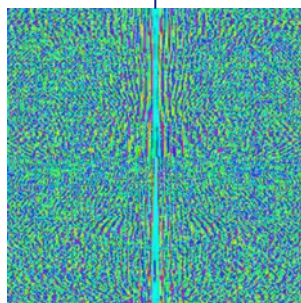
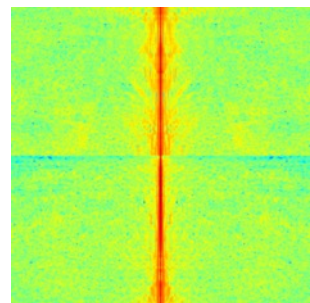
↓ FT



Fourier Phase
+ Lovelace amplitude



Lovelace Phase
+ Fourier Amplitude



Amplitude(log)

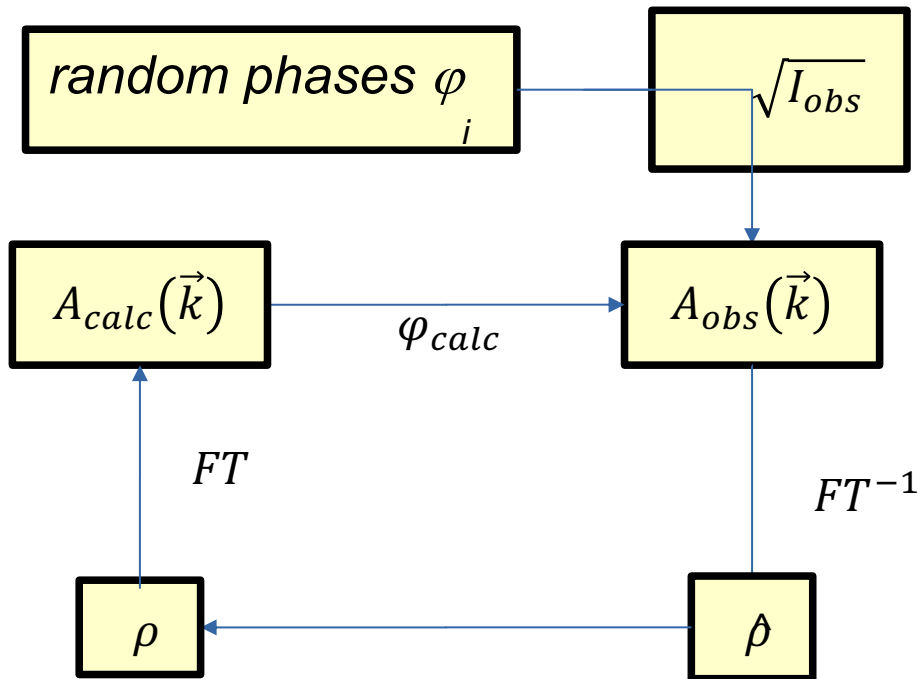
Phase

Amplitude(log)

Phase

THE PHASE PROBLEM

To compute the inverse Fourier Transform, both the phase and the amplitude are needed
→ Phase Retrieval algorithms are required



Algorithms:

Hybrid Input/Output (HIO)
Error Reduction
Charge Flipping

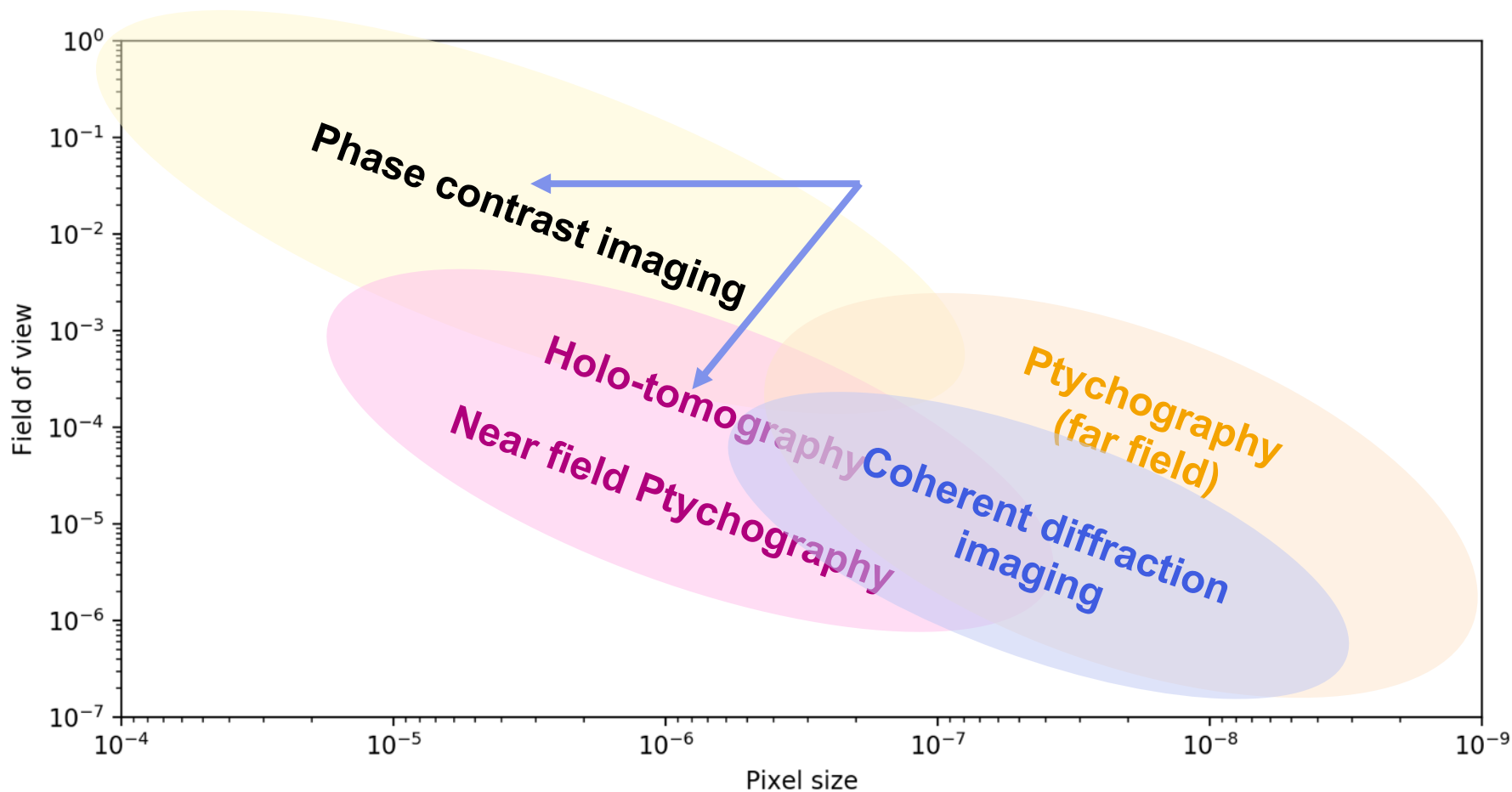
....

Density modification:

- Positivity (*)
- Finite support

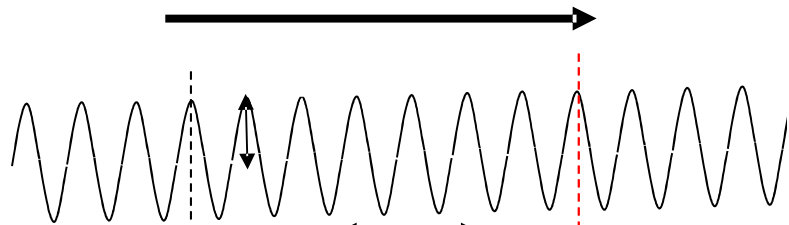
NB : the **'Fourier recycling'** algorithms are essentially the same as those used to obtain electronic density from anomalous (or isomorphous) diffraction data for macromolecular crystallography.

IMAGING: FIELD-OF VIEW VS RESOLUTION

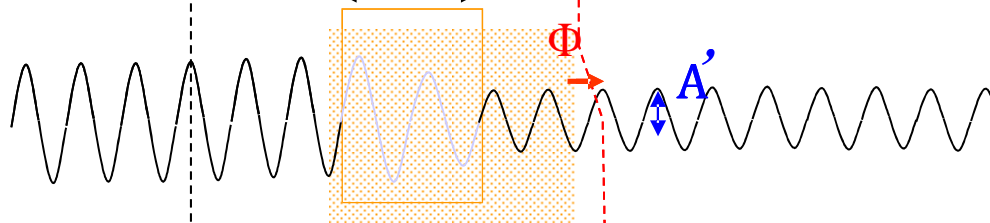


ATTENUATION VS PHASE CONTRAST

Photons/neutrons



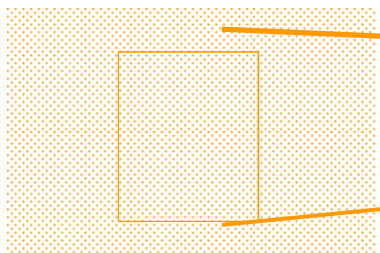
$$\Psi_{before} = Ae^{i\omega t - kx}$$



$$\Psi_{after} = \underbrace{Ae^{-2i\pi\frac{\delta}{\lambda}s}}_{\Phi} \underbrace{e^{-2\pi\frac{\beta}{\lambda}s}}_{A'/A} \Psi_{before}$$

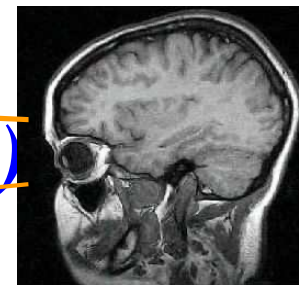
refractive index
 $n = 1 - \delta + i\beta$

real life:

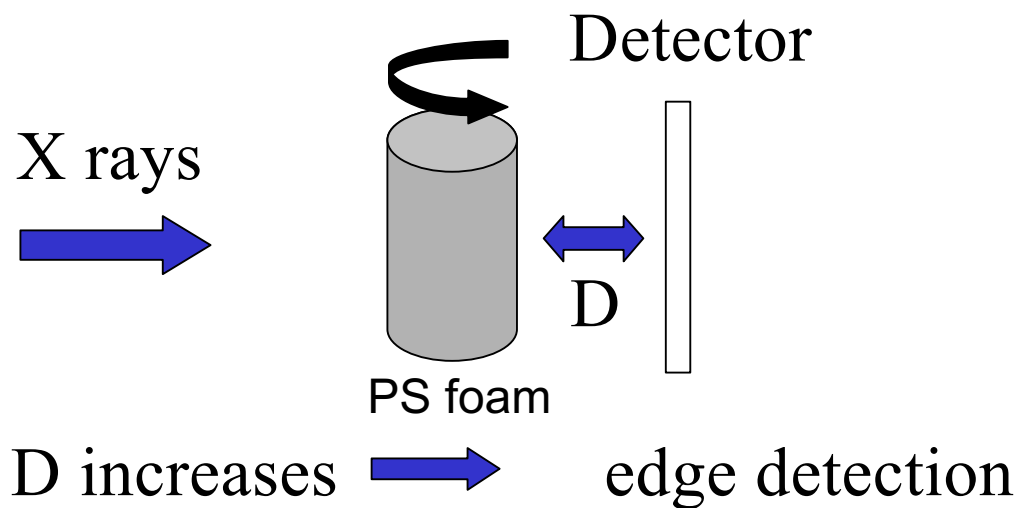


direct observation

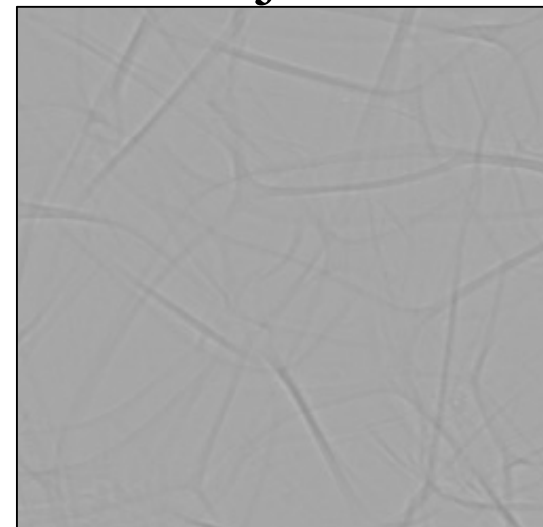
(A')



PHASE CONTRAST IMAGING (NEAR FIELD / FRESNEL REGIME)



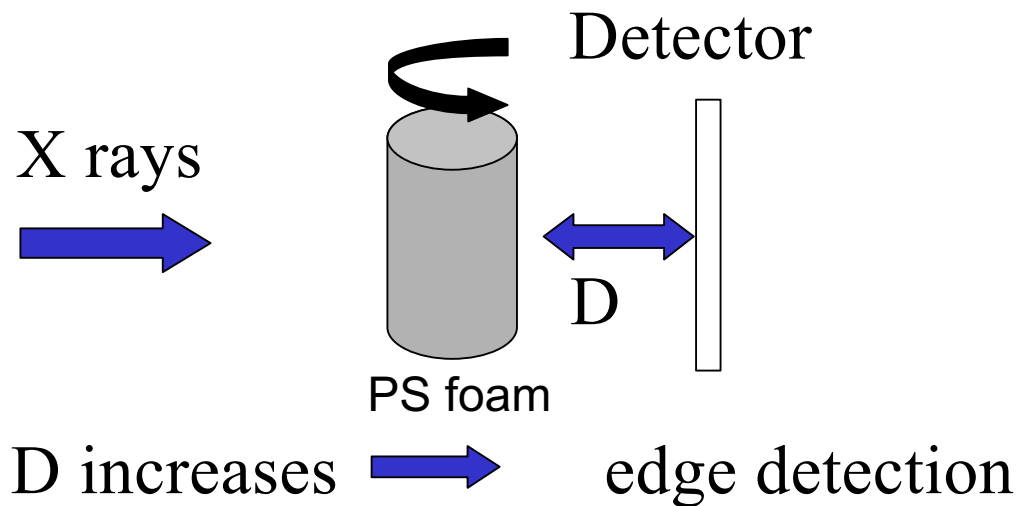
Projection



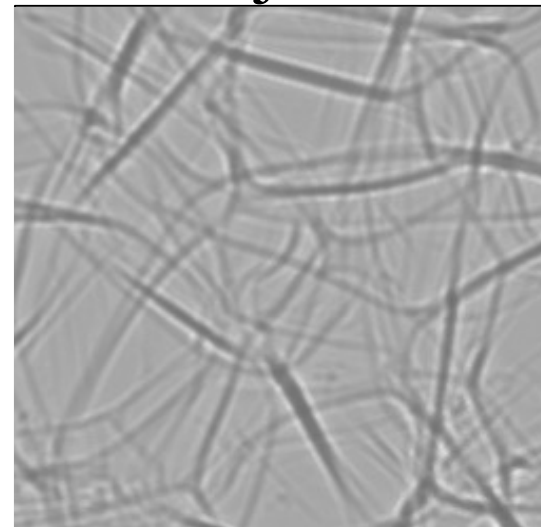
50 μ m

Only absorption, faint contrast

PHASE CONTRAST IMAGING (NEAR FIELD / FRESNEL REGIME)



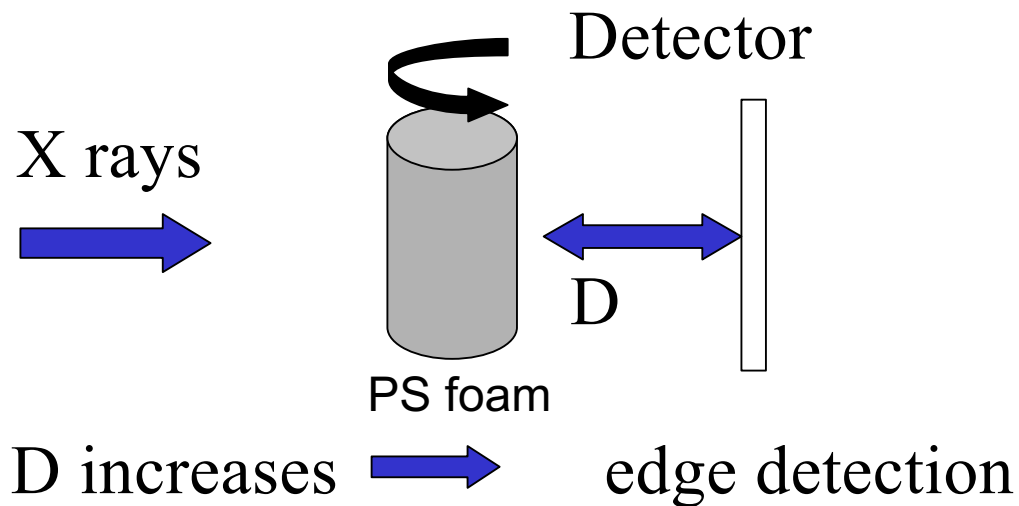
Projection



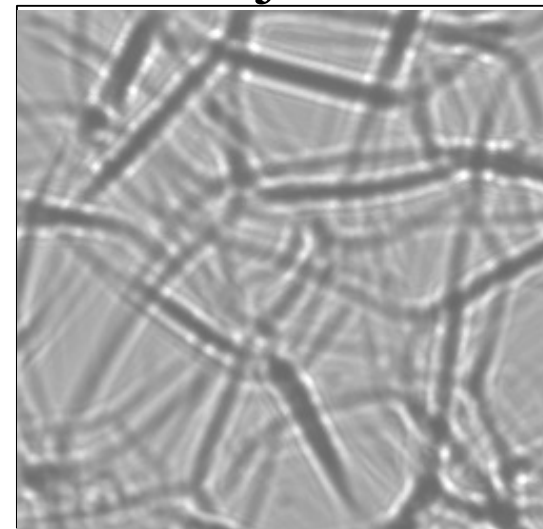
50 μm

Better contrast

PHASE CONTRAST IMAGING (NEAR FIELD / FRESNEL REGIME)



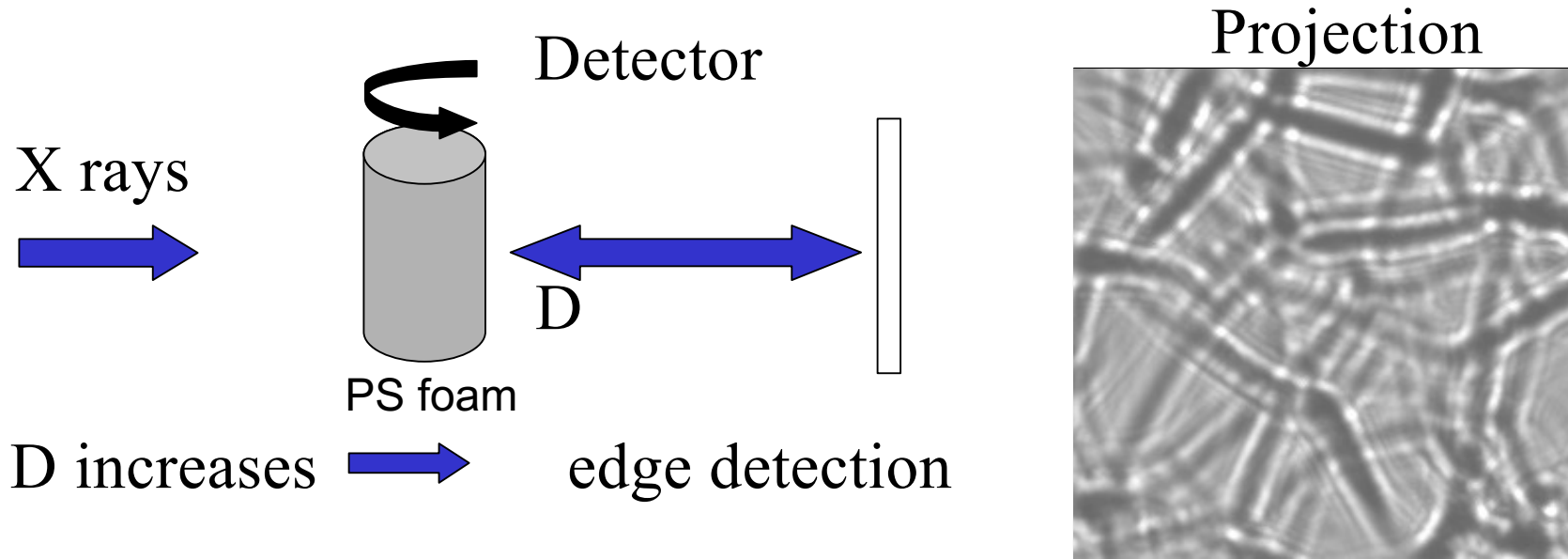
Projection



50 μm

Stronger contrast

PHASE CONTRAST IMAGING (NEAR FIELD / FRESNEL REGIME)

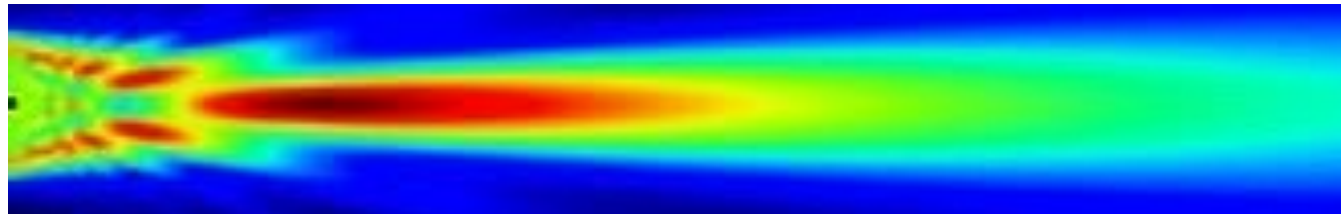


D increases



edge detection

Most contrast from phase 50 μm

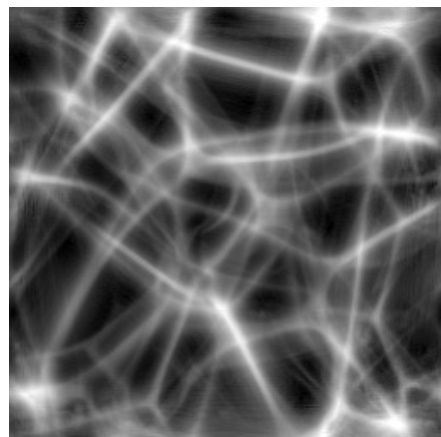
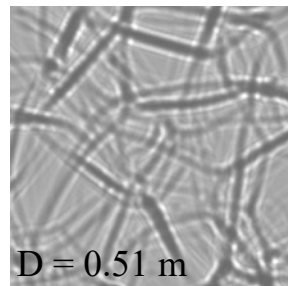
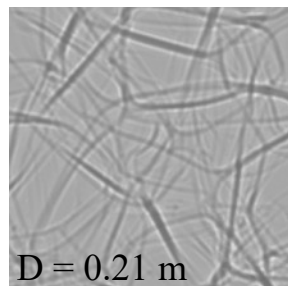
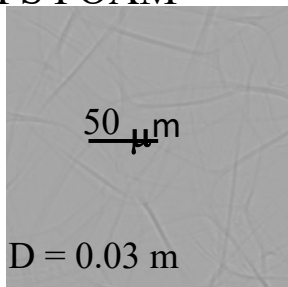


Slide from JY Buffiere



PHASE CONTRAST IMAGING: HOLOTOMOGRAPHY

PS FOAM



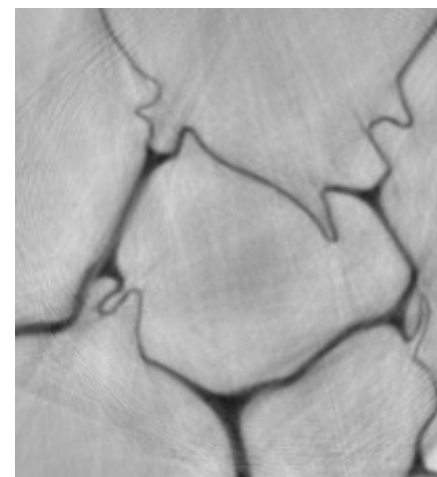
Phase retrieval

$$\Delta\varphi = 2\pi\delta t/\lambda$$

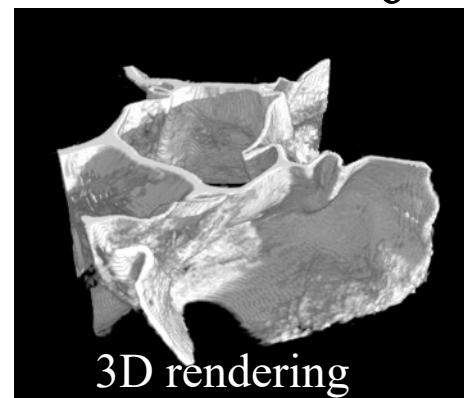
t : thickness

N projections \rightarrow

tomographic
reconstruction

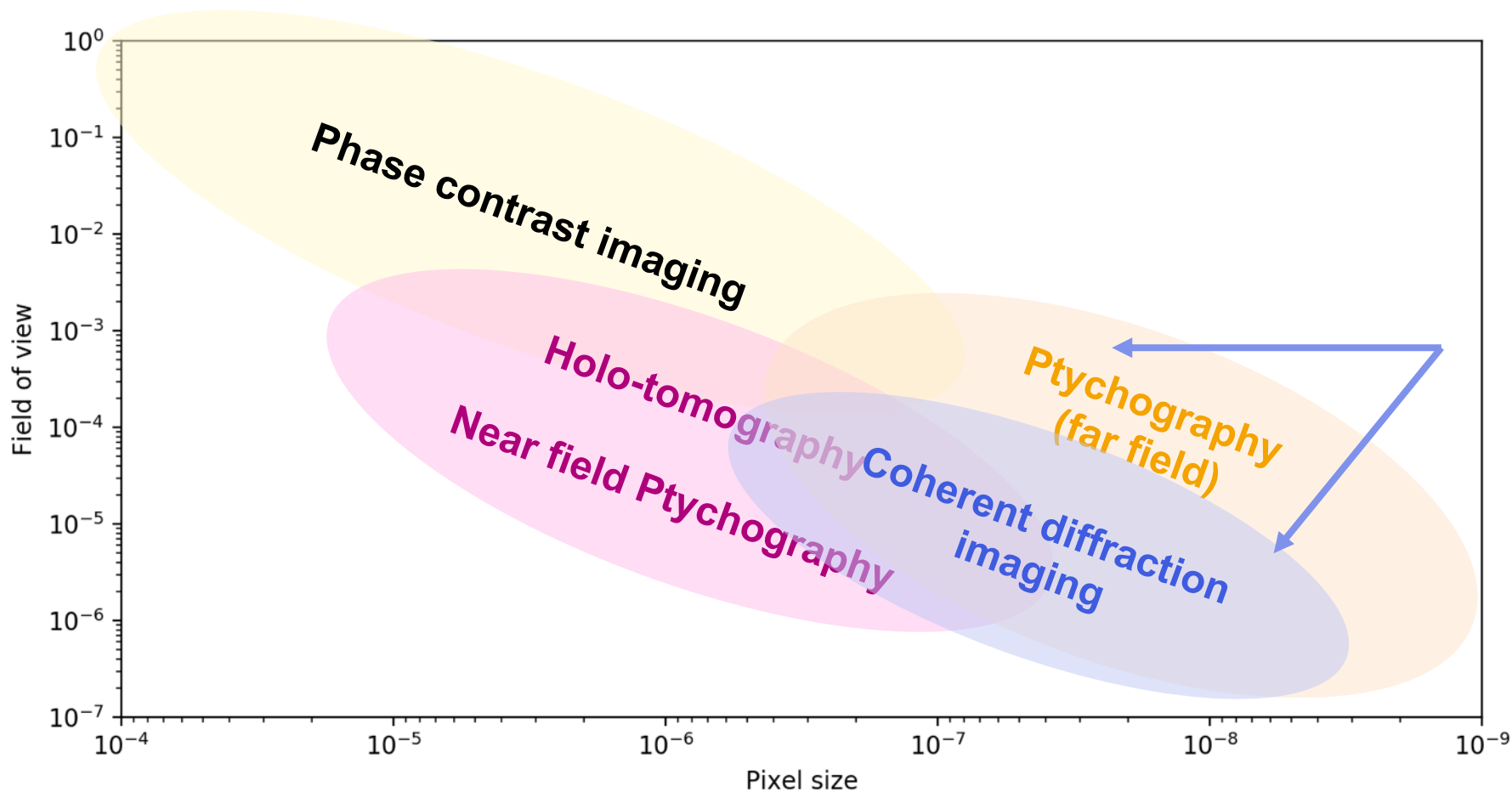


Reconstruction of δ (r)

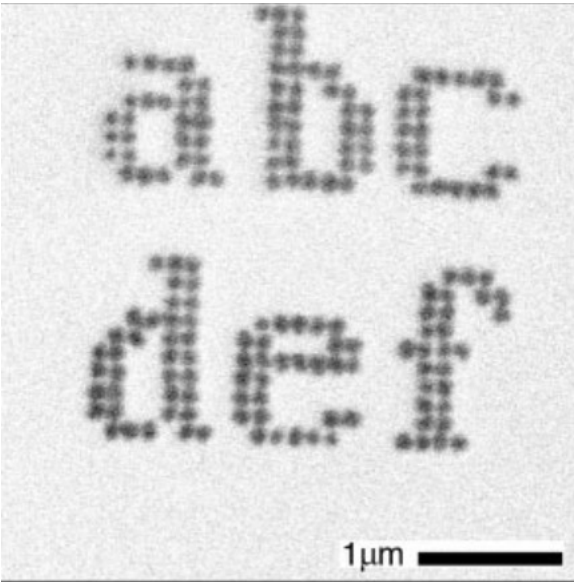


3D rendering

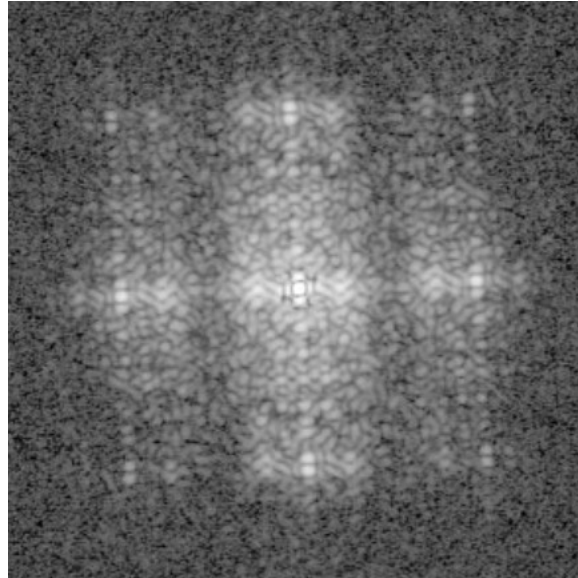
IMAGING: FIELD-OF VIEW VS RESOLUTION



COHERENT DIFFRACTION IMAGING



Gold dots



Diffraction image



Reconstruction

J. Miao, P. Charalambous, J. Kirz, and D. Sayre, Nature 400, 342 (1999).



Why use a method :

- Which is complex (experimentally : coherence)
- Where more than half the information (phase) is lost
- Where algorithms are not always robust

When there are simpler methods :

- Absorption imaging
- Phase contrast imaging
- Scanning microscopy

RESOLUTION

For scanning microscopy :

- Resolution = beam size

For phase contrast/absorption imaging :

- Resolution = pixel size
- With focusing optics (projection microscopy), down to a few 10's of nm



Why use a method :

- Which is complex (experimentally : coherence)
- Where more than half the information (phase) is lost
- Where algorithms are not always robust

When there are simpler methods :

- Absorption imaging
- Phase contrast imaging
- Scanning microscopy

For CDI

The smaller the object, the wider its Fourier Transform !

→ The resolution is inversely proportionnal to the extent of the measured scattering in reciprocal space

→ lower current limit : 5-10 nm for CDI at 8-15 keV

The resolution is smaller than the beam size → “ **super-resolution** ”

→ **High resolution studies and/or nano-objects**

MORPHOLOGY OF COCCOLITHS USING CDI

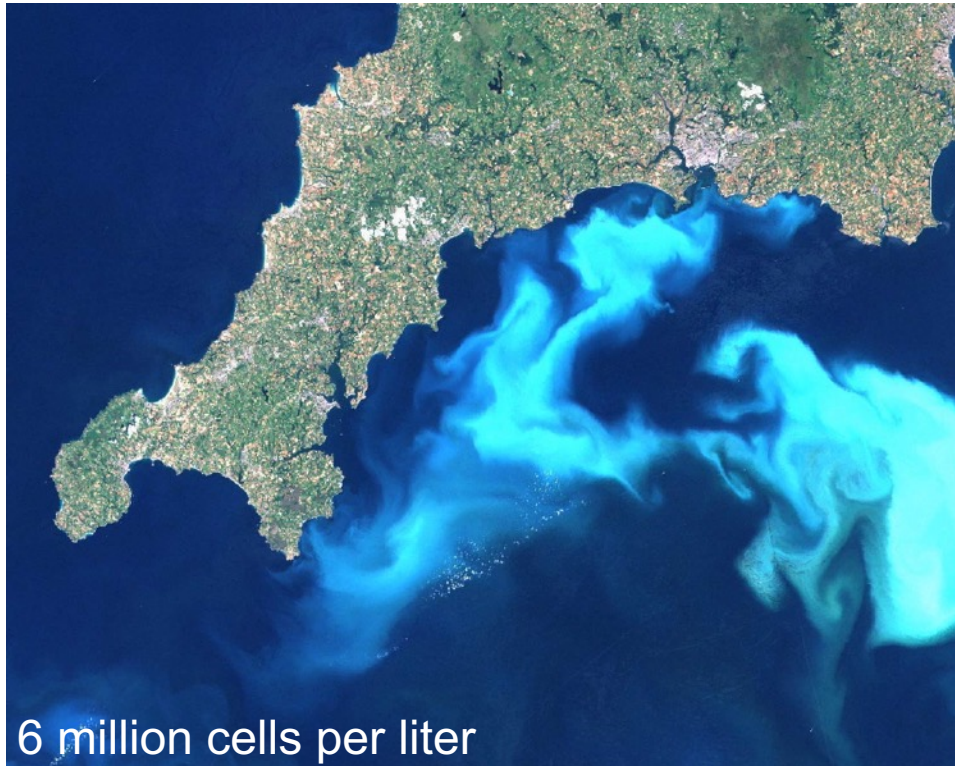
T. Beuvier^{a,b}, I. Probert^c, L. Beaufort^d, B. Suchéras-Marx^d, Y. Chushkin^b, F. Zontone^b and A. Gibaud^a

^a LUNAM, IMMM, UMR 6283 CNRS, Faculté des Sciences 72085 Le MANS Cedex 09, France,

^b European Synchrotron Radiation Facility, 71, avenue des Martyrs, 38000 Grenoble, France,

^c CNRS, Sorbonne Université Pierre et Marie Curie (UPMC) Paris 06, FR2424, Roscoff Culture Collection, Station Biologique de Roscoff, Place Georges Teissier, 29680 Roscoff, France,

^d Aix Marseille Univ, CNRS, IRD, CollFrance, CEREGE, Aix-en-Provence, France.



6 million cells per liter

Emiliania Huxleyi bloom
south of Cornwall (UK)

single-celled phytoplankton
covered with calcite disks
(coccoliths)

Landsat image from 24th July 1999, courtesy of Steve Groom, Plymouth Marine Laboratories.

MARINE ALGAE - COCCOLITHOPHORES

Emiliana huxleyi

Coccosphere

Coccolith

Production is light dependent

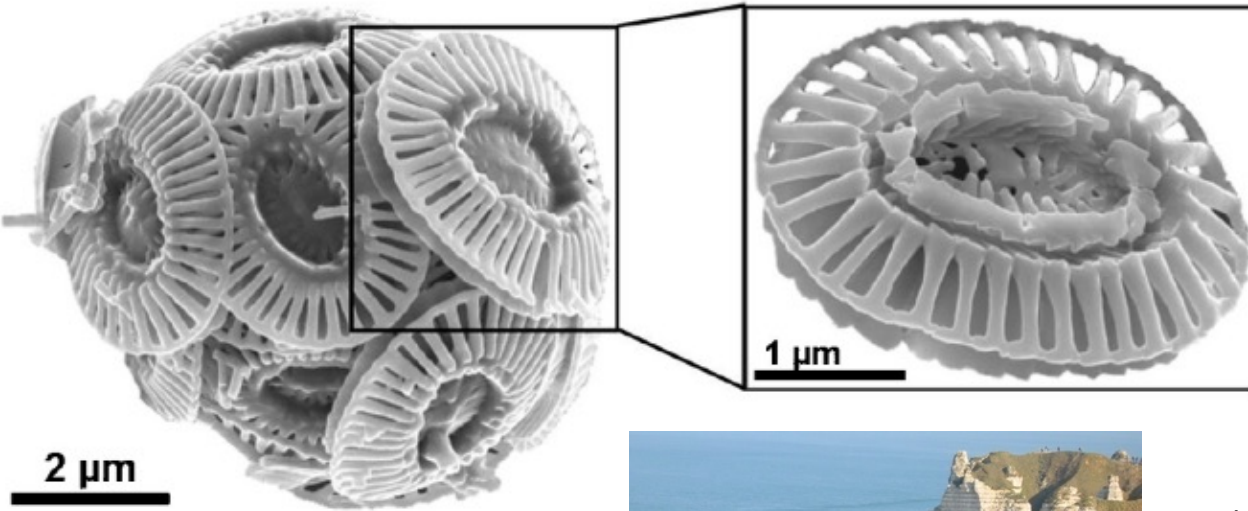
1 per 2 hours

Calcite crystals CaCO_3

Coccoliths 1-10 μm big

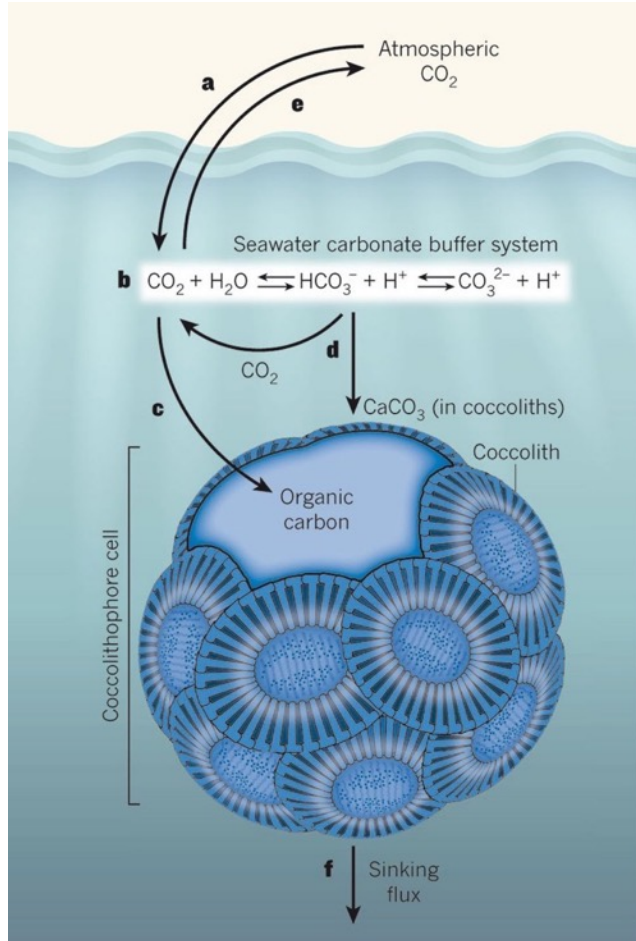
10-20 coccoliths per coccosphere

Most important calcifying organisms



Hoffmann et al (2014)

COCCOLITHOPHORE CARBON CHEMISTRY



Transfer of CO_2 from atmosphere to limestone

Last 200 years absorbed 50% of CO_2 emitted by human activities (>500 Gt CO_2)

Pre-industrial atmospheric CO_2 280 ppm

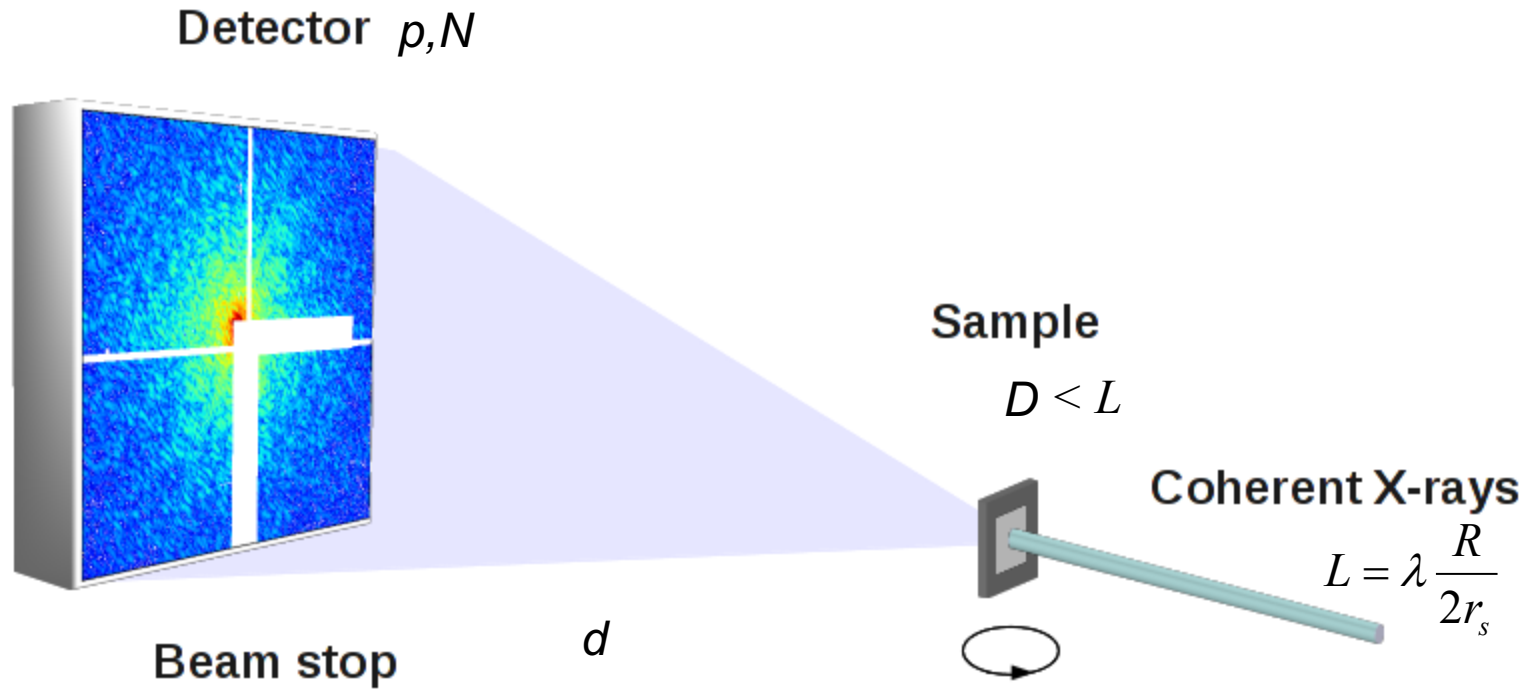
Today atmospheric CO_2 380 ppm

pH decreased of 0.1 units since pre-industrial times
pH of sea water today 8.2 ± 0.3

Calculating mass fluxes is major endeavour

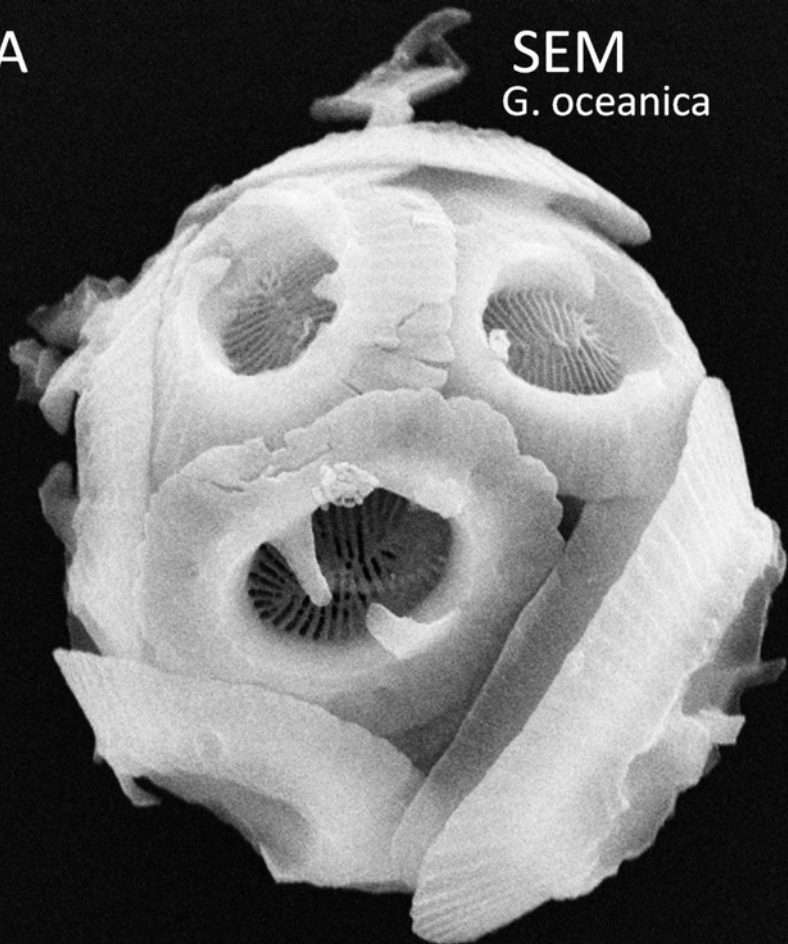
CXDI – ID10 BEAMLINE

Isolated sample $< 7 \mu\text{m}$ is illuminated with coherent plane wave
Voxel size $< 32\text{nm}$



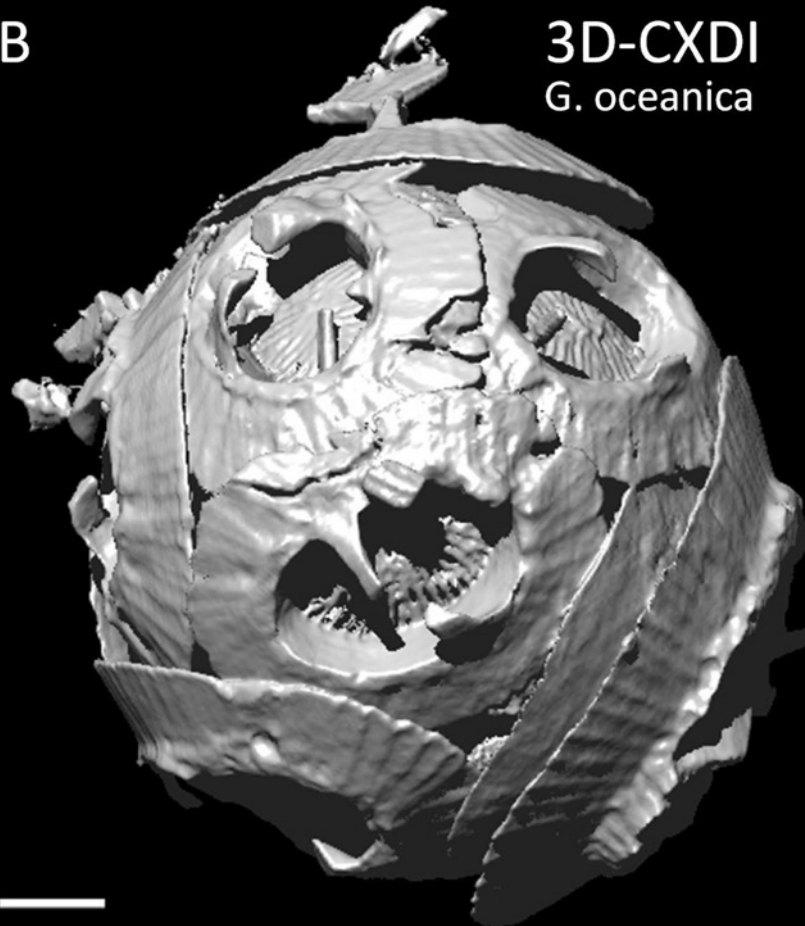
A

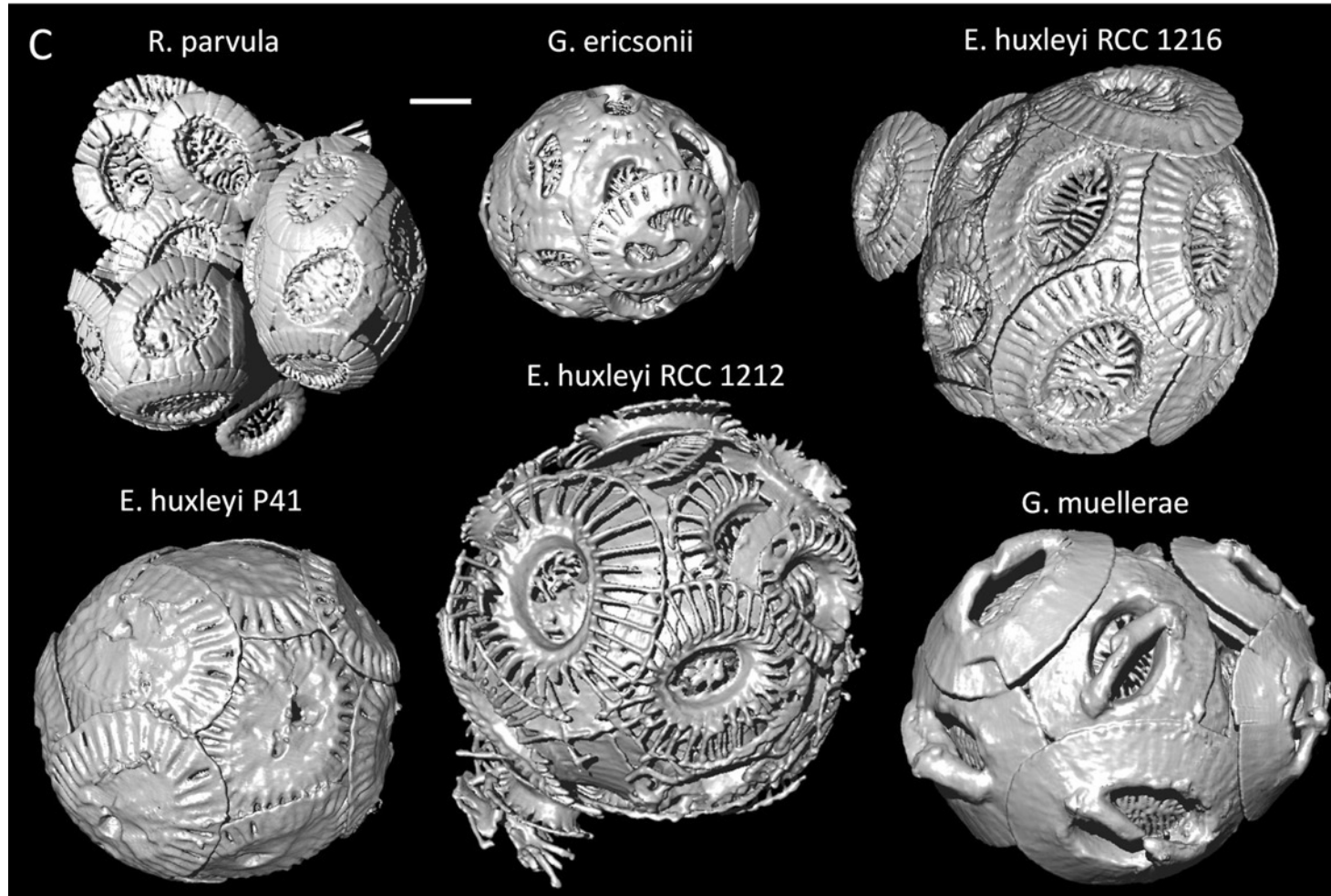
SEM
G. oceanica



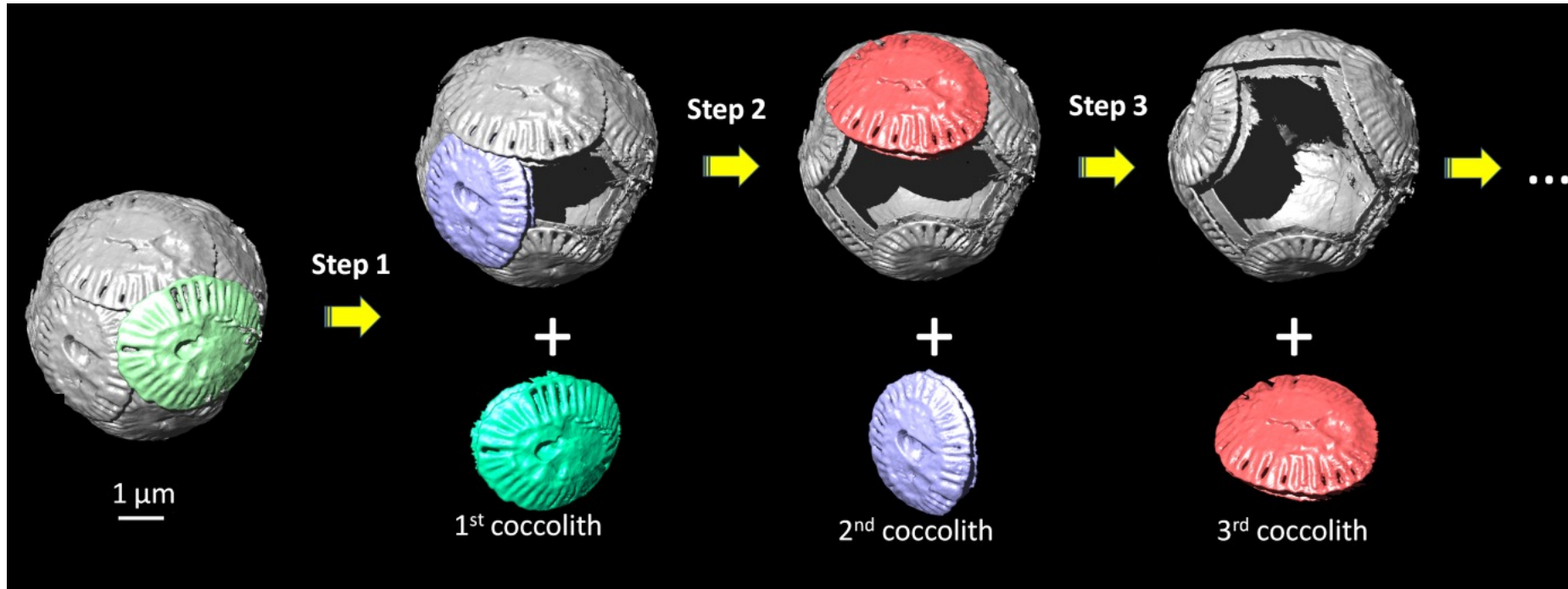
B

3D-CXDI
G. oceanica

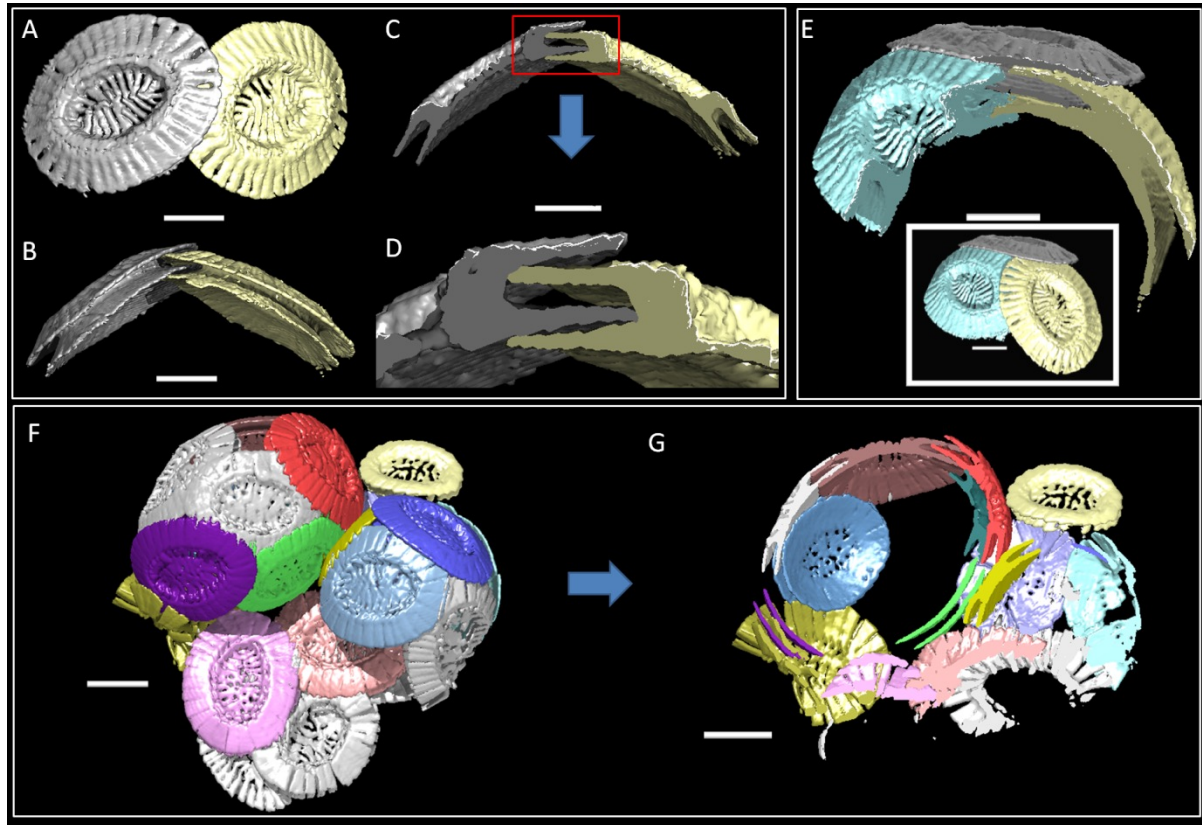




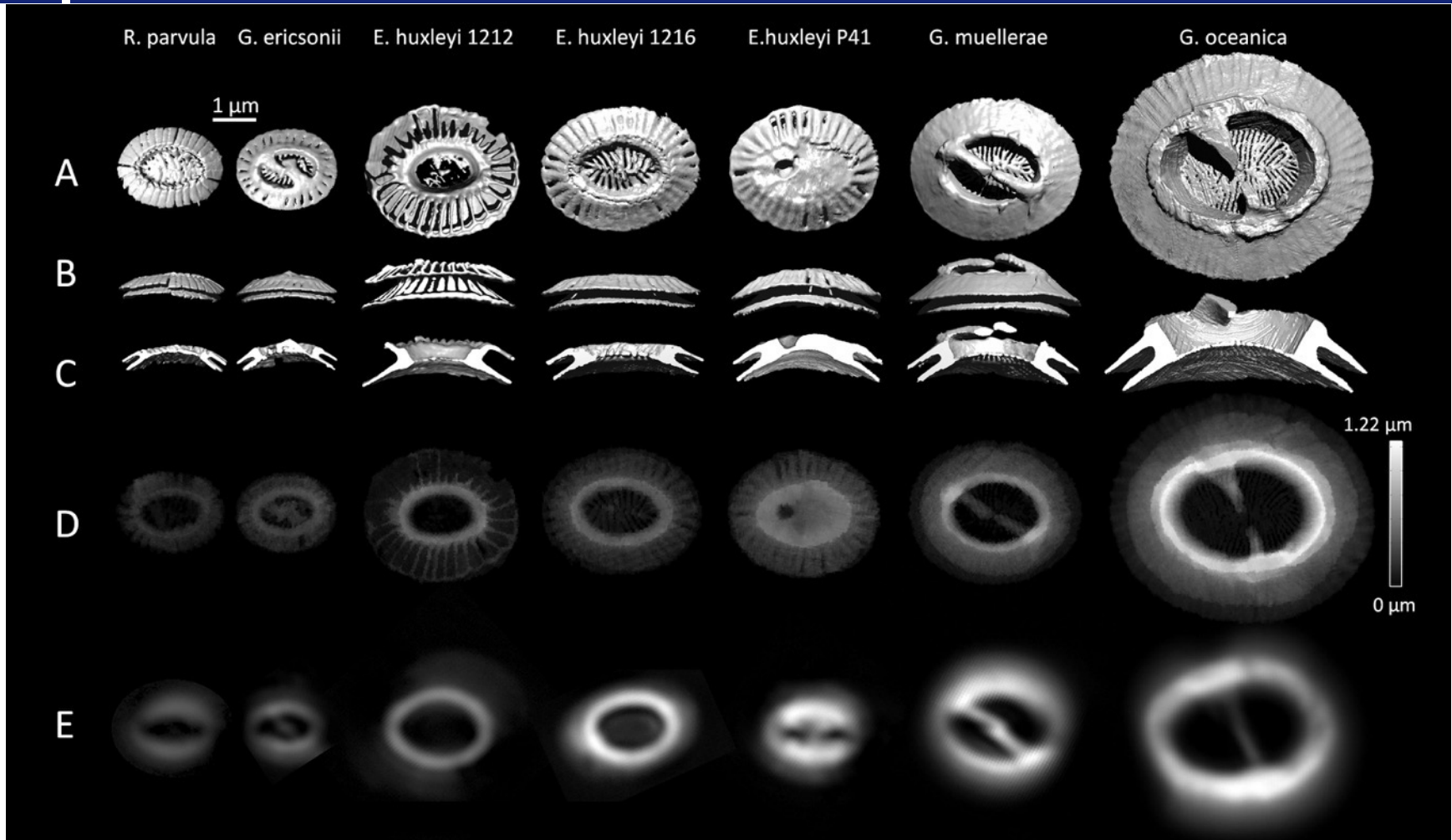
E. Huxleyi RCC1216



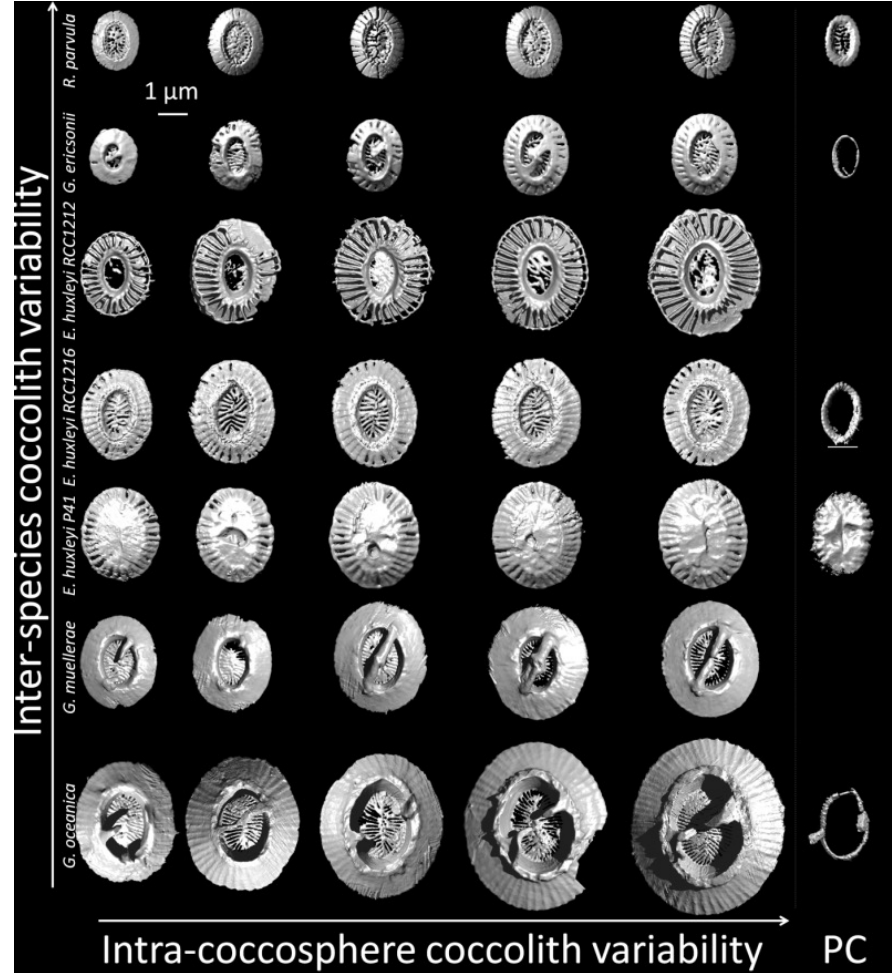
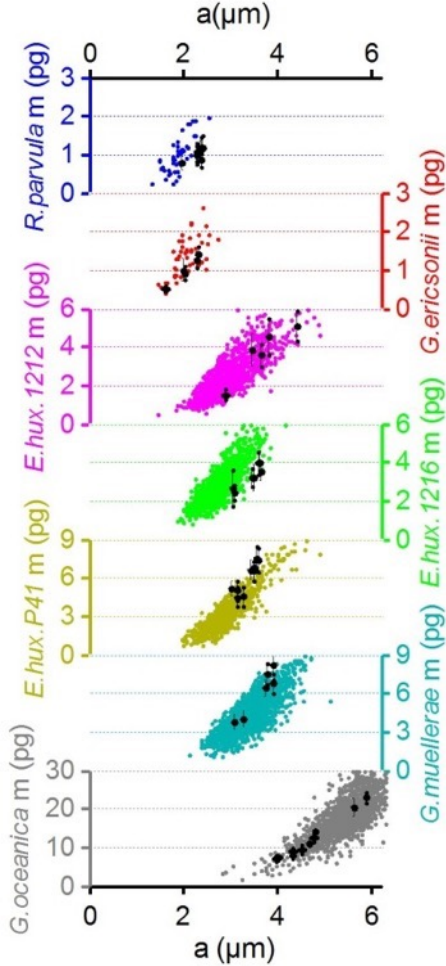
E. Huxleyi RCC1216



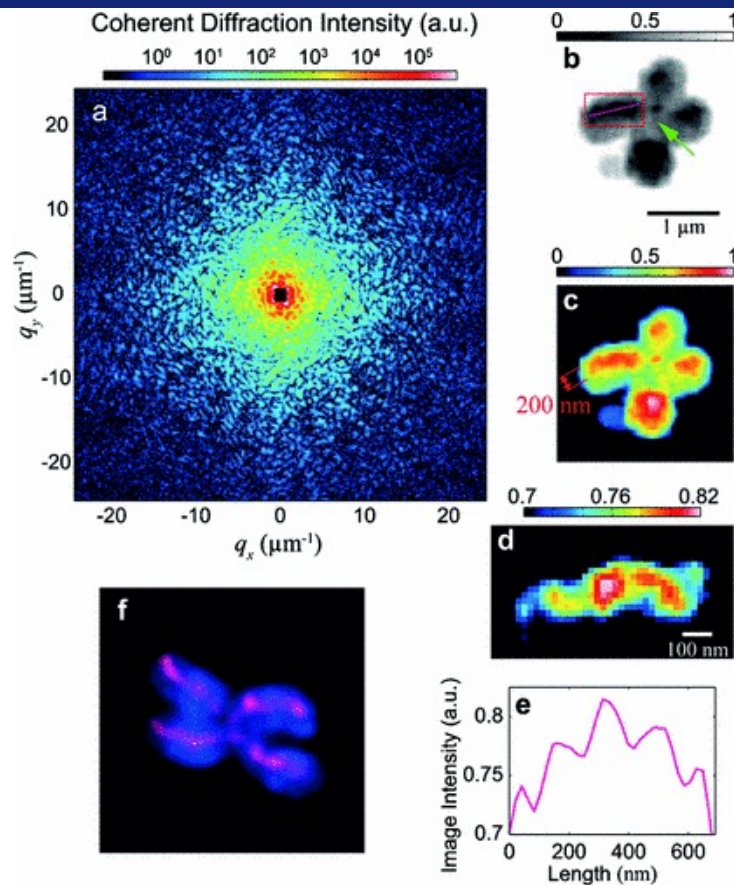
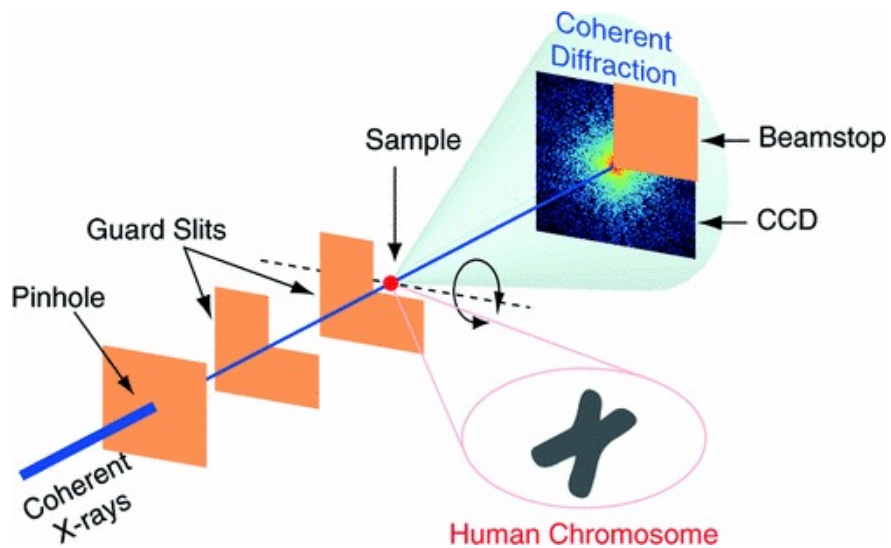
COCCOLITHS



COCCOLITHS - MASS

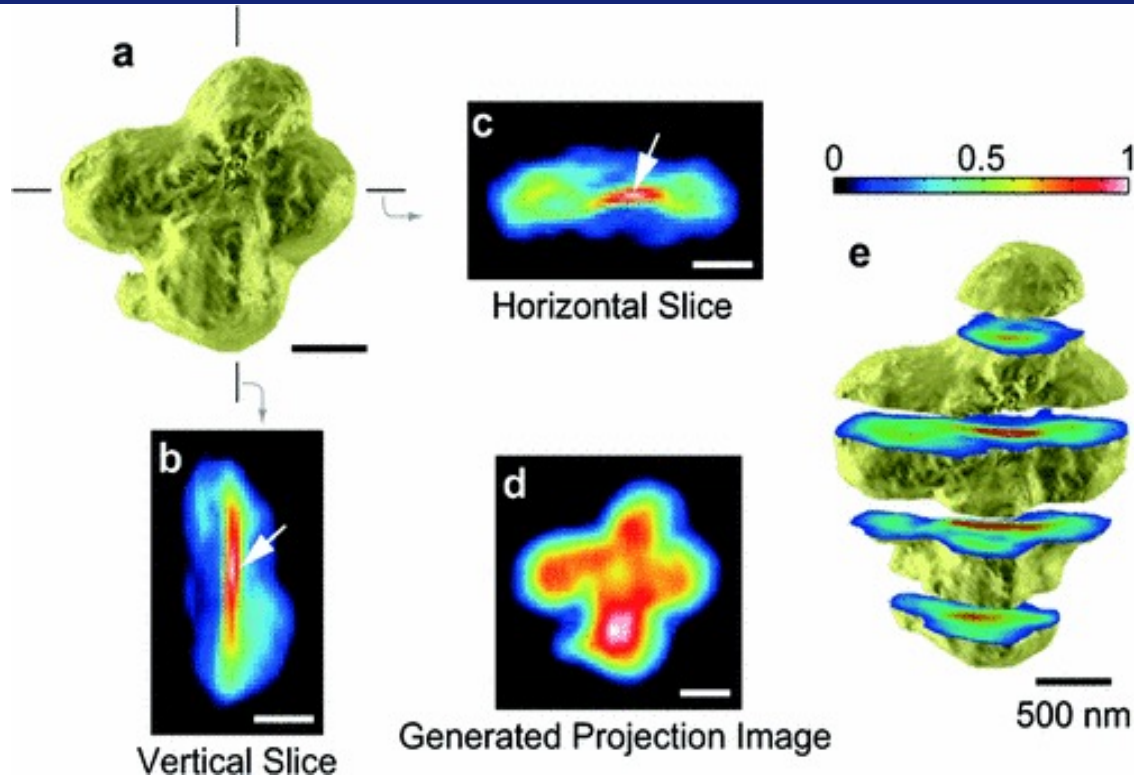


CDI: CHROMOSOME



Y. Nishino et al, Physical Review Letters 102, (2009).

CDI: CHROMOSOME



Reconstructed 3D structure :

- 2D resolution : 38 nm
- 3D resolution : 120 nm
- Dose : 2×10^{10} Gy

Y. Nishino et al, Physical Review Letters 102, (2009).

Limitations of CDI :

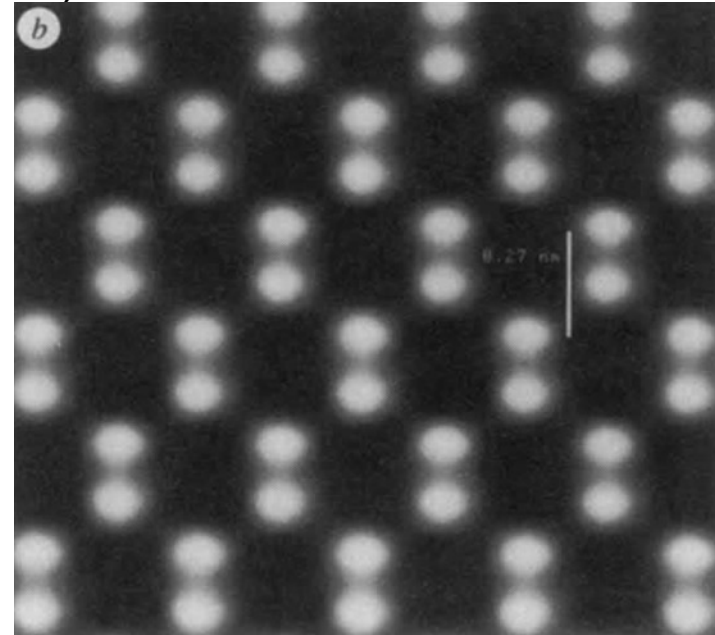
- No imaging of extended objects (limited to the coherent volume of the beam)
- Need oversampled diffraction data
- Convergence of the algorithm can be difficult (support evaluation)
- The probe (amplitude and phase of the incoming beam) is unknown
- Non-unicity of the solution

First experiments from electron microscopy :

Hoppe, Acta Cryst. A25 (1969) 459
Hoppe, Ultramicroscopy 10, 187 (1982)

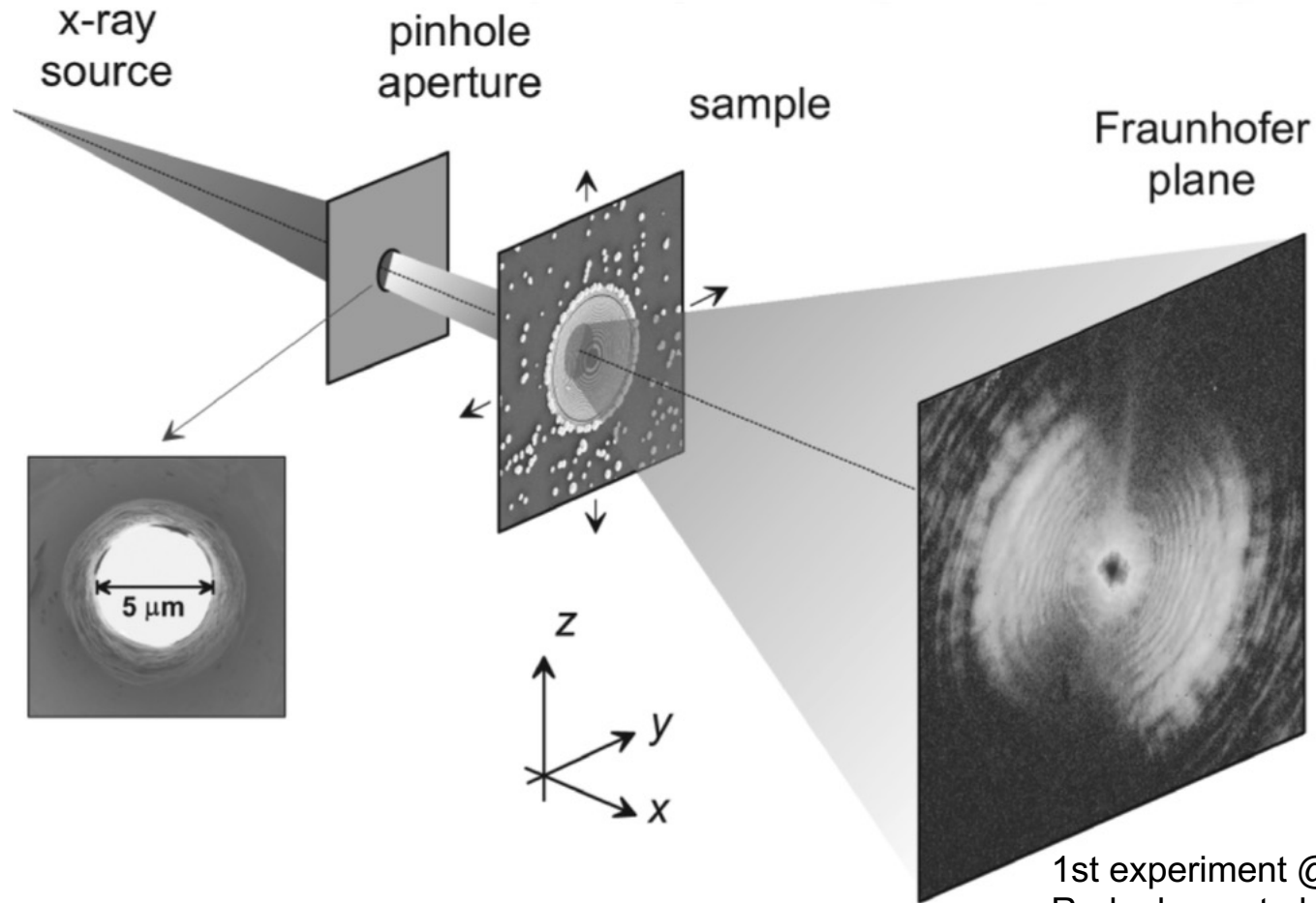
Nellist, McCallum & Rodenburg Nature A54 (1995) 49

“ Resolution beyond the 'information limit' in
transmission electron microscopy ”



Silicon reconstruction along [110]

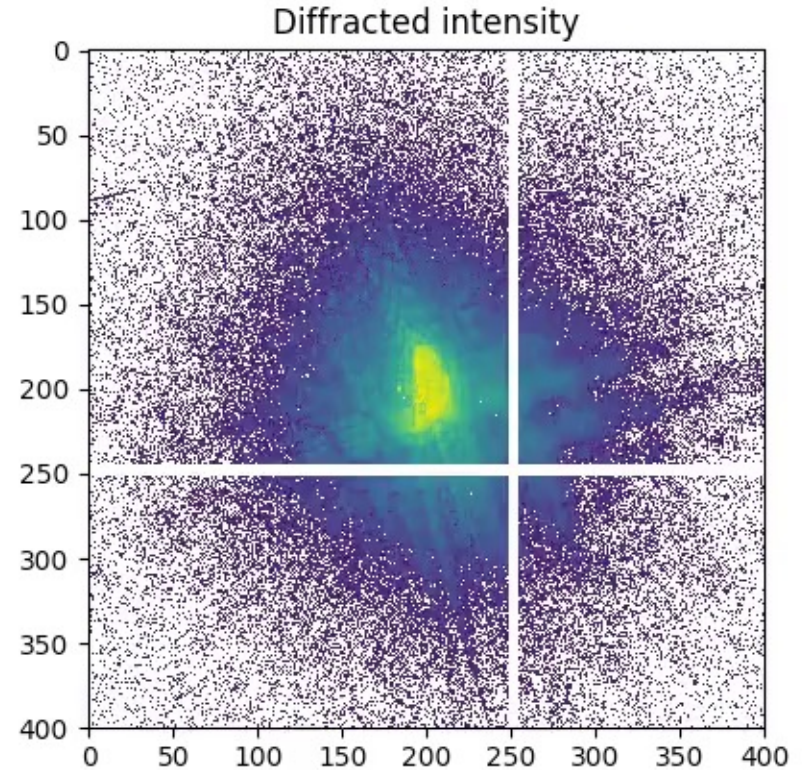
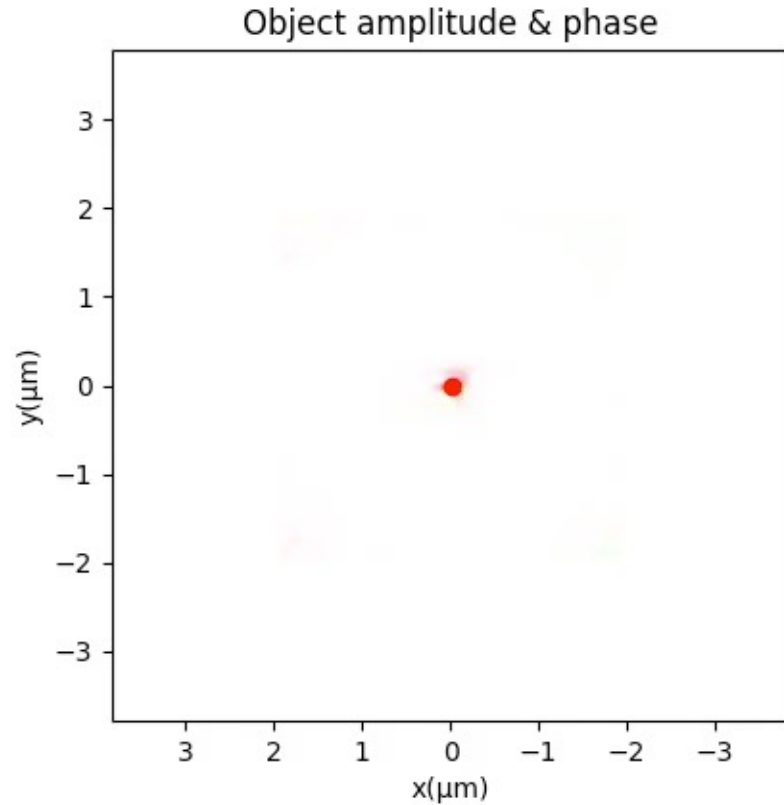
X-RAY PTYCHOGRAPHY



1st experiment @PSI
Rodenburg at al, PRL 98, 034801 (2002)

PTYCHOGRAPHY

ResultsScan0000/Run0000 - # 0



PTYCHOGRAPHY ANALYSIS WITH PYNX

jupyter-slurm.esrf.fr/user/favre/notebooks/pynx-private/pynx/ptycho/examples/position-optim-CXI-Copy1.ipynb



jupyterhub position-optim-CXI-Copy1 Last Checkpoint: 34 minutes ago (autosaved)

File Edit View Insert Cell Kernel Widgets Help

Code

Analysis is possible using <https://jupyter-slurm.esrf.fr>
Remote access to notebooks & compute cluster

Notebook: Ptychography reconstruction with positions update from a CXI dataset

This notebook uses the the runner API (`pynx.ptycho.runner`) to load the data and prepare the optimisation.

This is what is used for the command-line scripts, here we just grap the Ptycho object from the runner.

This uses a dataset recorded on id01@ESRF, which exhibits some position distortions near the center of the spiral scan.

```
In [ ]: # Select language and/or GPU name or rank through environment variable (optional)
#import os
#os.environ['PYNX_PU'] = 'cuda.0'

%matplotlib notebook
import matplotlib.pyplot as plt
from pynx.ptycho import simulation, shape

# Import Ptycho, PtychoData and operators (automatically selecting OpenCL or CUDA)
from pynx.ptycho import *

# Import CXI runner
from pynx.ptycho.runner.cxi import PtychoRunnerScanCXI
from pynx.ptycho.runner.runner import params_generic as params

# This can be used to have a wide screen for the notebook
from IPython.core.display import display, HTML
display(HTML("<style>.container { width:90% !important; }</style>"))
```

Load CXI data

dataset is available from the PyNX ESRF public folder

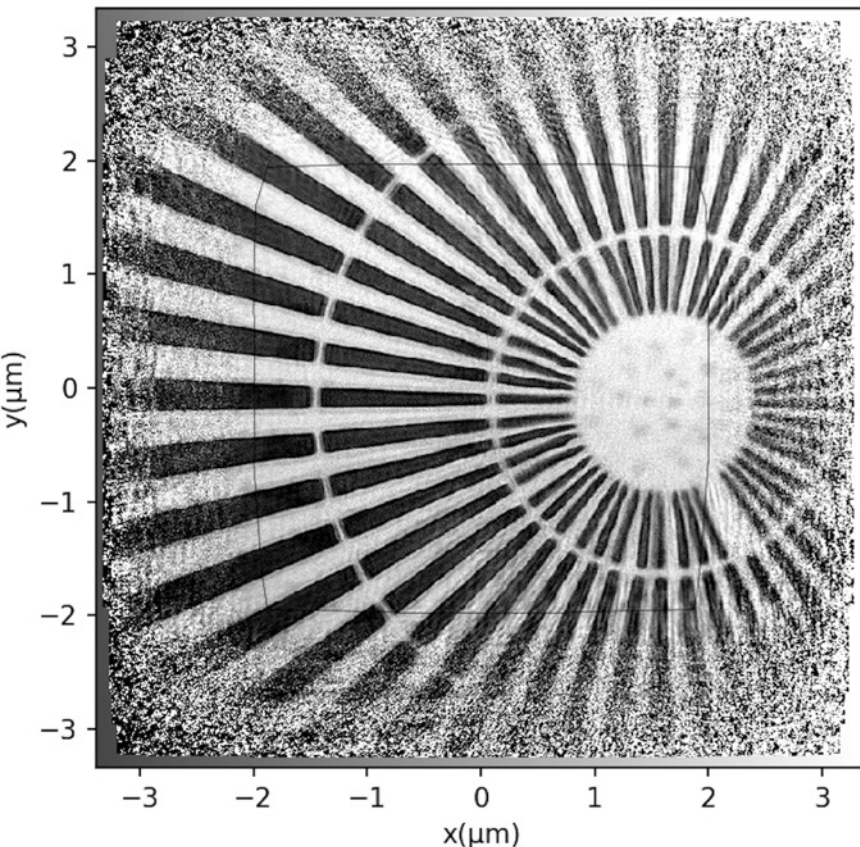
```
In [ ]: if not os.path.exists('ptycho-siemens-star-id01.cxi'):
os.system('curl -O http://ftp.esrf.fr/pub/scisoft/PyNX/data/ptycho-siemens-star-id01.cxi')
```

Load the data & setup the runner parameters

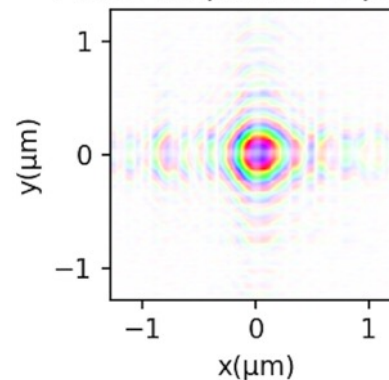
As we are using the `runner` API we can setup the parameters exactly as for the command line scripts

PTYCHOGRAPHY: PHASE & AMPLITUDE

Object phase [-0.12- 0.08 radians]



Probe amplitude & phase



id16A @17keV

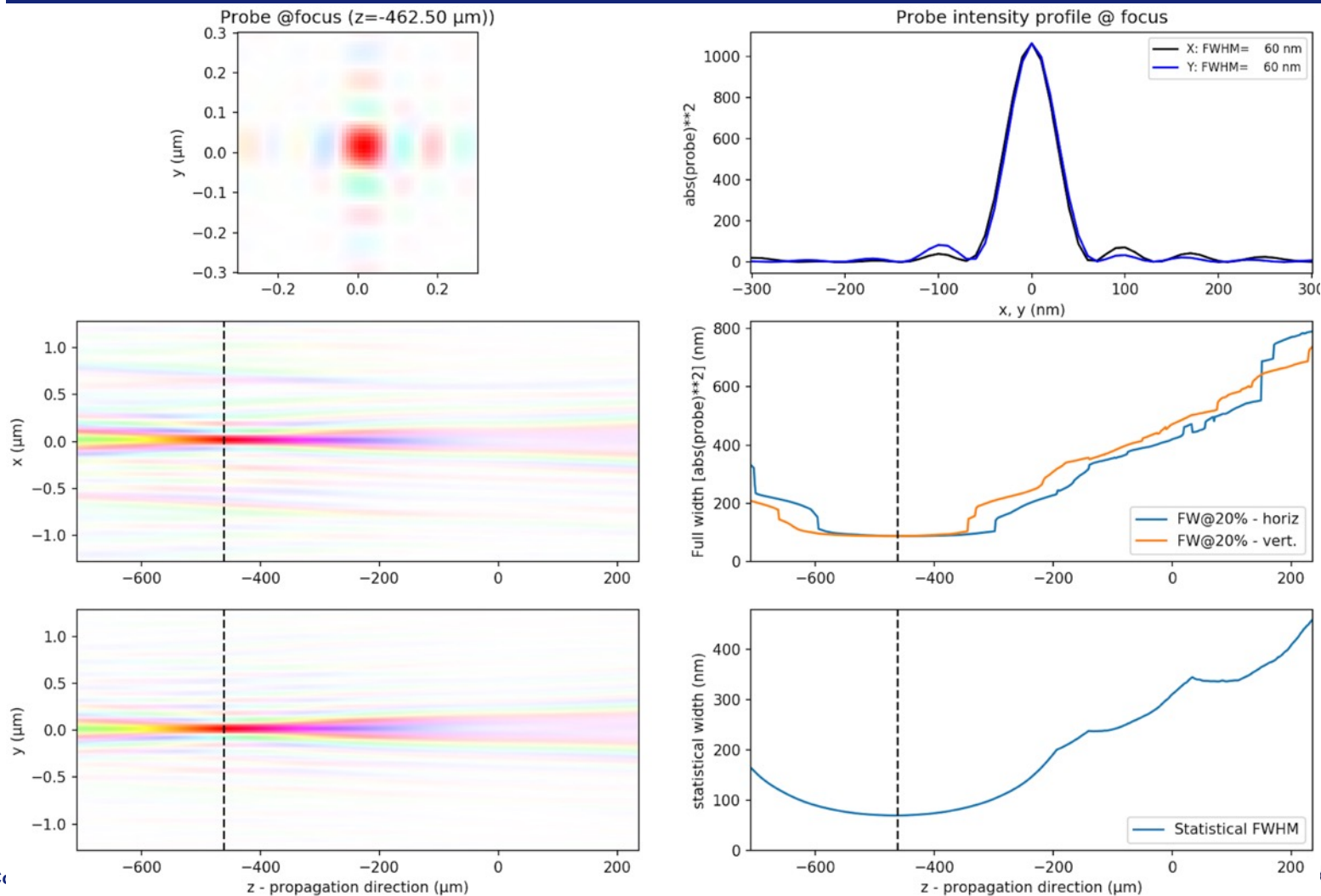
Lambda detector 256x256 GaAs

150 nm step, 125 ms/frame, 566 frames

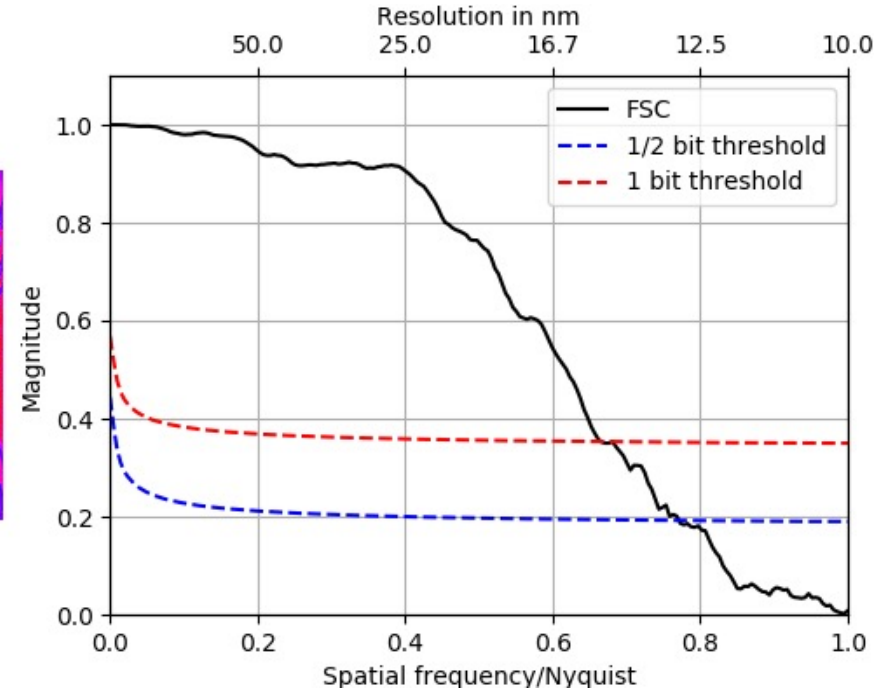
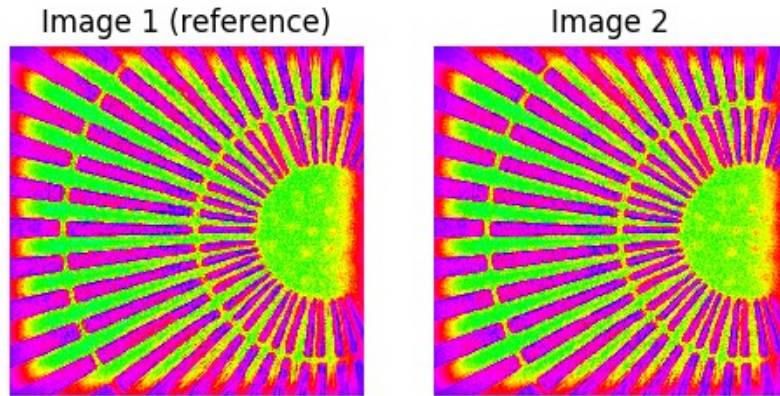
With filters (~20x) ☹

Both object and probe, amplitude and phase are recovered at the same time

PTYCHO: PROBE PROPAGATION



RESOLUTION: FOURIER SHELL (RING) CORRELATION

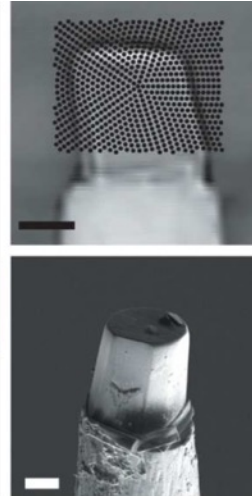
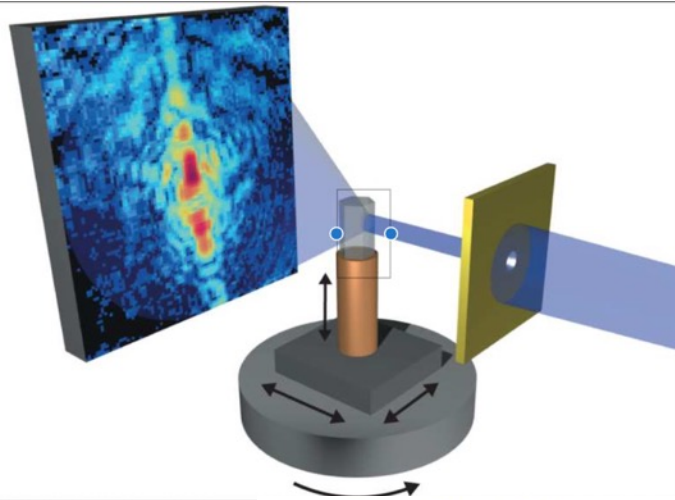


Resolution from Fourier shell (ring) correlation: ~15nm

(comparing two scans from the same area with different positions)

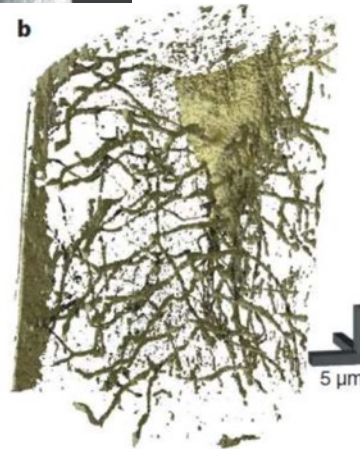
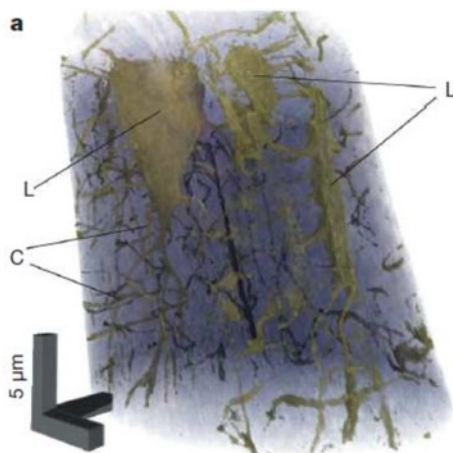
van Heel & M. Schatz, J. Struct. Biol. 151(2005), 250

PTYCHOGRAPHY-TOMOGRAPHY

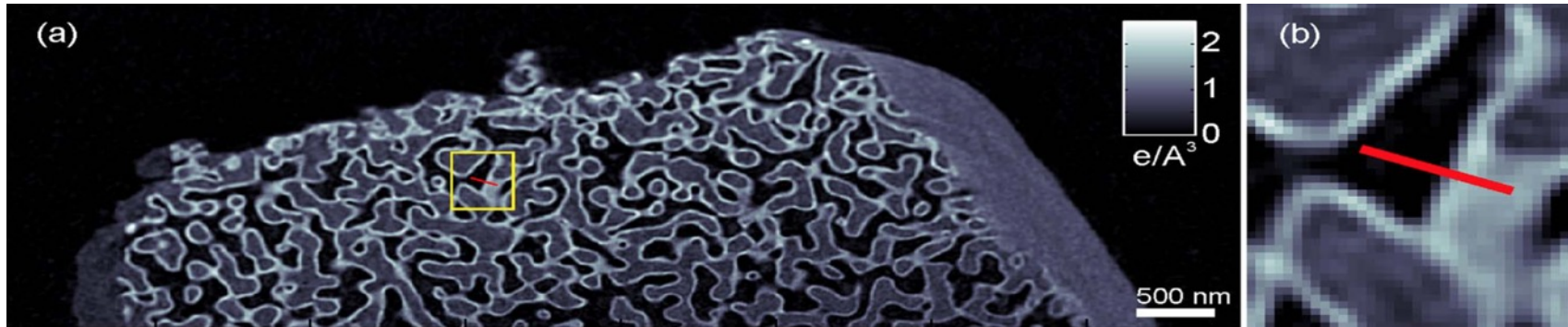


Experiments at cSAXS beamline SLS
Dierolf et al
Nature, 467, 436-439 (2010)

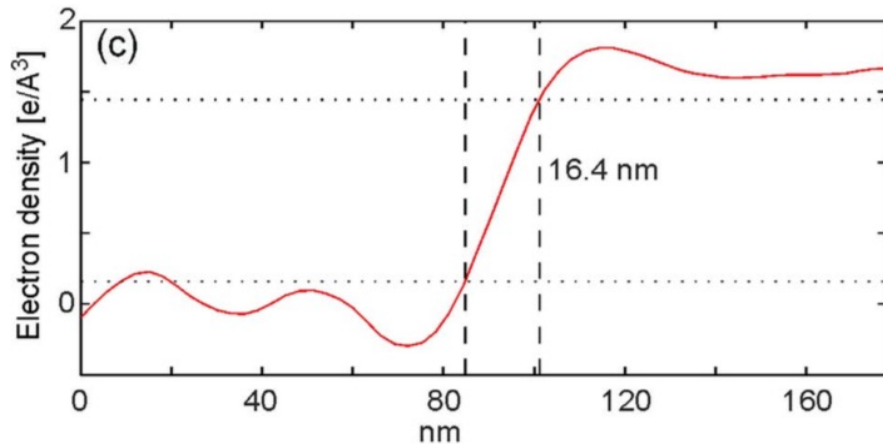
**Mouse femur bone, imaged in 3D at
120nm resolution
Voxel size : 65 nm**



PTYCHOGRAPHY: 3D HIGH RESOLUTION



Test object : porous SiO_2 structure of 139 nm mean pore size + Ta_2O_5 coating



3D reconstruction resolution : 16 nm

From :

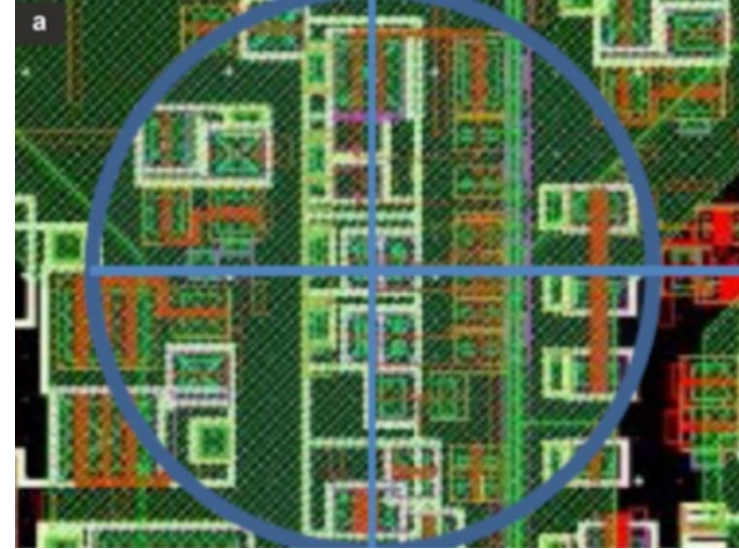
- 720 angular positions
- For each angle, 180 CDI images
- → 129600 images

HIGH RESOLUTION IMAGING WITH PTYCHO-(TOMO)

Sample: micro-processor:

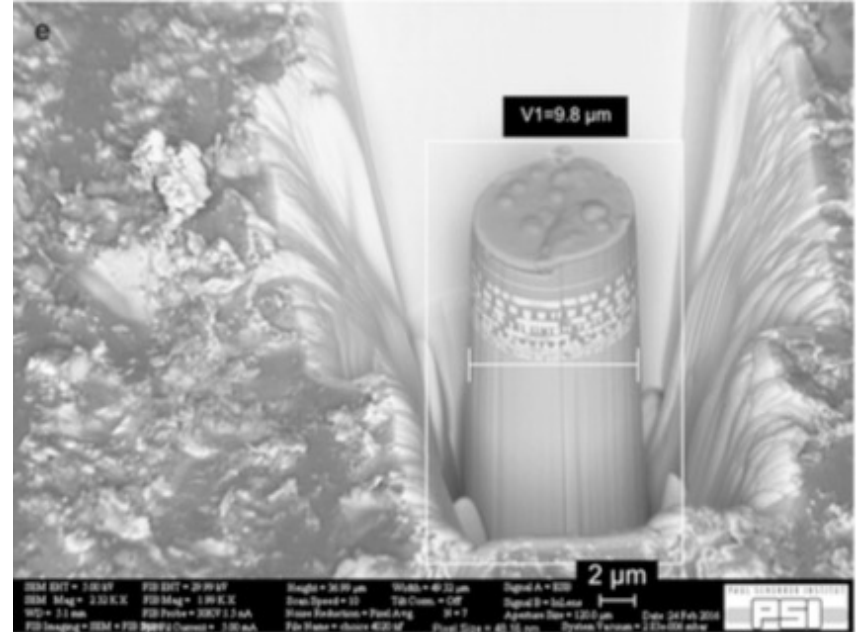
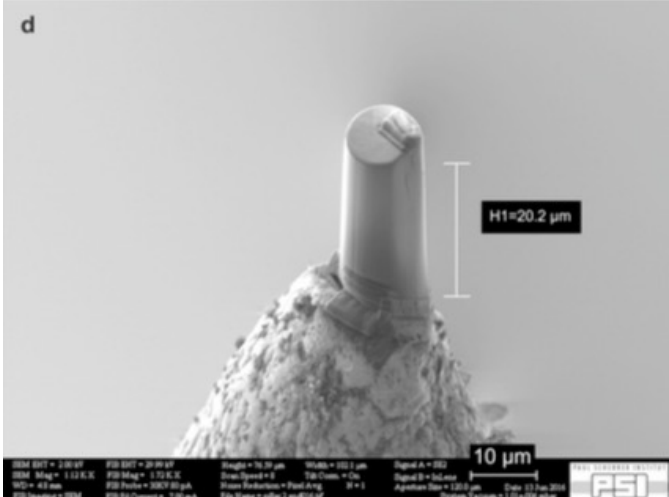
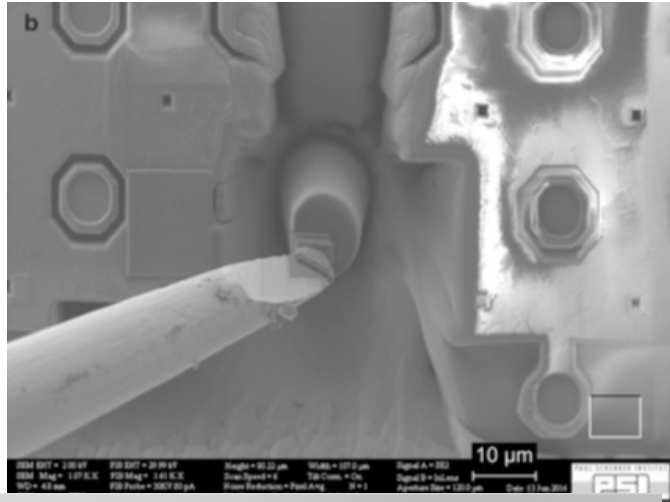


Socket LGA 1150, 3MB Cache, **22nm**



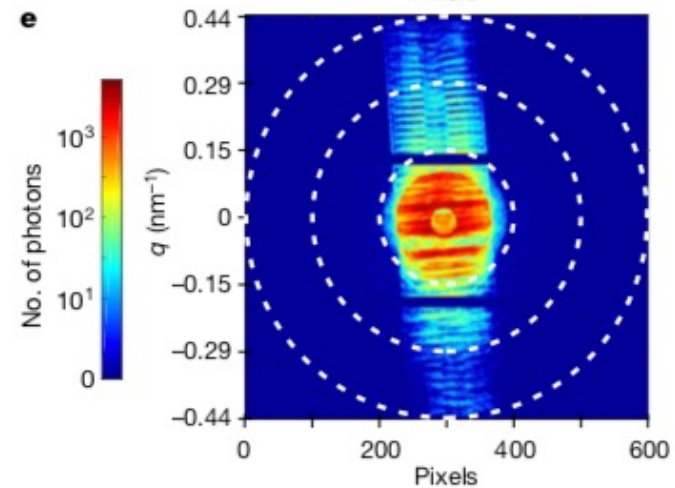
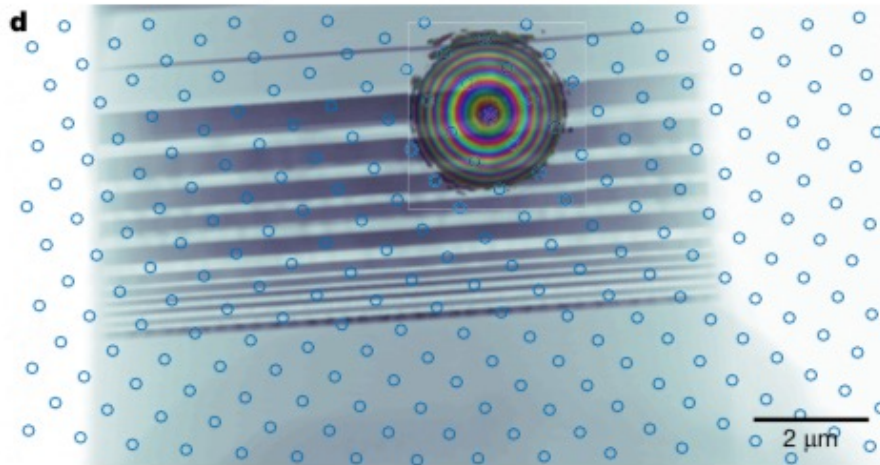
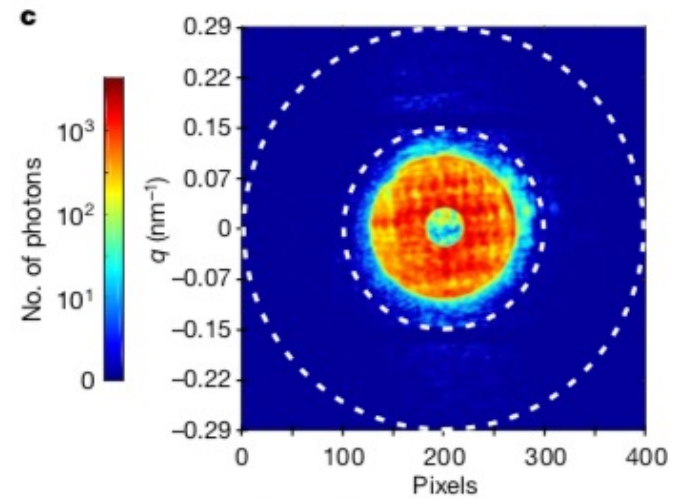
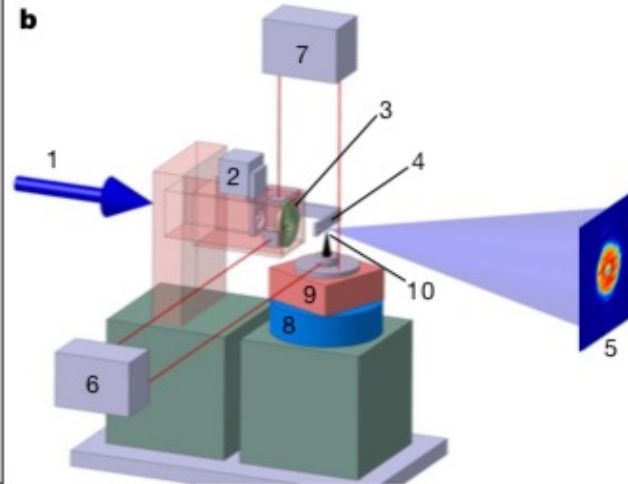
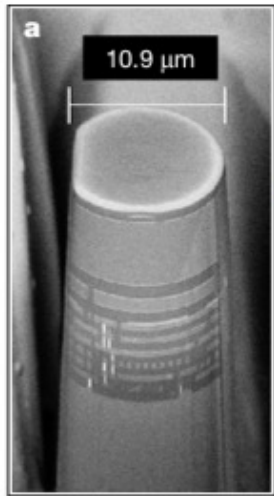
Select an area from the processor schematics

PTYCHO-TOMO OF INTEGRATED CIRCUITS

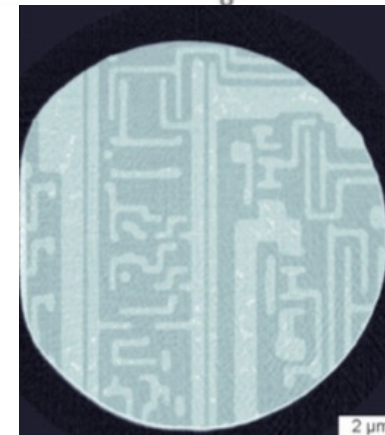
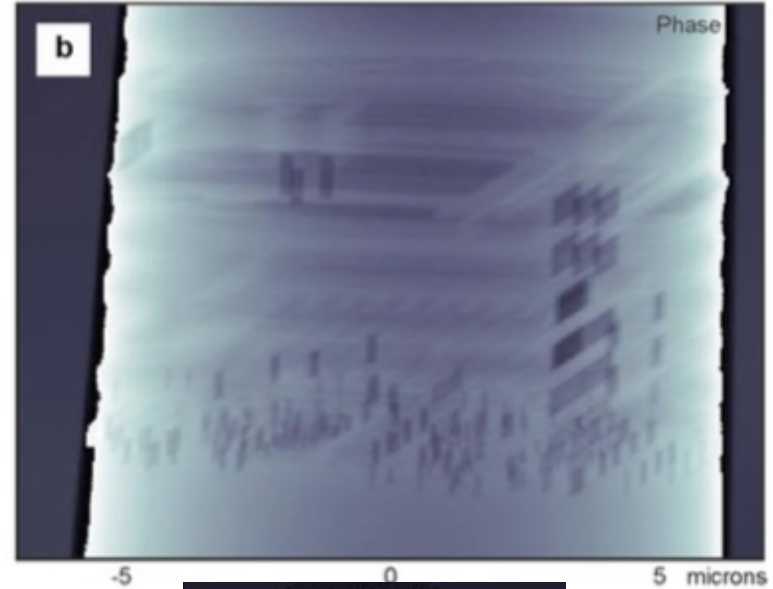
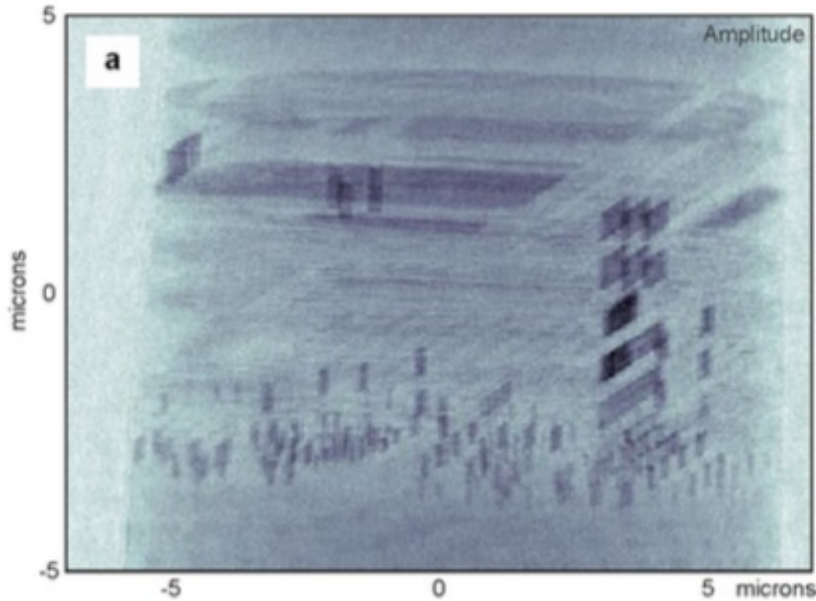


Extraction of a 10 µm pillar
Mount on a sample holder

PTYCHO-TOMO OF INTEGRATED CIRCUITS



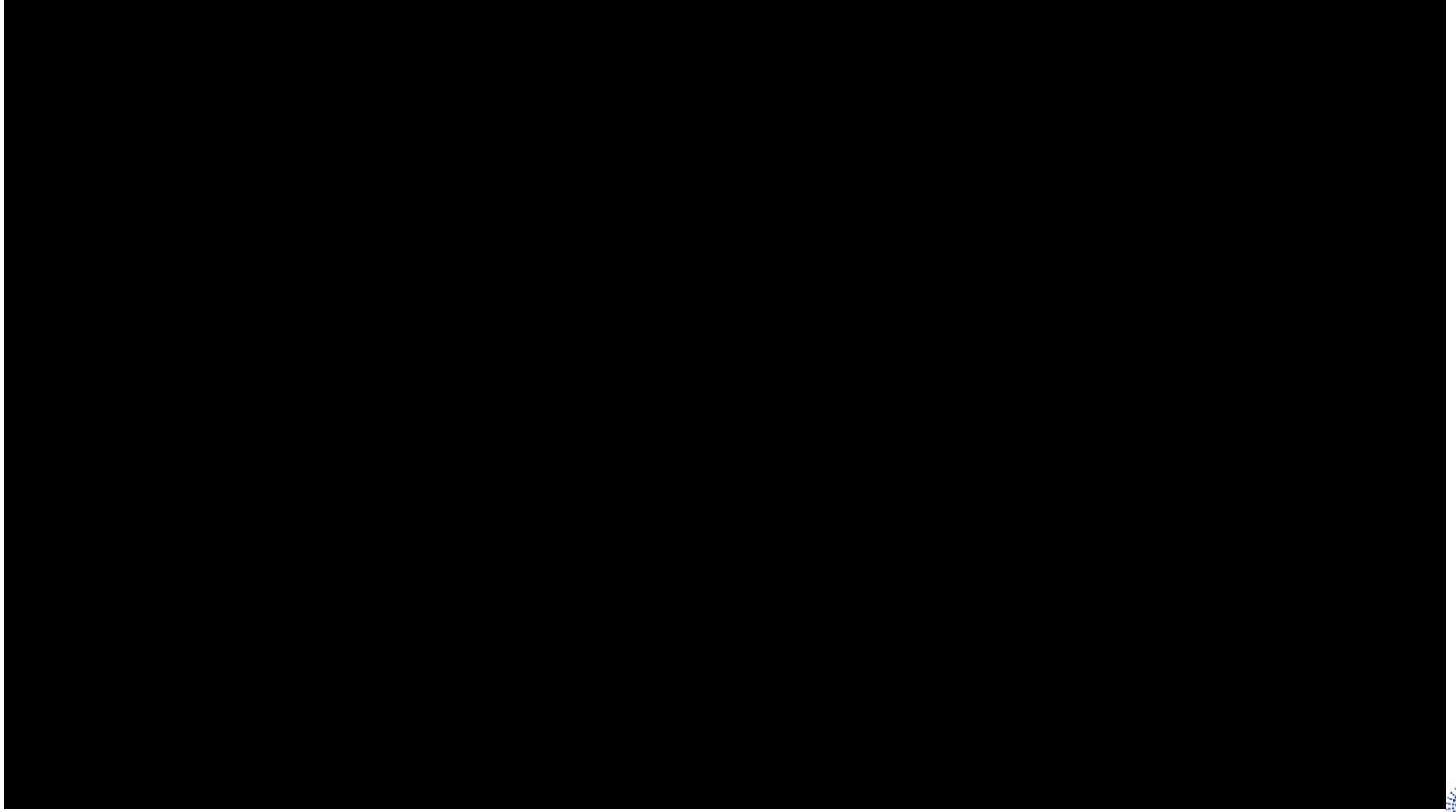
PTYCHO-TOMO OF INTEGRATED CIRCUITS



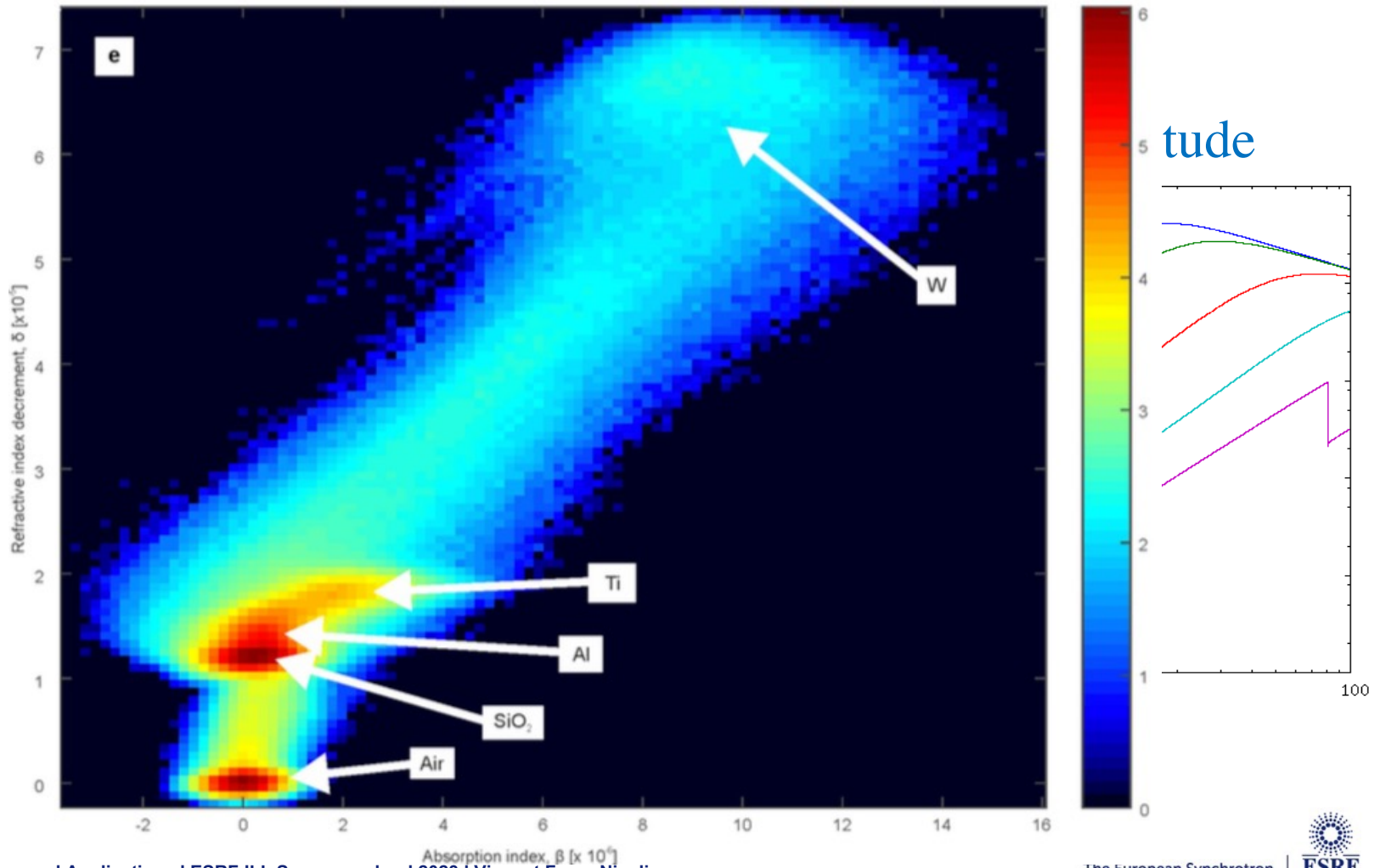
Result:

- 14.6 nm 3D resolution
- 1200 projections
- 24 hours
- 5850 resolution elements per second

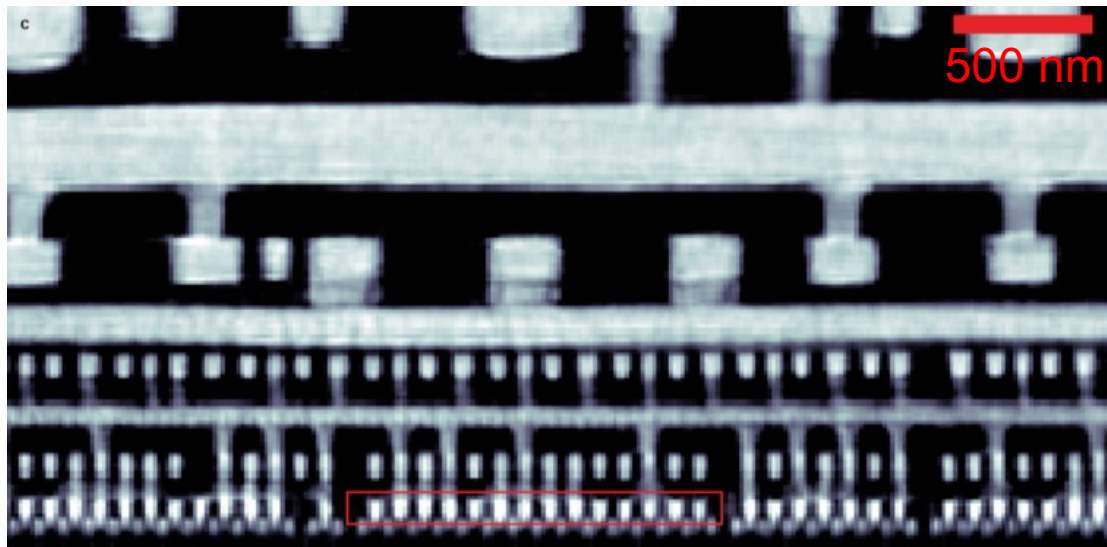
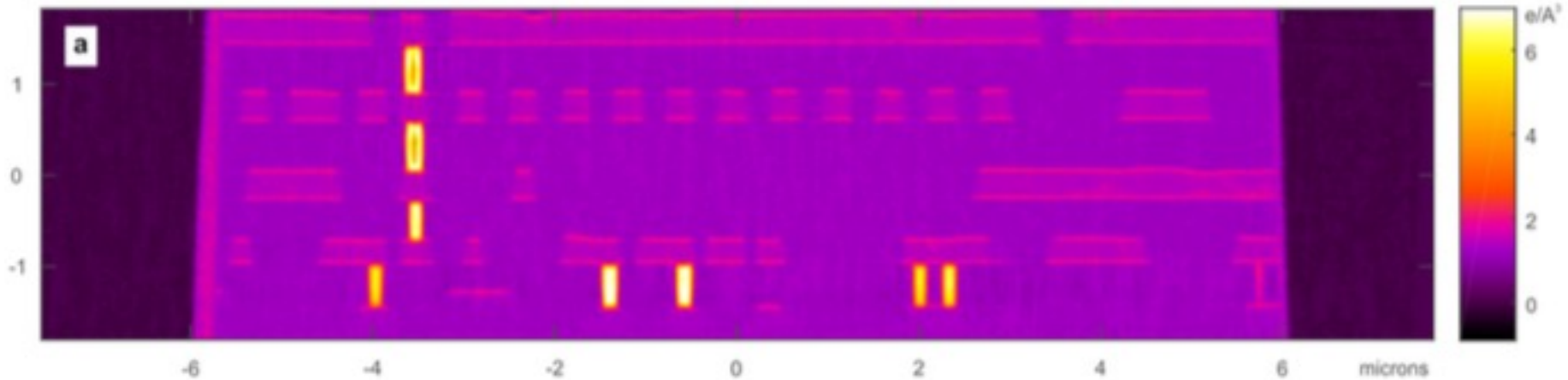
PTYCHO-TOMO OF INTEGRATED CIRCUITS



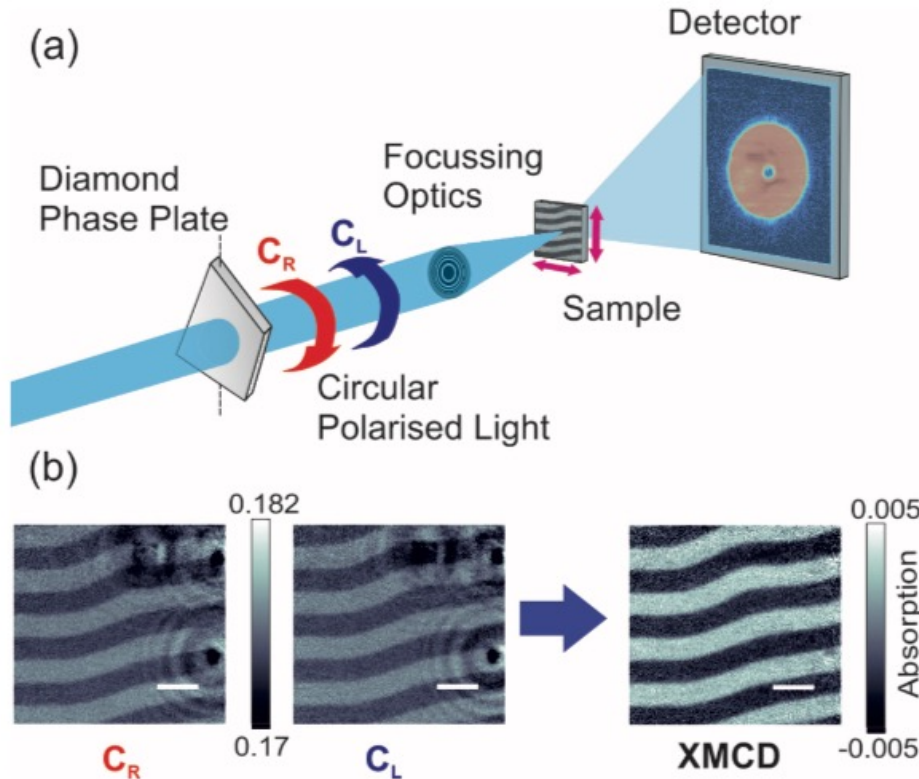
PTYCHO-TOMO OF INTEGRATED CIRCUITS



PTYCHO-TOMO OF INTEGRATED CIRCUITS



IMAGING MAGNETIC NANO-STRUCTURES WITH PTYCHO-TOMOGRAPHY

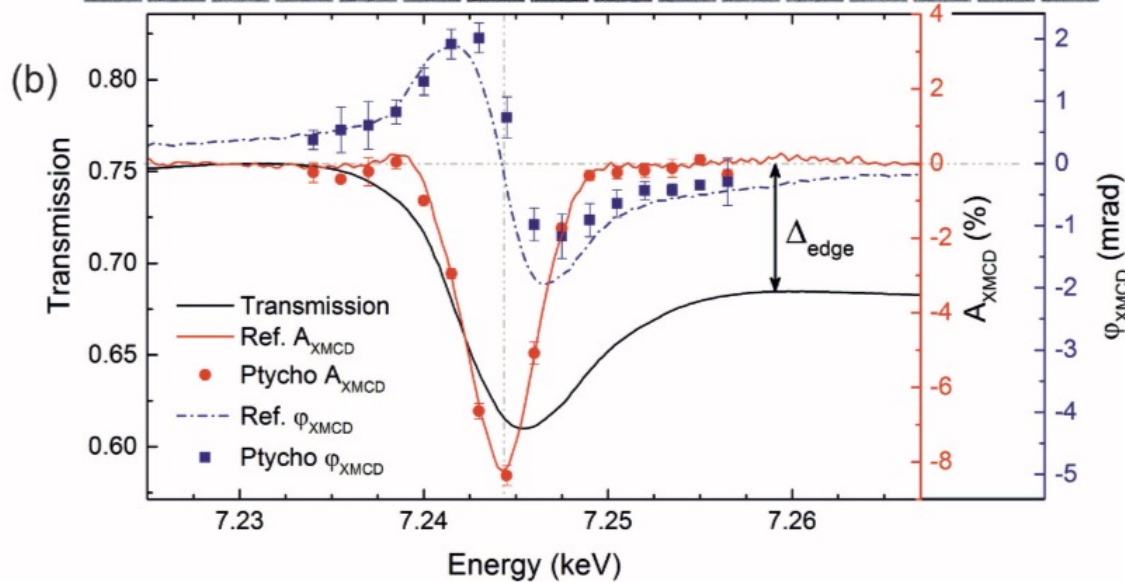
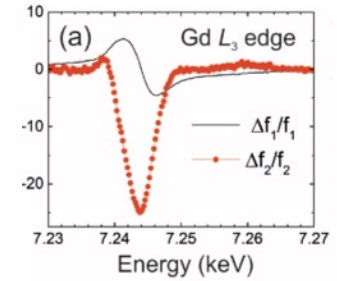
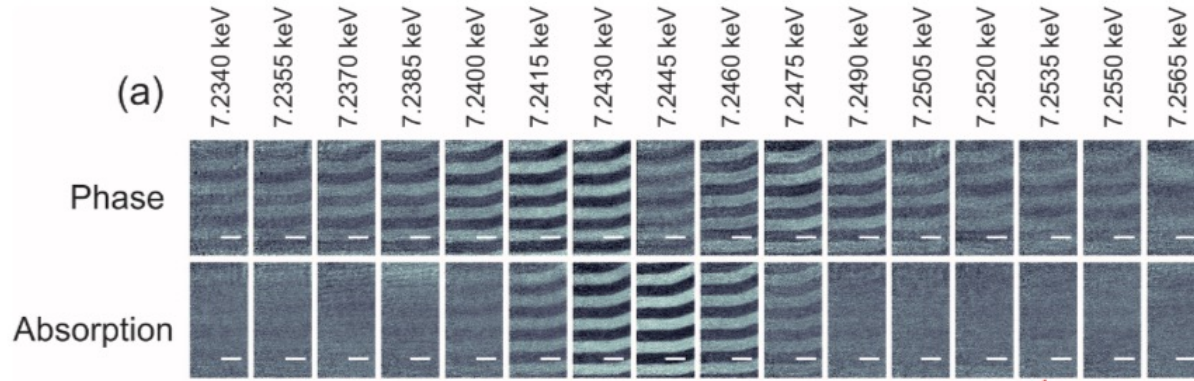


Synchrotron X-rays are normally (*) linearly polarised

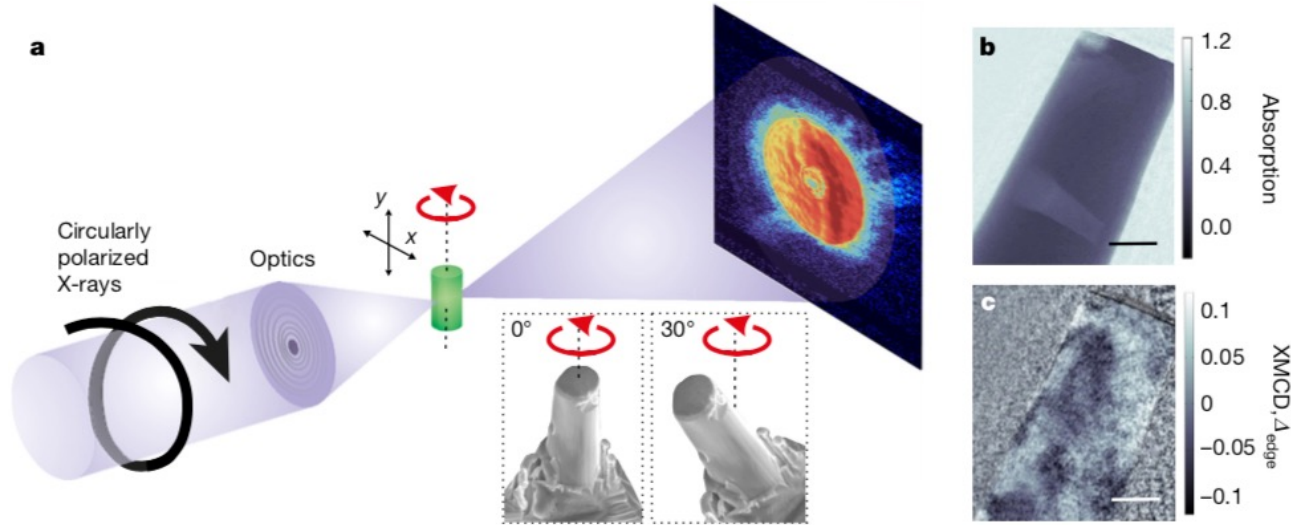
A phase plate can be used to transform this into a circularly-polarised X-ray beam

Sensitivity is maximal when magnetic moment is // to X-ray photon wavevector

IMAGING MAGNETIC NANO-STRUCTURES WITH PTYCHO-TOMOGRAPHY



IMAGING MAGNETIC NANO-STRUCTURES WITH PTYCHO-TOMOGRAPHY

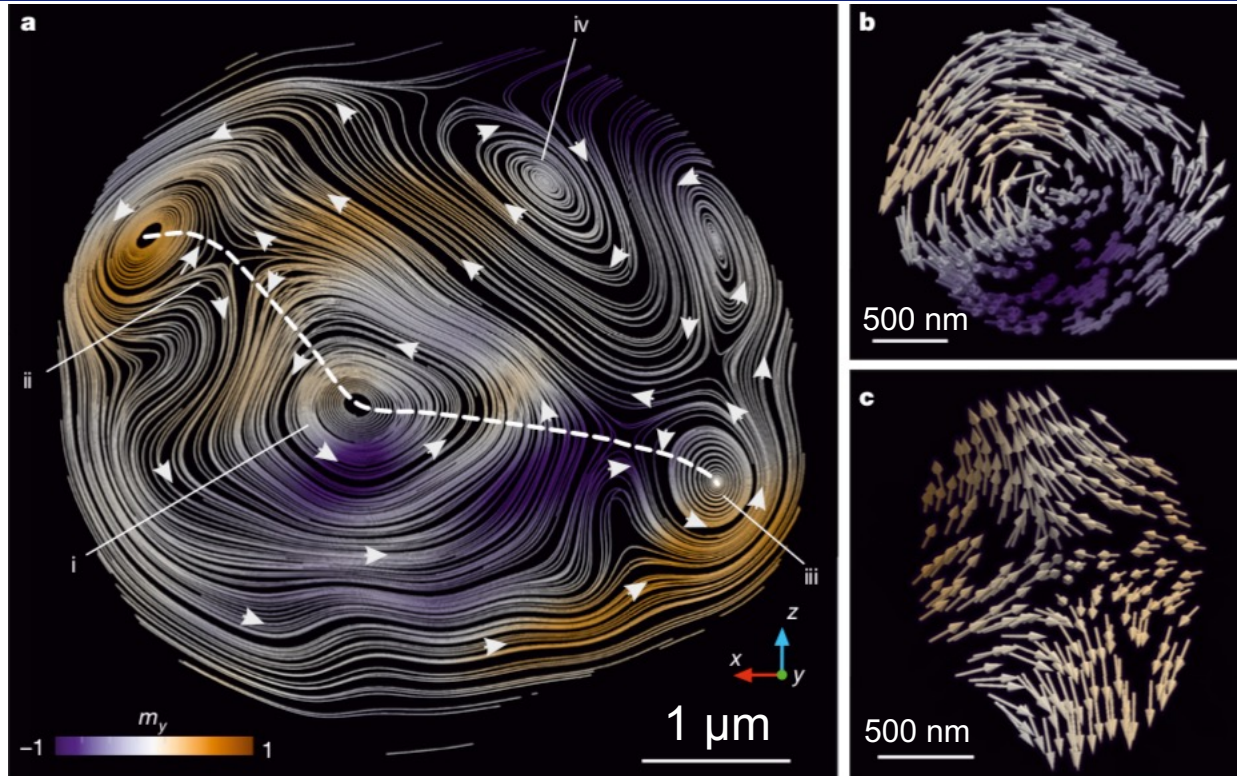


Setup for a ptycho-tomography setup

Using two tilt angles to be sensitive to the 3D orientation of magnetic domain

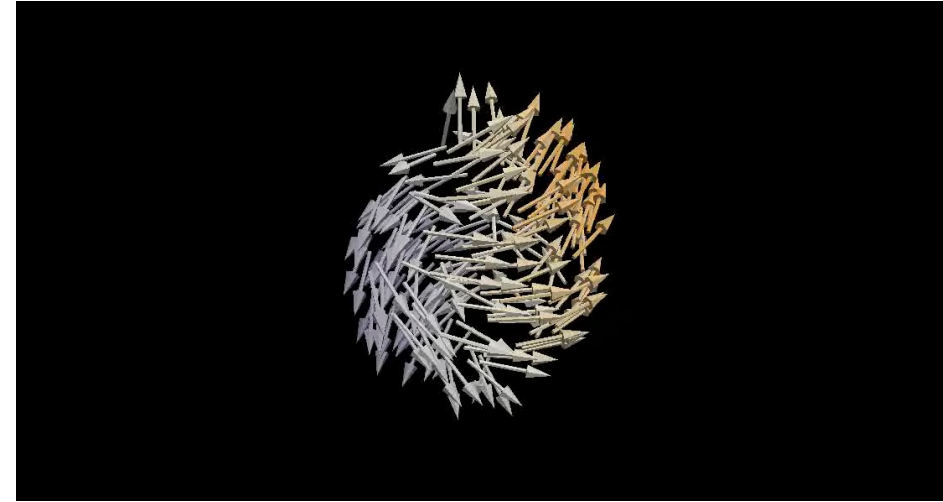
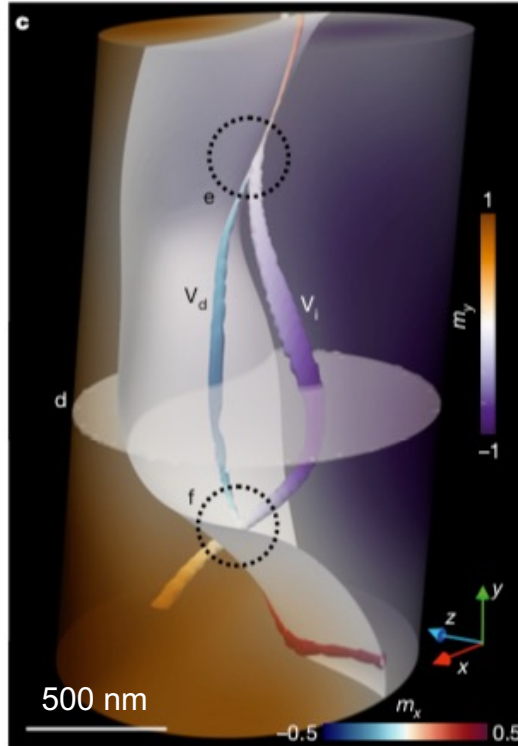
!! Magnetic structure is a **3D vector field**, not a 3D scalar !!

IMAGING MAGNETIC NANO-STRUCTURES WITH PTYCHO-TOMOGRAPHY



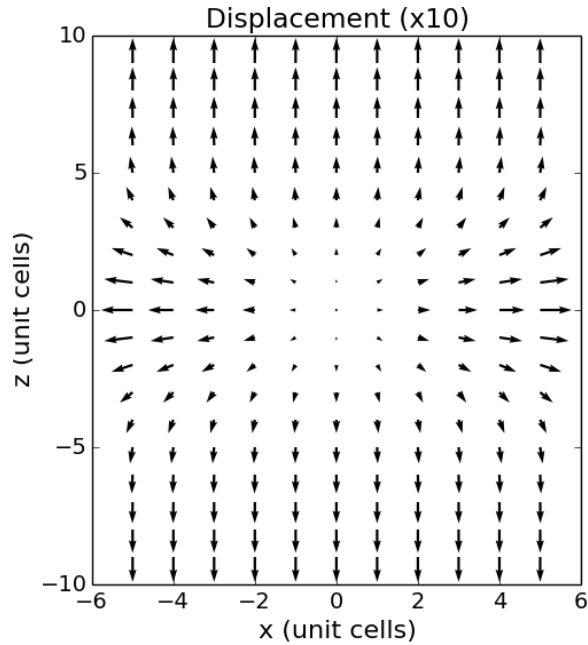
Axial tomographic slice of the reconstructed magnetization vector field
Note the anti- and clockwise vortices
Spatial resolution $\sim 100\text{-}200$ nm

IMAGING MAGNETIC NANO-STRUCTURES WITH PTYCHO-TOMOGRAPHY



3D view of the magnetic reconstruction with two main domains
Two vortices V_i and V_d intersect at two Bloch points ($m=0$)

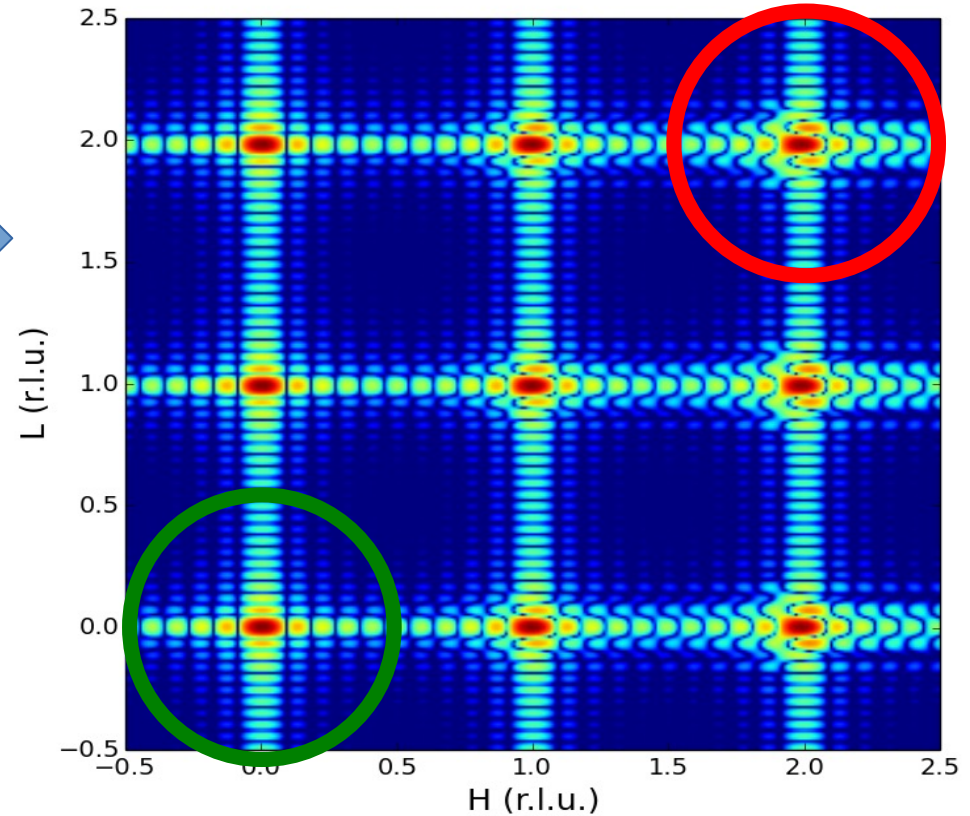
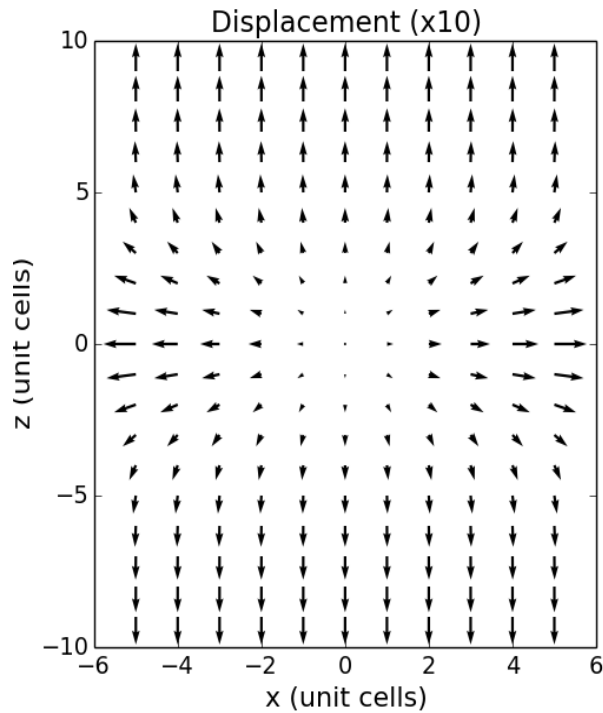
$$A(\vec{k}) \approx FT[\Omega(\vec{r})e^{2i\pi\vec{s}\vec{u}}]$$



- **Simulated nanowire w/insertion**
- - 2% strain along x

SMALL ANGLE VS BRAGG CDI

$$A(\vec{k}) \approx FT[\Omega(\vec{r})e^{2i\pi\vec{s}\vec{u}}]$$



- **Simulated nanowire w/insertion**
- **- 2% strain along x**

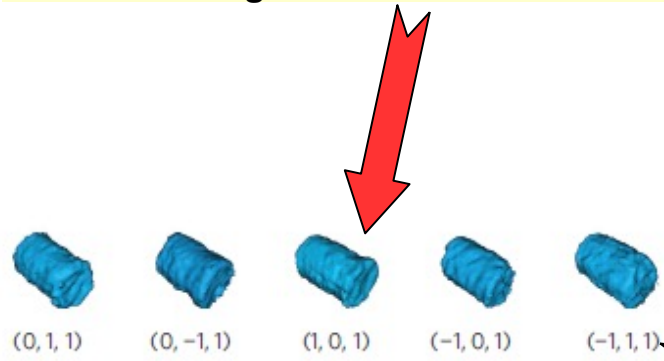
BRAGG CDI: STRAIN RECONSTRUCTION

(0,1,1) (0,-1,1) (1,0,1) (-1,0,1) (-1,1,1)



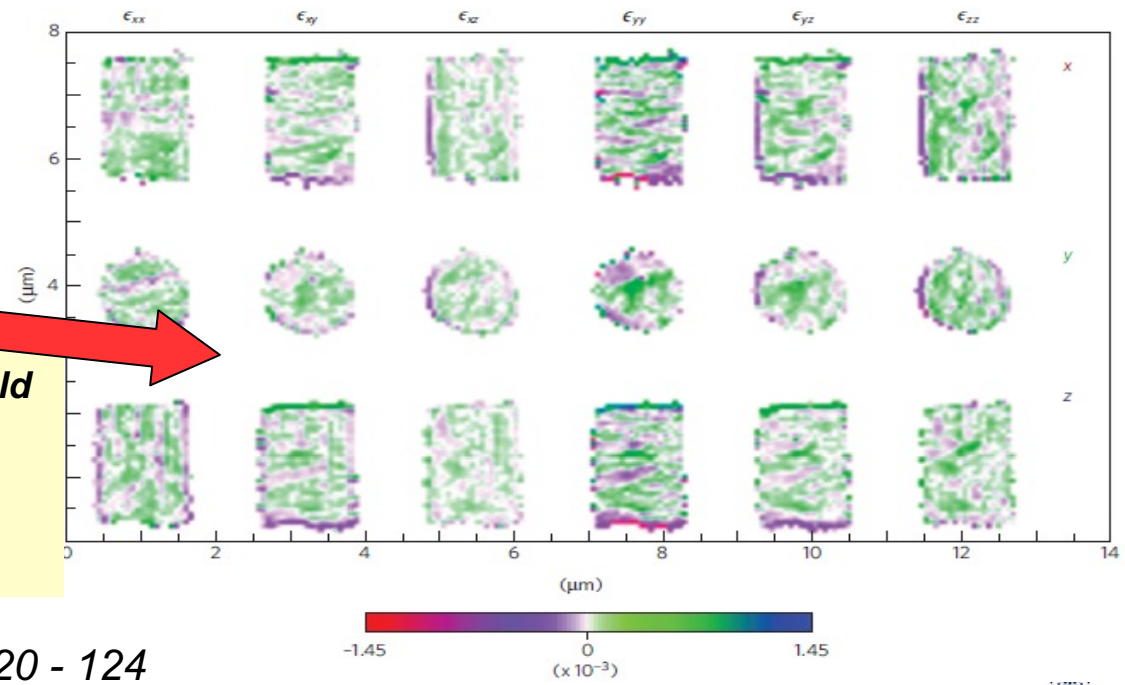
$$A(\vec{k}) \approx FT[\Omega(\vec{r})e^{2\pi i \vec{s}\vec{u}}]$$

- Measuring the CDI on at least 3 independent reflections



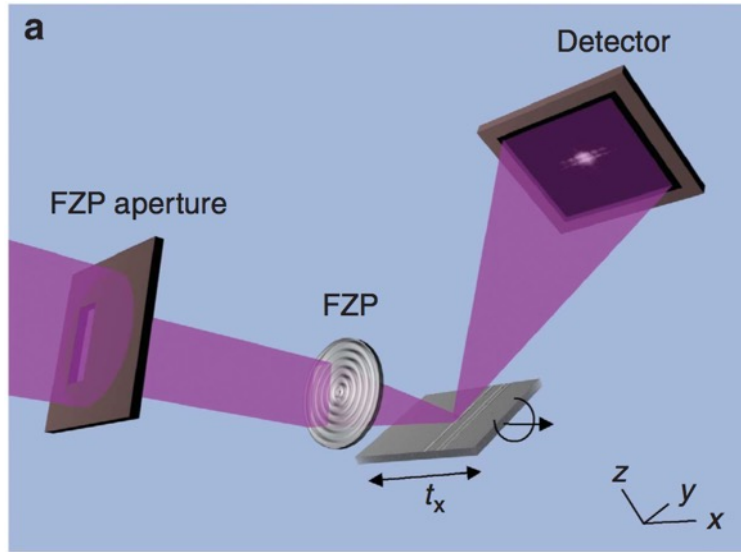
reconstruct the full 3D displacement field inside the nanocrystal

=> Access to the full strain tensor

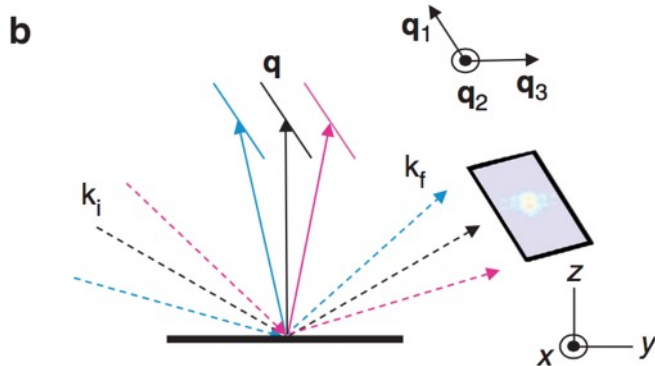


Newton et al., Nat. Mater. 9 (2010), 120 - 124

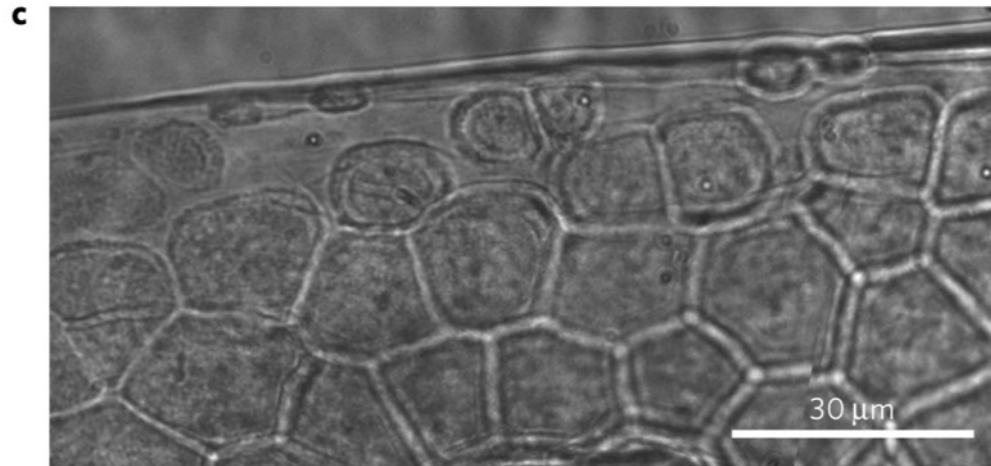
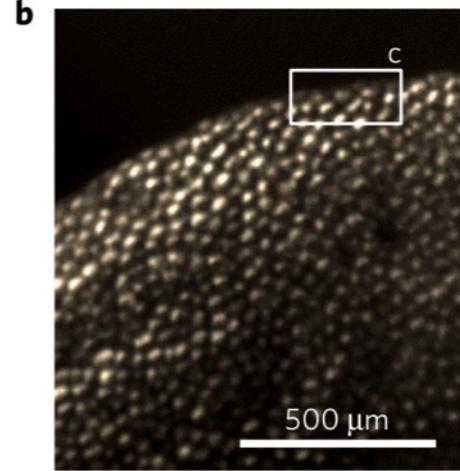
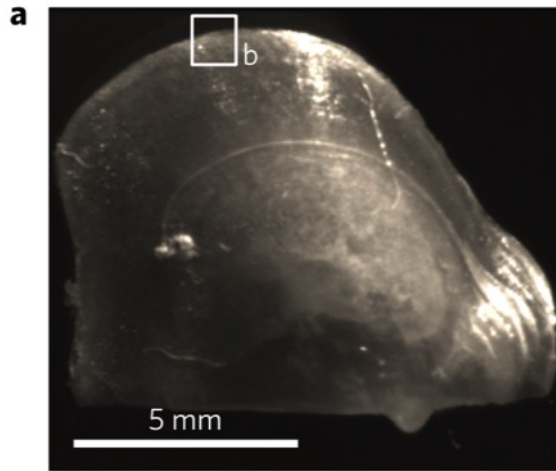
BRAGG PTYCHOGRAPHY



Bragg Ptychography :
2D Ptychography at different rotation angles
(angle step $\sim 0.04^\circ$, ptycho step 300 nm)

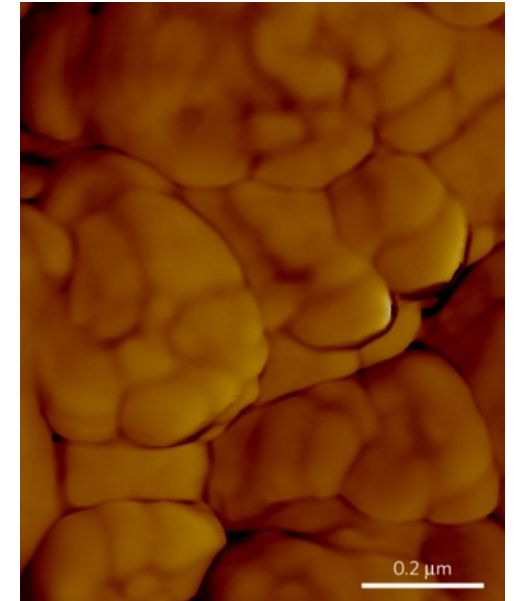


BRAGG PTYCHOGRAPHY ON MOLLUSC SHELLS

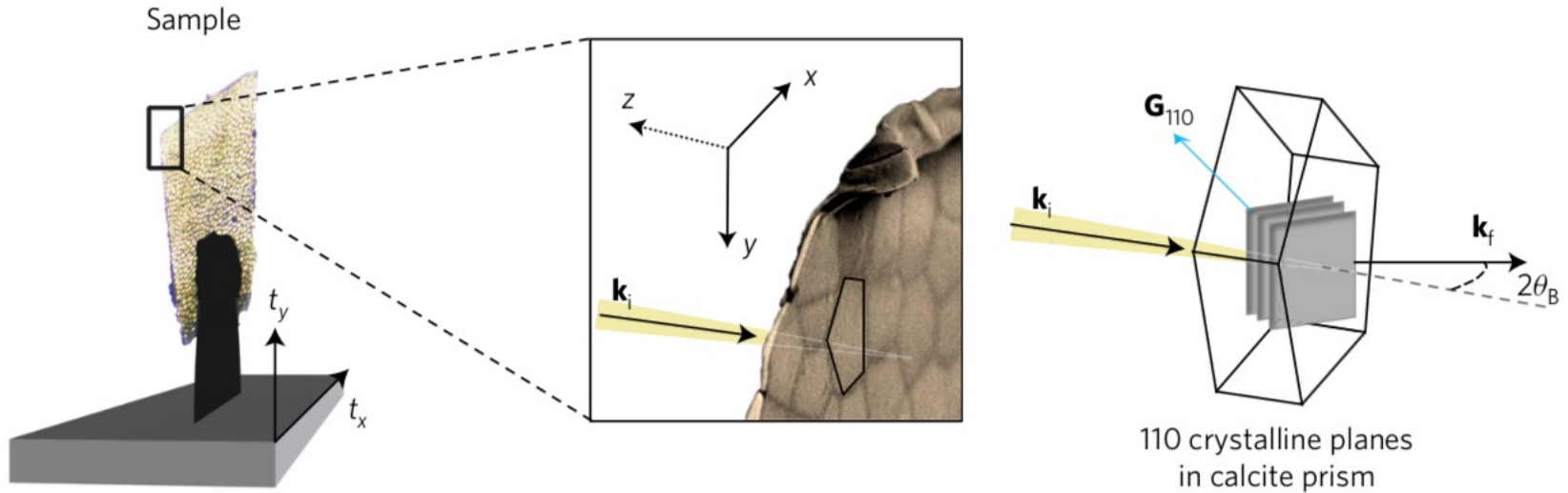


Pinctada margaritifera shell at different length scales

Calcite nano-structures

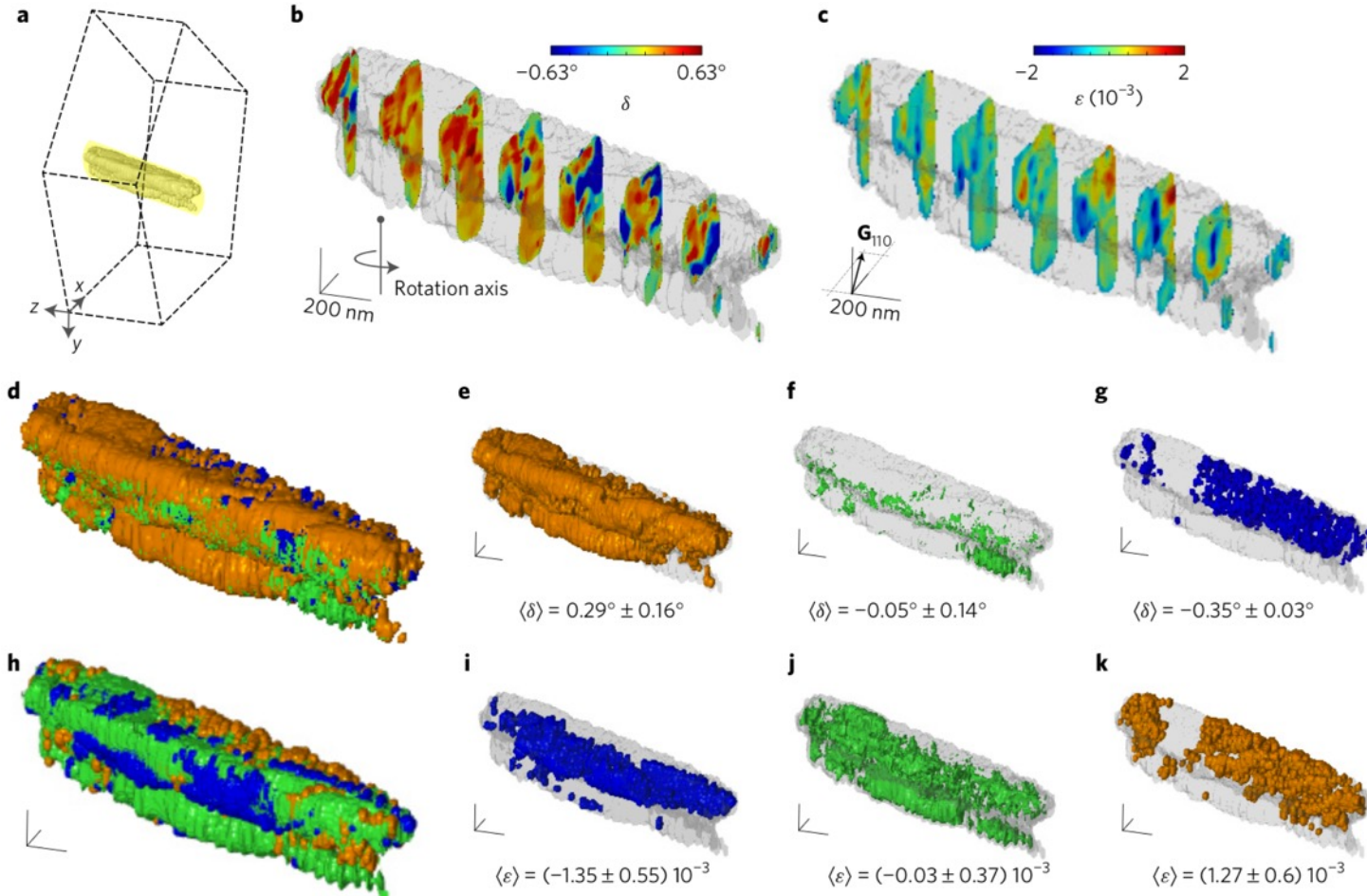


BRAGG PTYCHOGRAPHY ON MOLLUSC SHELLS



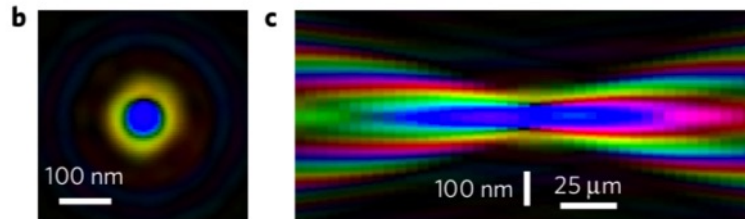
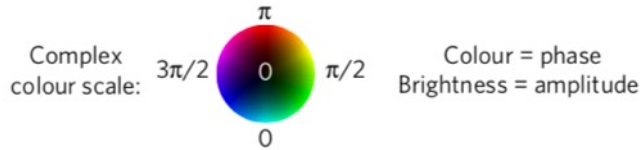
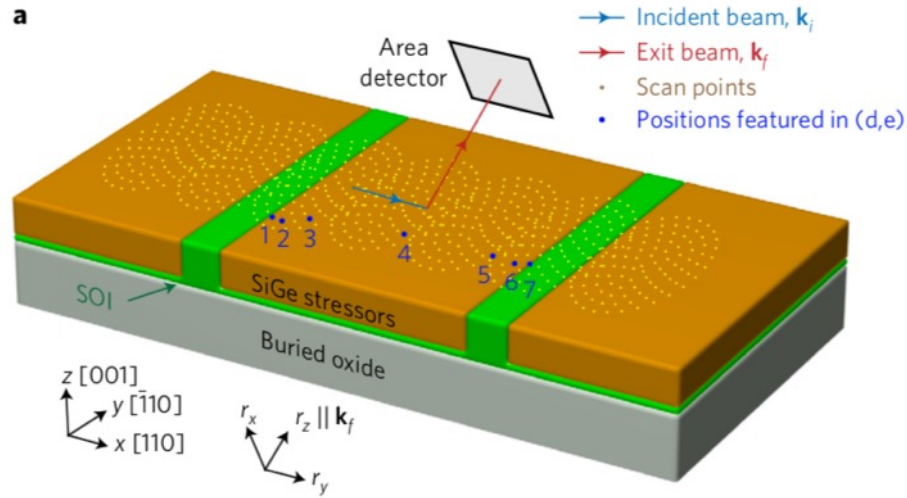
Diffraction on individual calcite prisms

BRAGG PTYCHOGRAPHY ON MOLLUSC SHELLS

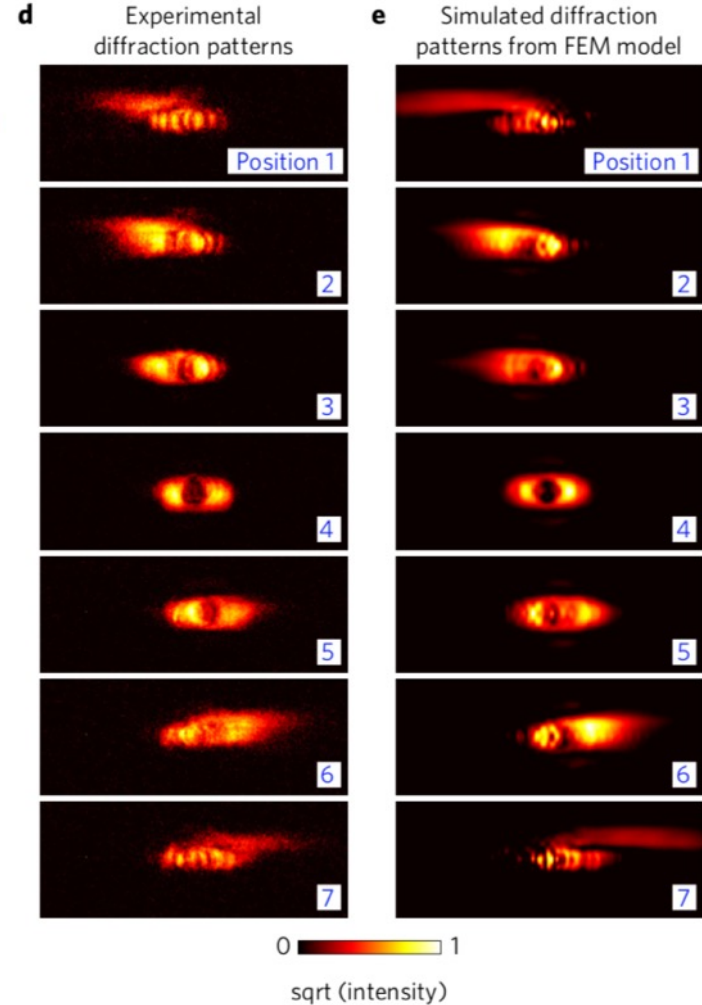


Bragg ptycho yields information on tilts at a micro/nano-scale

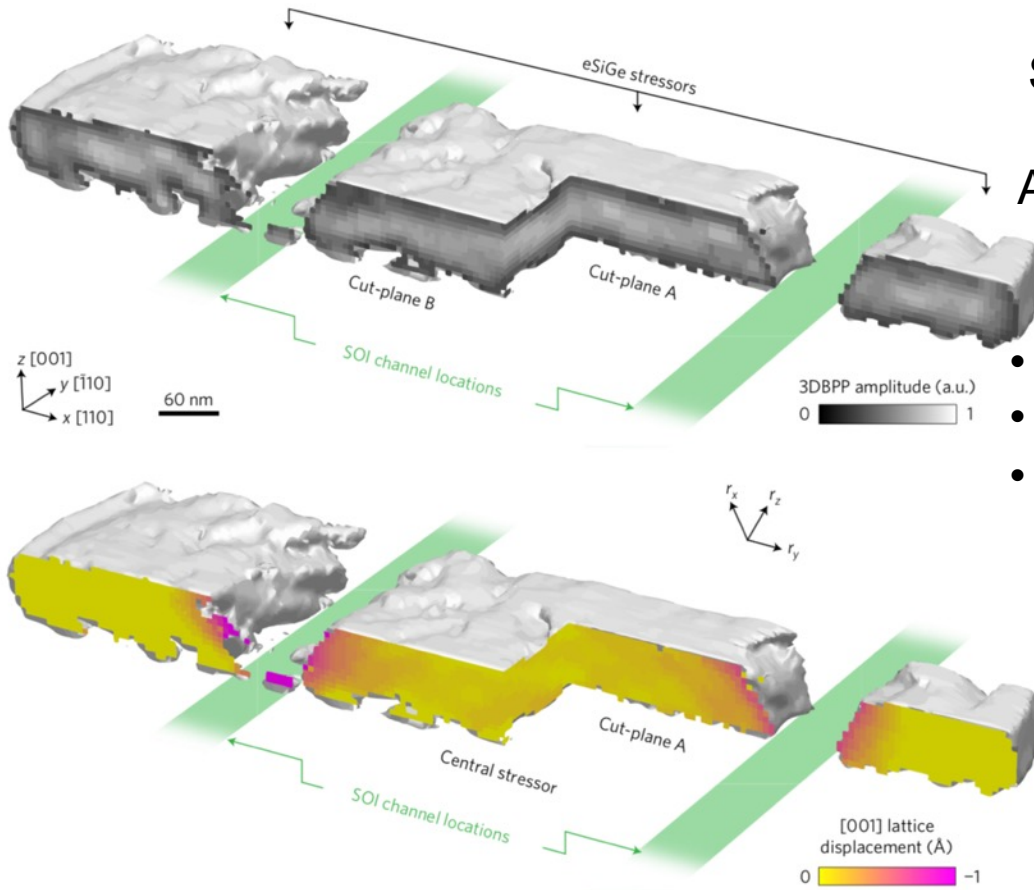
BRAGG PROJECTION PTYCHOGRAPHY



Single-angle Bragg ptychography



BRAGG PROJECTION PTYCHOGRAPHY

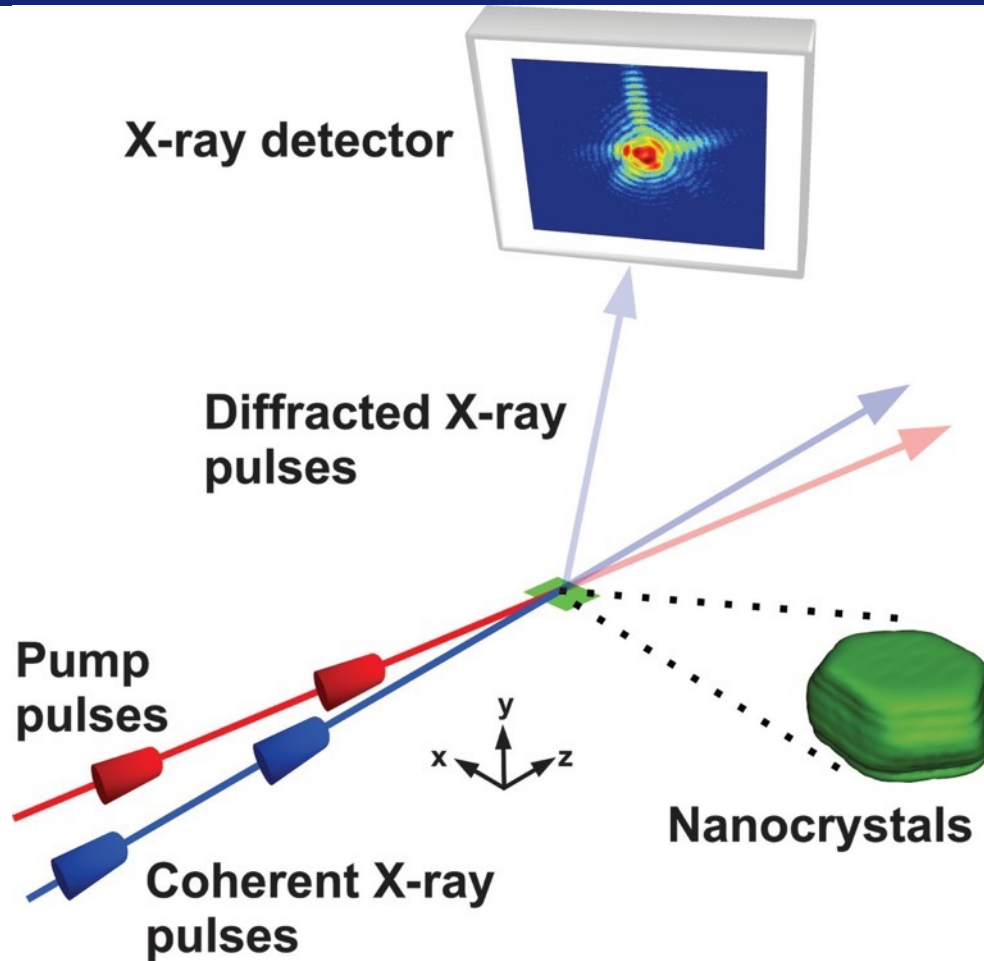


Single-angle Bragg ptychography

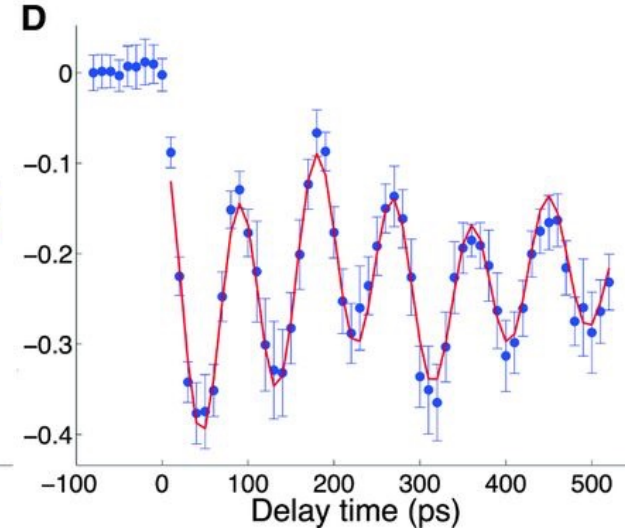
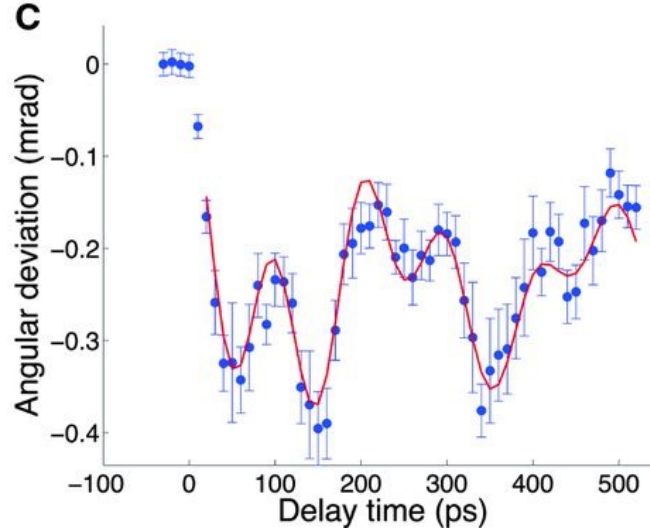
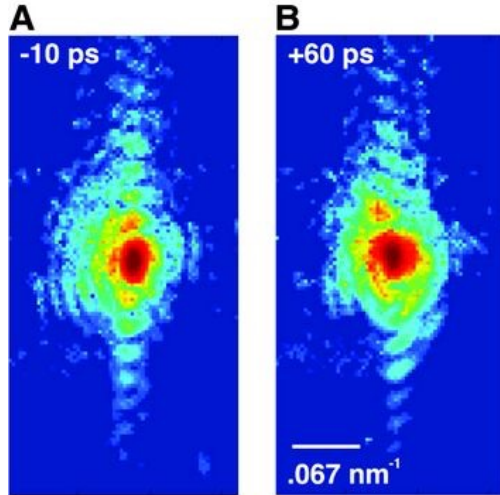
Allows reconstructing an extended object with limited data:

- Faster data collection
- Less radiation damage
- Only sensitive to deformation in the detector plane

XFEL LATTICE VIBRATION IMAGING USING BRAGG CDI



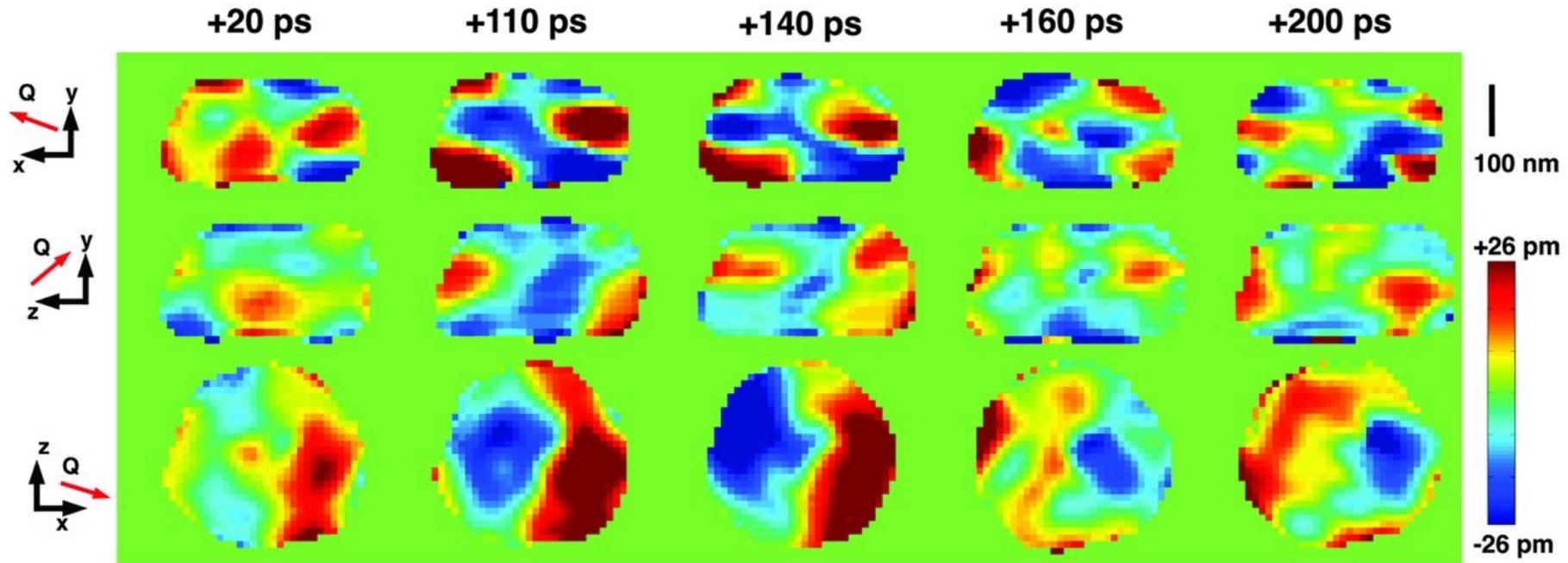
XFEL LATTICE VIBRATION IMAGING USING BRAGG CDI



Experimentally recorded coherent diffraction patterns from a single nanocrystal for delay times of -10 and +60 ps.

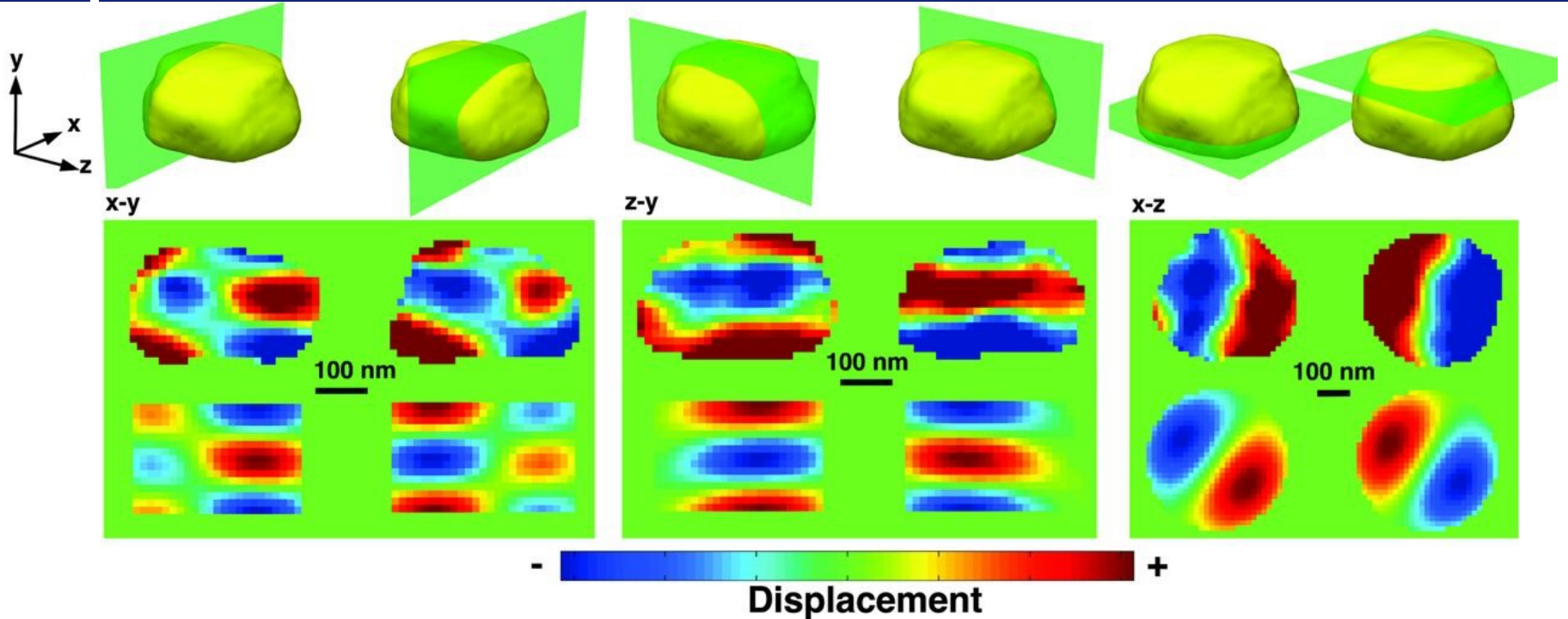
Bragg peak angular shift as a function of delay time for two different nanocrystals

XFEL LATTICE VIBRATION IMAGING USING BRAGG CDI



Orthogonal cut planes through the center of the nanocrystal showing the projected displacement as a function of delay time

XFEL LATTICE VIBRATION IMAGING USING BRAGG CDI



Orthogonal slices taken either side of the center (top) of the nanocrystal compare the projected displacement obtained from the experiment (middle) with a simulated (1, 1) mode for a cylinder (bottom).

THANK YOU FOR YOUR ATTENTION !

